

Longitudinal photocurrent spectroscopy of a single GaAs/AlGaAs v-groove quantum wire

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Abstract

Modulation-doped GaAs v-groove quantum wires (QWRs) have been fabricated with novel electrical contacts made to two-dimensional electron-gas (2DEG) reservoirs. Here, we present longitudinal photocurrent (photoconductivity/PC) spectroscopy measurements of a single QWR. We clearly observe conductance in the ground-state one-dimensional subbands; in addition, a highly temperature-dependent response is seen from other structures within the v-groove. The latter phenomenon is attributed to the effects of structural topography and localization on carrier relaxation. The results of power-dependent PC measurements suggest that the QWR behaves as a series of weakly interacting localized states, at low temperatures.

1. Introduction

In recent years there has been considerable effort expended in producing high quality quasi-one-dimensional (1D) electronic systems, in order to study their fundamental physical properties. Various techniques have been developed in order to realize these systems in semiconductors; these include the selective depletion of a high mobility two-dimensional electron gas (2DEG) [1], cleaved edge overgrowth of quantum wells (QWs) [2], and vicinal growth on patterned substrates [3]. There is currently increasing interest in the electronic transport properties of alternative quasi-1D materials, such as carbon nanotubes [4] and DNA molecules [5].

The optical properties of semiconductor v-groove quantum wire (QWR) structures have attracted extensive research, due to their potential use in optoelectronic devices including quantum wire lasers [6]; however, much less attention has been devoted to their electronic properties. The possibility of constructing nanostructure device networks using low-dimensional interconnects necessitates a greater understanding of carrier transport into and through such wires [7]. Photocurrent (PC) spectroscopy provides a means

of directly investigating the excited state properties of a semiconductor structure. This is especially pertinent when injecting carriers from nanoscale electrical contacts into low-dimensional structures. For the case of QWRs, PC spectroscopy provides information on the specific real-space electronic transitions involved in transport, which cannot be obtained by standard conductance measurements.

Previous work in this area has mainly concentrated on *transverse* (i.e. perpendicular to the wires) photocurrent measurements of v-groove p–i–n structures [8, 9], involving large ensembles of QWRs [10]. These investigations provided information on the energy levels within QWR structures but have given only limited insight into their overall electrical transport properties. Saraydarov *et al* [11] have recently reported longitudinal PC measurements of 24 wires that suffered from significant parallel transport through the substrate and buffer layers.

In this paper we report temperature- and power-dependent *longitudinal* photocurrent spectroscopy measurements of a *single* QWR. We have grown modulation-doped GaAs QWRs in v-grooves with quasi-{111}A sidewalls. In order to electrically contact the wires, wide grooves with an aperture of $\sim 50\ \mu\text{m}$ have been fabricated at each end of the v-groove (see figure 1(b)): as the wire grows in the v-groove, a contiguous

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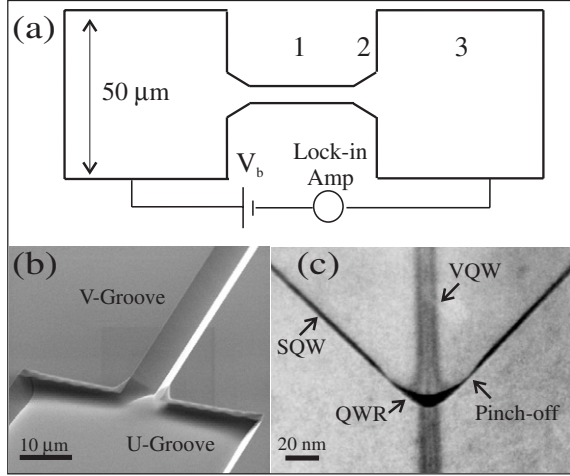


Figure 1. (a) Schematic diagram of a single groove (approximately to scale) showing the different regions of interest: (1) v-groove ($\sim 25 \mu\text{m}$ length), (2) 1D–2D transition ($\sim 5 \mu\text{m}$ length), (3) end contact 2DEG. (b) SEM image of one end of a v-groove showing the transition into the u-groove. (c) TEM cross-section of a v-groove showing the formation of a quantum wire, side QWs and a Ga rich vertical QW.

2DEG is formed in the (100) bottom quantum well (BQW) which grows on the bottom facet of the wide groove. This technique creates a tapering from 2D to 1D, and effectively eliminates the high contact resistance that occurs in sidewall contact techniques due to a large wavefunction mismatch between the sidewall quantum well (SQW) and QWR [12].

We have observed conductance in the ground-state one-dimensional subbands, and we have identified additional conductance features that originate from other structures within the v-groove. Small changes in lattice temperature reveal a strong dependence of excited state dynamics and relaxation on structural topography, and highlight the importance of carrier localization. We find that the power dependence of the PC response can be closely fitted by a rate equation model for a two-level quantum dot system. This indicates that, at low temperatures, the QWR behaves as a series of quasi-0D boxes rather than a quasi-1D system.

2. Experimental details

A semi-insulating (100) GaAs substrate was patterned with an array of $\sim 25 \mu\text{m}$ long $[01\bar{1}]$ -oriented v-grooves with $5 \mu\text{m}$ apertures; each groove was separated by $50 \mu\text{m}$ of SiO_2 mask. A nominal 2.5 nm GaAs quantum well (QW) was grown within $\text{Al}_{0.3}\text{Ga}_{0.7}\text{As}$ barriers by low-pressure metal–organic vapour-phase epitaxy. A 10 nm thickness of the barrier was modulation doped 30 nm above (below) the QW with a Si concentration of $8 (4) \times 10^{17} \text{ cm}^{-3}$. Polycrystalline growth occurs between the grooves due to the residual SiO_2 , which prevents the formation of a planar (100) QW at the top of the SQWs and ensures that each groove is electrically isolated [13]. Electrical contacts were made onto the 2DEG regions that terminate each groove (figure 1(b)). Low temperature current–voltage (I – V) measurements indicate that the v-groove structures are ohmic at low fields, and give a total wire resistance of $\sim 5 \text{ k}\Omega$. A detailed description of the growth and characterization of these quantum wires is given elsewhere [13].

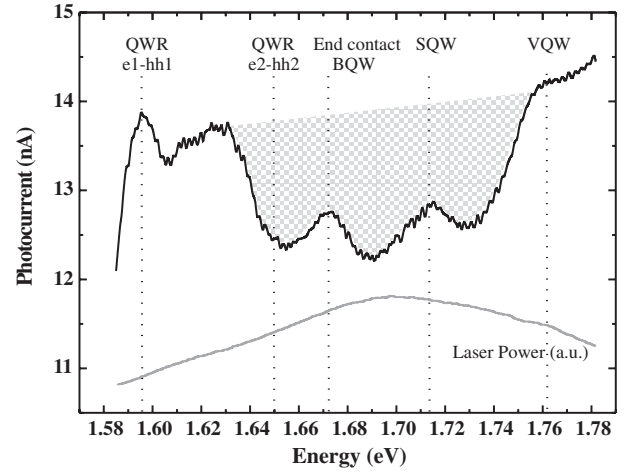


Figure 2. PC spectrum (10 K) obtained with a maximum power of $150 \mu\text{W}$ and a 80 mV bias. The shading indicates the region of reduced PC observed at low temperature. The dotted lines correspond to the positions of luminescence features. The tuning curve for a typical scan of the dye laser is also shown.

Figure 1(a) shows a plan-view schematic diagram of a single groove and the three different regions of interest: (1) the v-groove, (2) a 1D–2D transition region, (3) a wide u-groove containing a 2DEG onto which electrical contacts are made. A cross-section of a v-groove is shown in figure 1(c), obtained by transmission electron microscopy (TEM); a crescent-shaped GaAs QWR forms at the vertex of two SQWs, separated from them by a narrow constriction or pinch-off. A vertical quantum well (VQW) also forms at the centre of the groove, perpendicular to the substrate, due to the higher mobility of Ga atoms in the AlGaAs [14].

Photocurrent measurements were made using an Ar^+ pumped automatically tuneable dye laser (~ 1.6 – 1.8 eV), focused to a spot size of $\sim 35 \mu\text{m}$ on the v-groove surface. The sample was mounted on a variable temperature continuous flow cold finger, with a base temperature of 4.5 K. A DC voltage generator was used to apply a bias of a few hundred mV between the end contacts; by mechanically chopping the laser-beam, the resultant photocurrent was obtained with standard lock-in techniques.

3. Results and discussion

In figure 2 we present a PC spectrum taken at 10 K. The dotted vertical lines indicate the positions of photoluminescence (PL) features observed for the various groove structures [15]. The lines at 1.595 and 1.65 eV originate from e_1 – hh_1 and e_2 – hh_2 ($n = 1, 2$) recombination in the QWR respectively. These assignments have been confirmed by spatial mapping of the luminescence from the v-groove, using near-field scanning optical microscopy [16, 17], and also by theoretical modelling and PL-excitation (PLE) measurements [15]. A strong PC response can be seen at the $n = 1$ QWR energy, indicating efficient exciton generation and subsequent carrier transport in the ground state subbands. This is anomalous as the structure is supposedly heavily n-doped and hence this transition should be forbidden; however, there is also no obvious absorption edge seen at the Fermi energy in PLE measurements taken for

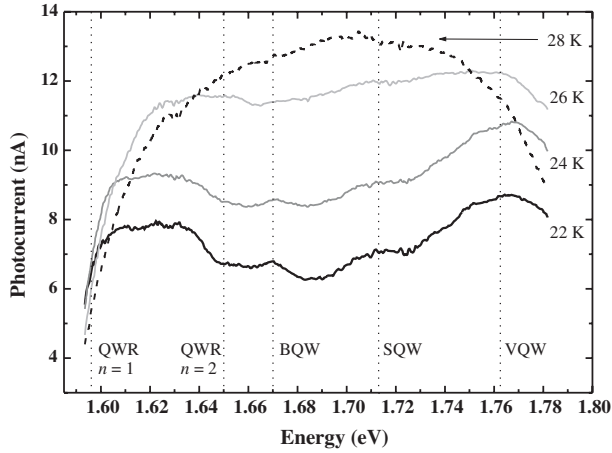


Figure 3. PC spectra for a single groove at different temperatures (200 mV, 30 μm spot, maximum power 4 μW). The dotted lines correspond to the features in figure 2.

this sample. The issue of modulation doping in these v-groove structures is complicated and subject to ongoing research. It is possible that the transfer of free carriers into the QWR is inefficient under certain growth or experimental conditions.

There is a prominent drop in the overall PC response between the low energy side of the $n = 2$ QWR transition and the VQW energy, as indicated by the shaded region in figure 2. This is evidence of reduced extraction efficiency of electron-hole pairs through the device at low temperature, as discussed below. Absorption is also clearly evident at the SQW energy, which is consistent with parallel conduction mechanisms; at low temperatures carrier relaxation from the sidewalls into the QWR is greatly restricted by the pinch-off regions [15]. Despite the large area presented by the SQWs for conduction, the PC response at this energy is relatively small. This is due to discontinuity of the $\{111\}$ A facet at the ends of the groove, which prevents efficient carrier transfer between the end BQWs and the SQWs (see figure 1(b)). Another strong peak occurs at the VQW energy; this is indicative of carrier transfer between the VQW and QWR [18], as parallel conduction is unlikely due to strong compositional disorder in the VQW.

3.1. Temperature characteristics

PC spectra were recorded at different temperatures and are presented in figure 3. With a small rise in temperature the absorption features disappear, and the PC increases greatly in the middle energy range. This is especially noticeable at the $n = 2$ QWR energy.

In the QWR, the e_2 electronic wavefunction is localized over the $\{311\}$ A facets which create the pinch-off [19]. These facets are subject to large interfacial disorder, which causes strong exciton localization [20], and greatly inhibited relaxation [21]. These issues are discussed in detail elsewhere [15]. At very low temperatures, localization effects play a dominant role in limiting carrier transport; this results in a reduction of the photocurrent contribution from certain parts of the structure, as seen in figure 2 between 1.63 and 1.75 eV. The radiative recombination rate for an exciton is found to be proportional to its coherence volume in the confining potential [22]. Excitons localized over lengths L

comparable to the excitonic Bohr radius (a_B) have a relatively long lifetime of up to 500 ps at low temperatures [20], and are thus susceptible to dissociation by hot carriers. In the limits of $L \gg a_B$ and $L \ll a_B$, the radiative lifetime tends to that of a free exciton, ~ 150 ps [23]. The $n = 1$ and 2 QWR states are expected to have different localization lengths as their respective wavefunctions are centred on crystal facets with different interfacial roughnesses [19]. Thus, at low temperature, excitons in the $n = 2$ QWR state may recombine before they are dissociated, and are not able to contribute to the photocurrent. As the lattice temperature is increased, these excitons are de-trapped and can relax more efficiently to the QWR ground-state where they are ionized within the recombination time. This results in an increase in the PC at the $n = 2$ QWR energy, as seen in figure 3. The PC at the SQW energy also increases with temperature due to a real-space transfer of carriers into the QWR [15, 17], which results in increased carrier-carrier scattering and smearing of the absorption features.

In this discussion we have assumed that radiative recombination is the dominant mechanism, as non-radiative processes are expected to become significant only at temperatures greater than the range considered here [24]. We have performed time-resolved measurements on similar QWR structures and obtain low temperature lifetimes of ~ 350 and ~ 150 ps for the $n = 1$ and 2 states respectively [15], which supports the above conclusions. Above 30 K, we observe a decrease in the relaxation time of the $n = 2$ state to the ground state, which is also consistent with these measurements.

3.2. Power characteristics

In figure 4 we show the PC response as a function of laser power, when illuminating at various energies. In all cases the PC increases rapidly with a small increase in power, and then saturates at a value that depends on the bias.

The high sensitivity of the PC to low powers explains why the spectrum in figure 2 shows a clear peak at ~ 1.6 eV while those in figure 3 do not. The laser is at the extreme of its tuning range at this energy, and the excitation power (and hence PC response) can vary significantly on different days depending on the fine tuning of the laser.

Beham *et al* [25] have observed a similar power-dependent PC response in single InGaAs quantum dots (QDs), which they attributed to bleaching of the dot ground-state. We outline their model below and discuss its relevance to a 1D wire system.

The saturation of the PC in a QD with increasing excitation power can be described by a fundamental rate equation model for a two-level system. This assumes the dot is in level 1 if it is unoccupied, and in level 2 if it is occupied with one exciton. Stimulated processes are assumed to occur from level 2 with a probability M , and radiative recombination will occur in a time τ_r . Excitons can also escape out of the dot by field-dependent carrier tunnelling, with a escape time $\tau_{\text{esc}}(F)$. This leads to occupation-dependent absorption behaviour in the QD, which produces a nonlinear power dependence of the PC resulting in saturation. At higher electric fields the tunnelling time is reduced and the QD returns faster to its initial empty state; therefore, a higher absorption rate is possible, leading to a

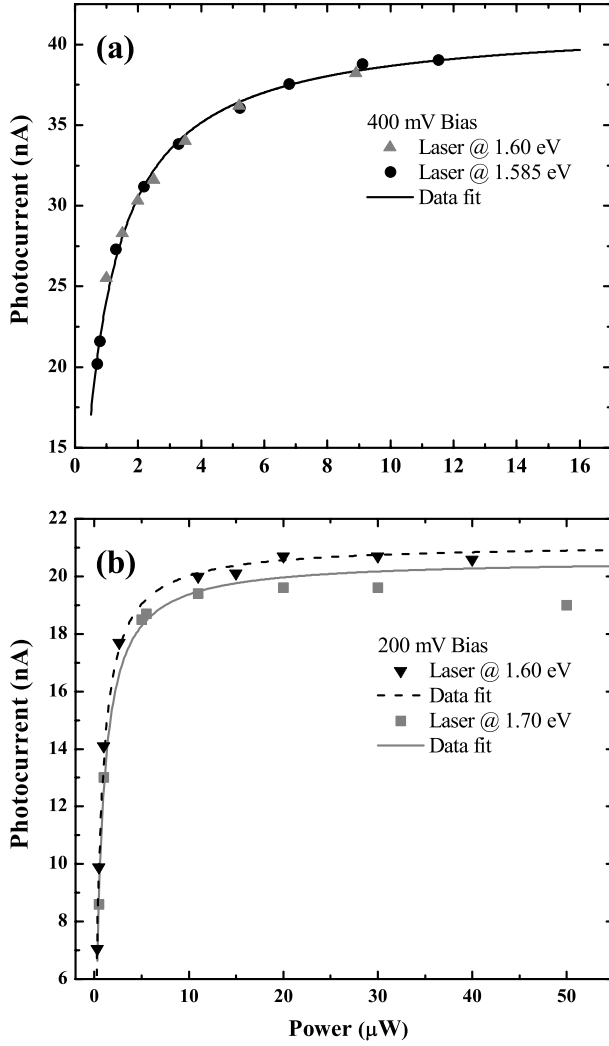


Figure 4. Photocurrent as a function of laser power, for different laser energies at 10 K; (a) 400 mV bias across the groove, (b) 200 mV bias, for higher laser powers. The data have been fitted with a function based on a rate equation model described in the text.

higher saturated PC. With an excitation power P , the steady-state photocurrent is found to be

$$I_{PC} = \left(\frac{e}{2\tau_{esc}} \right) \frac{P}{P + (1/2M)(1/\tau_{esc} + 1/\tau_r)}. \quad (1)$$

This expression has been used to fit the data points in figure 4; we assume $\tau_r \equiv \tau_{loc} \simeq 350$ ps [15]. From the data in figure 4(a), with resonant excitation of the QWR at 1.585 eV, we obtain values of $M = 0.36$ and an escape time $\tau_{esc} = 1.9$ ps for 400 mV bias (80 V cm^{-1}). At 200 mV bias, the data fits in figure 4(b) give $M = 0.24$ (0.20) and $\tau_{esc} = 3.8$ (3.9) ps for illumination at 1.6 eV (1.7 eV).

The closeness of the fits, at least over low powers, suggests that the model developed for QDs is still valid in our 1D QWR system. As discussed above, this is likely to be a result of localization effects at low temperatures; monolayer growth fluctuation will generate a series of quasi-0D boxes along the wire axis [26], thus emulating the isolated QD model. The escape time at 400 mV bias is half the value at 200 mV, which suggests that in our case τ_{esc}^{-1} gives an indication of the rate of

dissociation of localized excitons by field-accelerated carriers. These values of τ_{esc} are much smaller than those obtained by Beham *et al* even though our fields are 1000 times less; this is likely to be due to the relatively high density of ‘0D’ localization sites along the wire ($\sim 5 \times 10^4 \text{ cm}^{-1}$ [15]) and the presence of a 1D delocalized exciton continuum.

Figure 4(b) shows that the saturation PC is significantly smaller when exciting at 1.70 eV rather than quasi-resonantly with the $n = 1$ state at 1.6 eV. This is consistent with the discussion above; at low temperatures, excitons generated in excited states of the wire recombine more rapidly than ground state excitons, thus generating fewer dissociated free carriers.

There are not sufficient data to comment on the field and excitation energy dependence of M , and a detailed analysis of the true meaning of M and τ_{esc} in a 1D system will require a more comprehensive investigation. The deviation of the PC from a saturation value, observed at higher powers in figure 4(b), may be due to the presence of the 1D continuum. Scattering with phonons and other carriers may be significant at higher powers and has also been neglected in the model.

4. Conclusion

In summary, we have presented longitudinal photocurrent measurements of a single v-groove quantum wire. The photocurrent response from certain structures within the v-groove is found to be highly temperature dependent. In particular, a marked drop is observed in the contribution from the first excited state of the quantum wire, below 30 K. This has been attributed to frustrated carrier relaxation resulting from localization effects.

The total photocurrent through the v-groove has been measured as a function of incident laser power; the PC response can be closely fitted by a rate equation model for a two-level quantum dot system. This is consistent with the scenario that the QWR behaves as a series of quasi-0D boxes, at low temperatures.

Acknowledgments

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References

- [1] Thornton T J, Pepper M, Ahmed H, Andrews D and Davies G J 1986 *Phys. Rev. Lett.* **56** 1198
- [2] Pfeiffer L, West K W, Störmer H L, Eisenstein J P, Baldwin K W, Gershoni D and Spector J 1990 *Appl. Phys. Lett.* **56** 1697
- [3] Colas E, Simhony S, Kapon E, Bhat R, Hwang D M and Lin P S D 1990 *Appl. Phys. Lett.* **57** 914
- [4] Tans S J, Devoret M H, Dai H, Thess A, Smalley R E, Geerligs L J and Dekker C 1997 *Nature* **386** 474
- [5] Fink H-W and Schönenberger C 1999 *Nature* **398** 407
- [6] Sirigu L, Oberli D Y, Degiorgi L, Rudra A and Kapon E 2000 *Phys. Rev. B* **61** 10575
- [7] Ferry D K and Goodnick S M 1997 *Transport in Nanostructures* (Cambridge: Cambridge University Press)
- [8] Hamoudi A, Ogura M and Wang X L 1997 *J. Appl. Phys.* **81** 6229

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- [9] Fu Y, Willander M, Liu X Q, Lu W, Shen S C, Tan H H, Jagadish C, Zou J and Cockayne D J H 2001 *J. Appl. Phys.* **89** 2351
 - [10] Cingolani R, Rinaldi R, De Vittorio M, Vasanelli L, Cola A, Marti U and Reinhart F K 1996 *J. Appl. Phys.* **80** 936
 - [11] Saraydarov M, Donchev V, Germanova K, Wang X L, Kim S J and Ogura M 2004 *J. Appl. Phys.* **95** 64
 - [12] Lelarge F, Kaufman D, Dwir B, Mautino S, Rudra A and Kapon E 2000 *J. Cryst. Growth* **221** 540
 - [13] Schäpers Th, Hartmann A, Schwarz A, Hardtdegen H, Bongartz M, Dieker Ch and Lüth H 1998 *Appl. Surf. Sci.* **123/124** 687
 - [14] Biasiol G, Reinhardt F, Gustafsson A, Martinet E and Kapon E 1996 *Appl. Phys. Lett.* **69** 2710
 - [15] Cade N I, Roshan R, Hauert M, Maciel A C, Ryan J F, Schwarz A, Schäpers Th and Lüth H 2004 *Phys. Rev. B* **70** 195308
 - [16] Hauert M, Roshan R, Maciel A C, Kim J, Ryan J F, Schwarz A, Kaluza A, Schäpers Th and Lüth H 2001 *Proc. 25th Int. Conf. on the Physics of Semiconductors (Osaka)* ed N Miura and T Ando (Berlin: Springer) p 703
 - [17] Cade N I, Roshan R, Ryan J F, Maciel A C, Schwarz A, Schäpers Th and Lüth H 2002 *Physica B* **314** 413
 - [18] Kiener C, Rota L, Maciel A C, Freyland J M and Ryan J F 1996 *Appl. Phys. Lett.* **68** 2061
 - [19] Wang X L, Ogura M, Matsuhata H and Hamoudi A 1997 *Appl. Phys. Lett.* **71** 2130
 - [20] Bellessa J, Voliotis V, Grousson R, Wang X L, Ogura M and Matsuhata H 1997 *Appl. Phys. Lett.* **71** 2481
 - [21] Bellessa J, Voliotis V, Grousson R, Wang X L, Ogura M and Matsuhata H 1998 *Phys. Rev. B* **58** 9933
 - [22] Oberli D Y, Vouilloz F, Ambigapathy R, Deveaud B and Kapon E 2000 *Phys. Status Solidi a* **178** 211
 - [23] Feldmann J, Peter G, Göbel E O, Dawson P, Moore K, Foxon C and Elliott R J 1987 *Phys. Rev. Lett.* **59** 2337
 - [24] Citrin D S 1992 *Phys. Rev. Lett.* **69** 3393
 - [25] Akiyama H, Koshiba S, Someya T, Wada K, Noge H, Nakamura Y, Inoshita T, Shimizu A and Sakaki H 1994 *Phys. Rev. Lett.* **72** 924
 - [26] Beham E, Zrenner A, Findeis F, Bichler M and Abstreiter G 2002 *Physica E* **13** 139
 - [27] Hasen J, Pfeiffer L N, Pinezuk A, He S, West K W and Dennis B S 1997 *Nature* **390** 54