Incommensurate structures and radiation damage in Rb2V3O8 and K2V3O8 mixed-valence vanadate fresnoites

**Andrzej Grzechnikab\*, Vaclav Petricekc, Dmitry Chernyshovd, Charles McMonagled, Tobias Geiseb, Hend Shahedeb and Karen Frieseeb**

aJülich Centre for Neutron Science (JCNS-4), Forschungszentrum Jülich GmbH, Jülich, 52425, Germany
bInstitute of Crystallography, RWTH Aachen University, Aachen, 52066, Germany
cInstitute of Physics, Czech Academy of Sciences, Prague 8, 18221, Czech Republic
dSwiss-Norwegian Beamlines, European Synchrotron Radiation Facility, Grenoble, 38043, France
eJülich Centre for Neutron Science (JCNS-2), Forschungszentrum Jülich GmbH, Jülich, 52425, Germany

 Correspondence email: a.grzechnik@fz-juelich.de

**Funding information**      Bundesministerium für Bildung und Forschung (grant No. 05K19PA2 and the grant for the Palestinian German Science Bridge); Deutsche Forschungsgemeinschaft (grant No. SFB 917).

1. The structures and phase transitions to the incommensurate structures in Rb2V3O8 and K2V3O8 mixed-valence vanadate fresnoites are studied with the synchrotron single-crystal diffraction as a function of radiation dose at low temperatures and ambient pressure. Strategies to collect and analyse single-crystal diffraction data measured with very intense synchrotron radiation using modern low-noise pixel area detectors are discussed.
2. The structures and phase transitions to incommensurate structures in Rb2V3O8 and K2V3O8 mixed-valence vanadate fresnoites are studied with synchrotron single-crystal diffraction at low temperatures and ambient pressure. Although mixed satellite reflections are absent, the modulated structure of K2V3O8 below 115 K is better described in (3+2) than in (3+1) dimensional space. The geometries of the VO4 and VO5 building units are rigid and it is mainly slight rotations of these polyhedra and small variation of the intermediate K-O distances that are modulated. The prolonged exposure to the high-brilliance synchrotron beam suppresses the incommensurate phase. The previously postulated phase transition to the incommensurate phase in Rb2V3O8 at 270 K is not observed in our data. One of the reasons could be that the intense radiation also affects the modulation in this material. Strategies to collect and analyse single-crystal diffraction data measured with very intense synchrotron radiation using modern low-noise pixel area detectors are discussed.
3. Synchrotron radiation; single-crystal diffraction; incommensurate structures; radiation damage.

Introduction

Mixed-valence A2V3O8 vanadate fresnoites have a structure (*P*4*bm*, *Z* = 2) built up of layers with corner-sharing V5+O4 tetrahedra and V4+O5 square pyramids separated by the A+ cations, where A+ is K+, Rb+, Cs+, or (NH4)+ (Galy and Carpy, 1975; Theobald *et al*., 1984; Ha-Eierdanz and Müller, 1992; Yeon *et al*., 2013) (Figure 1). Their magnetic and magneto-optical properties have been extensively investigated in the past (Liu and Greedan, 1995; Choi *et al*., 2001; Sales *et al*., 2002; Rai *et al*. 2006; Lumsden *et al*., 2006; Choi *et al*., 2012; Yeon *et al*., 2013). K2V3O8 is a potential material for photocatalysis (Basu *et al*., 2022) and batteries (Lu *et al*., 2018; Jo *et al*., 2019; Yang *et al*., 2019; Li *et al*., 2020; Li *et al*., 2022),

The mineral fresnoite Ba2TiSi2O8 has an incommensurate (3+2)-*d* structure with the modulation vectors **q**1 = 0.3020(3)(***a***\*+***b***\*) and **q**2 = 0.3020(3)(-***a***\*+***b***\*) and superspace group *P*4*bm*(, , ½)(-, , ½)*0gg* (Bindi *et al*., 2006). The displacive modulations of the Ba2+ and O2- ions lead to a variation in the coordination (eight-, nine- and tenfold) of the Ba2+ ions. The modulation of the TiO5 square pyramids and SiO4 tetrahedra is mainly rotational, where the former units stay essentially rigid while the pyrosilicate group also shows a significant change of geometry. On the other hand, the closely related compound Ba2TiGe2O8 has a (3+1)-*d* modulated structure with superspace group *Cmm*2(0, , ½)s00, with a β component of the **q**-vector of ~ 0.635 (Hoche *et al*., 2003; Höche, 2005). The modulation can mainly be attributed to rotations of the TiO5 square pyramids and one of the pyrogermanate groups, while the second Ge2O7 unit is in addition deformed. This again leads to a variation of the coordination number of the barium ions between eightfold and tenfold. Based on these observations, Höche (2005) indicated that “underbonding” of the large cations (Ba2+ and Sr2+) is responsible for the displacive modulations in silicate and germanate fresnoites. In the average structures of the vanadium fresnoites K2V3O8 and Rb2V3O8, both K+ and Rb+ cations are “overbonded” (Höche, 2005).

Vibrational studies by Choi *et al*. (2001, 2012) revealed lattice instabilities in K2V3O8 at about 110 K and 60 K. The former was argued to be due to a local distortion of the V4+O5 square pyramids. Later it was shown that K2V3O8 undergoes a structural phase transition at 115 K (Chakoumakos *et al*., 2007; Takeda *et al*., 2019). While there is an agreement in the literature on the fact that a modulated structure is formed below this temperature, there are discrepancies with respect to the modulation vector. Below 115 K, weak incommensurate satellite reflections were observed in electron diffraction patterns with the modulation vector **q**1 ~ 0.12***c***\* (Höche, 2005). However, laboratory single-crystal X-ray data showed only weak satellites in the <110> projections at 90 K. These satellites were not analysed in detail and Höche (2005) solely presented a refinement of the average structure (*P*4*bm*, Z = 2) at 90 K without taking satellite reflections into consideration in the analysis. The results of neutron powder diffraction and laboratory single-crystal X-ray diffraction on K2V3O8 suggested that weak superlattice reflections can be indexed with the commensurate wave vector ± ⅓<110>\*+½***c***\* below 115 K (Chakoumakos *et al*., 2007). The proposed possible space groups for the average structure are *P*42*bc* or *P*4*nc*. Based on single-crystal *time-of-flight* Laue neutron diffraction data, Wang and Petricek (2019) showed that K2V3O8 is incommensurately modulated and that the satellite reflections can be indexed with **q** vectors of ± (⅓+δ)<110>\*+½***c***\*. They also discussed refinements in (3+2)-*d* tetragonal and (3+1)-*d* orthorhombic superspace groups.

Rb2V3O8, which is isostructural to K2V3O8 at ambient conditions, was found to undergo a phase transition to a modulated (3+1)-*d* structure at 270 K. Based on electron diffraction data, the wave vector was determined as **q**1 ~ 0.16***c***\* and the superspace group was assumed to be *P*4*bm*(00) (Withers *et al*., 2004; Höche, 2005). The modulation was supposed to arise from rotations and shifts of rigid V5+O4 and V4+O5 polyhedra. No incommensurate satellite reflections were detected with laboratory single-crystal X-ray diffraction in Rb2V3O8 (Withers *et al*., 2004).

In this work, we are interested in the evolution of the average structures and the phase transitions to the incommensurate structures in Rb2V3O8 and K2V3O8 based on the synchrotron single-crystal diffraction at low temperatures and ambient pressure. Using the recently implemented cyclic refinement option in the program Jana2020, which is a modified version of Jana2006 (Petricek *et al*., 2014), we follow the temperature dependence of various lattice and structural parameters of the average structures in the vicinity of the reported phase transitions at 270 K in Rb2V3O8 and at 115 K in K2V3O8 and examine details of the incommensurate phases. We also investigate the effect of the high-brilliance synchrotron beam on the crystal structures of both compounds.

**Experimental**

The synthesis of Rb2V3O8 and K2V3O8 crystals was described by Lumsden *et al*. (2006). The preselection of the grown crystals for further synchrotron measurements were performed on a laboratory X-ray diffractometers IPDS-II (STOE) and Supernova (Rigaku Oxford Diffraction), both with Mo-K. Each of the crystals in our study was about 50x40x20 m3 in size. The IPDS-II data analysis with the program X-Area (Stoe & Cie, 2011) and the refinements with the program Jana2020 (a modified version of Jana2006; Petricek *et al*., 2014) showed that the structures of Rb2V3O8 and K2V3O8 can be described in 3-dimensional space(*P*4*bm*, Z = 2) at ambient temperature. The crystals contain only one enantiomorph, i.e., the refined volume fractions of the two individuals assuming inversion twinning were 1 and 0 within estimated standard deviations.

Synchrotron single-crystal diffraction measurements ( = 0.69127 Å) in the ranges 108-350 K for Rb2V3O8 and 88 – 298 K for K2V3O8 with the temperature step of 2 K at ambient pressure were performed on the BM01 station of Swiss-Norwegian Beamlines (SNBL) at the European Synchrotron Radiation Facility (Grenoble, France) (Dyadkin *et al*., 2016). The crystals were mounted on glass pins and placed in the stream of nitrogen from Oxford Cryostream 700+. The data (a full rotation of 360°) were collected using the Pilatus 2M detector. After several test exposures, our data collection strategy was to measure frames with an angular step of 1° and the exposure time was 0.1 sec per frame. Such a combination of an angular slicing and exposure time is a compromise that (i) keeps the detector in a linear response range even for strong reflections, (ii) uses rotation speed (10°/sec) that provides precise angular positioning, (iii) reduces number of files to a reasonable and manageable number. Finer angular slicing is always better, but to stay with the same dose accumulation rate (same exposure per 0.1°) we would have to use the 0.1°/0.01s strategy, that is still possible at BM01. Unfortunately, it implies 10 time more data that is not very practical if the goal is to collect many datasets for mapping temperature or time evolution of the crystal structures. A discussion of data collection with pixel photon counting detectors at third-generation synchrotron sources was presented by Krause *et al*. (2020).

 With an angular step of 1° and the exposure time was 0.1 sec per frame. One data collection, i.e., one scan, took 36 seconds. The relatively high starting temperature (350 K) for the measurements on Rb2V3O8 was chosen so that several data points could be collected above the postulated phase transition at 270 K (Withers et al., 2004). Using the same scan parameters, additional measurements to detect weak satellite reflections and to follow the evolution of the modulated structures as a function of the scan number (i.e., the exposure time or radiation dose) were performed at 100 K ( = 0.69127 Å) and 250 K ( = 0.60523 Å) for Rb2V3O8 on two different crystals with 20 and 40 scans, respectively. Similar measurements on one crystal of K2V3O8 at 100 K and 150 K were repeated 20 times ( = 0.69127 Å). Additionally, they were repeated 360 times ( = 0.60523 Å) on a second crystal of K2V3O8 at 100 K. The dose calculations were done with RADDOSE 3D software (Bury *et al*., 2018). The average radiation dose per scan on Rb2V3O8 was 200 Gy while it was 160 Gy for K2V3O8. The corresponding frames from the separate scans were subsequently binned using the SNBL ToolBox software (Dyadkin *et al*., 2016). All the synchrotron data were analysed and processed with the program Crysalis (Rigaku Oxford Diffraction, 2021). Solution and refinement of the structures were carried out with the program Jana2020 (Petricek *et al*., 2014). For this, we made use of the newly developed option of cyclic refinements for single crystal data, which was recently incorporated into Jana2020 (Petricek *et al*., 2014).

**Results and Discussion**

A series of measurements on K2V3O8 was performed with single scans in the temperature range 88-298 K. Afterwards, the crystal was heated to 100 K and then to 150 K and 20 consecutive scans were collected at both temperatures. Analysis of reciprocal space in all the data shows that first-order satellite reflections can only be observed when the frames from separate 20 scans at 100 K are binned into one single data set (Figure 2).

*Temperature dependence of the average structure of K2V3O8*

Considering solely the main reflections above and below the phase transition at 115 K, there is no indication for a change of space group symmetry *P*4*bm* of the average structure. Consequently, all the data collected with one scan between 88 K and 298 K were processed in space group symmetry *P*4*bm* using the cyclic refinement option in Jana2020 (Petricek *et al*., 2014). The starting data set for the cyclic refinement was the one measured at 88 K. It was refined first and, once a satisfactory agreement was achieved, the resulting structural parameters were then automatically passed on to the next higher temperature by the program. The next temperature point was then refined and this was subsequently repeated until the dataset at the highest temperature was reached. The refined parameters at all temperatures included anisotropic displacement parameters and an isotropic extinction correction, Giso. This new option in Jana2020 also allows to directly plot the obtained structural parameters as a function of the corresponding variable, in this case the temperature. Since a trial refinement of the inversion twinning showed that the twin volume fractions for the enantiomorphs were 1 and 0 within standard deviation, no volume fractions were refined. The origin along the *c* direction was fixed to the position of the K atom (*z* = 0.5). Lowering symmetry to the non-isomorphic subgroups *P*4, *Cmm*2, and *Pba*2 of *P*4*bm* and introducing additional twinning due to the loss of rotational symmetry elements in the refinements of the data below 115 K did not improve the result for the average structure, despite the increased number of free parameters.

The temperature dependence of the normalized lattice parameters and unit-cell volumes of the average structure, *P*4*bm*, in K2V3O8 is shown in Figure 3. Down to about 115 K, the *a* lattice parameter is essentially constant and the volume contraction is entirely due to the change in the *c* lattice parameter in K2V3O8. At lower temperatures, the *a* lattice parameter contracts while the temperature dependence of the *c* lattice parameter changes its slope. The volume contraction is altogether linear in the entire temperature range 88 - 298 K, V(T) = 412.77(1) + 0.01902(5)\*T. The anomalies in the evolution of the *a* and *c* lattice parameters indicate the phase transition at 115 K in K2V3O8 reported earlier (Choi *et al*., 2001, 2012; Höche, 2005; Chakoumakos *et al*., 2007; Takeda *et al*., 2019). This phase transition is also reflected in the slope changes of the isotropic displacement parameters for all the atoms in the average structure (Figure 4).

It is striking that there are no obvious anomalies in various geometrical parameters such as interatomic distances and angles at the phase transition at 115 K in K2V3O8 (Figures 5-7, additional Figures S1-S4 are in the supporting information). The unit-cell contraction along the *c* axis in the entire temperature range studied here is mainly due to the shortening of the K-O distances between the V3O8 slabs (Figure 5). This observation is supported by slightly increasing volumes of the VO4 tetrahedra and VO5 square pyramids upon cooling calculated using the program ToposPro (Blatov *et al*., 2014). The distances V1-O1 and O1-O3 (the O1 and O3 atoms being at the apex and the base of the VO5 polyhedron, respectively) increase upon cooling indicating that the square pyramid becomes elongated (Figure 6). On the other hand, the V-O and O-O distortions of the VO4 tetrahedra (as defined by Baur, 1974) decrease. The O3-O4-O3 and O3-O3-O3 angles between two tetrahedra and between the tetrahedra and square pyramids, respectively, are essentially constant (Figure 7).

*Incommensurate structure of K2V3O8*

No satellite reflections are seen in reciprocal space sections of our data with 20 binned scans recorded at 150 K. The very weak first-order satellite reflections, which appear in the *hk0.5* section of reciprocal space at 100 K, can be described with two vectors **q**1 = (, , 0.5) and **q**2 (-, , 0.5), where  ~ 0.3 (Figure 2). No higher order satellites or satellite reflections due to combination of the **q**1 and **q**2 vectors are visible in the corresponding sections of reciprocal space perpendicular to [001]. We also do not detect any satellites with the modulation vector **q** ~ 0.12***c***\* reported earlier by Höche (2005). These observations allow two different interpretations of the structure of K2V3O8 below 115 K: (1) the structure is (3+2)-*d* or (2) the structure is (3+1)-*d* and the observed diffraction pattern can be explained on the basis of additional twinning due to a 90° rotation around ***c***\*. The fact that in the latter case all satellites can be indexed with one single vector **q**1 = (, , 0.5) in combination with twinning would also imply that the overall symmetry is no longer tetragonal.

We first performed a refinement in (3+2)-*d* space*.* Starting from the space group *P*4*bm* of the average structure and the two **q** vectors, **q**1 = (0.313, 0.313, ½) and **q**2= (-0.313, 0.313, ½), all different translational parts for *x4* and *x5* of two superspace group symmetry operators were considered and tested (altogether 16 possibilities). The combination leading to superspace group *P*4*bm*(*αα*1/2)(*-αα*1/2)0*gg* led to the best refinement result. Two modulation waves both for the positions and anisotropic displacement parameters of all the atoms were considered [[1]](#footnote-1). The refinement converged to about *R*(all) = 2.07%, 1.31%, and 7.17% for all reflections, main reflections, and first-order satellites, respectively (Table 1). The superspace group *P*4*bm*(*αα*1/2)(*-αα*1/2)0*gg* is analogous to the one observed in the natural fresnoite Ba2TiSi2O8 (Bindi *et al*., 2006), but with different **q** vectors.

The second model in (3+1)-*d* space implies that the modulation happens along one direction only. Therefore, orthorhombic symmetry is more likely for the (3+1)-*d* model. It was derived from the (3+2)-*d* one by reducing symmetry to orthorhombic (*C*-centered cell with *a*ortho = *a*tetra - *b*tetra, *b*ortho = *a*tetra + *b*tetra, cortho = ctetra) and changing the number of modulation waves from two to one. The resulting superspace group for the (3+1)-*d* model is then *Cmm*2(0*β*1/2)*s*00 with one modulation vector **q**1 = (0, 0.626, ½). Two twin domains originating from the loss of the 4-fold axis were introduced. The twin volume fractions refined to 0.469(2):0.531(2). Final refinement values for the 4-dimensional model are *R*(all) = 2.40%, 1.39% and 8.48% for all, main and satellite reflections (Table S1 in the supporting information) 1. The superspace group and **q** vector of this model correspond to those of Ba2TiGe2O8 fresnoite (Hoche *et al*., 2003; Höche, 2005).

As the detection of mixed satellites would allow to uniquely identify the (3+2)-*d* model as the correct one, we collected a larger number of scans (360) on a second crystal and then binned them together (the crystal was in the beam for one scan at room temperature to check its quality). Even in these data, no mixed satellites could be observed in K2V3O8. Therefore, an unambiguous decision about the correct model for the modulated structure of K2V3O8 is not straightforward. As agreement factors for the (3+2)-*d* model are slightly better than for the (3+1)-*d* one and as the number of parameters in the refinement is significantly lower (96 parameters as compared to 144, respectively), the (3+2)-*d* model is more likely. Also, a comparison of the standard deviations of the atomic coordinates and atomic displacement parameters shows, that the errors on the parameters are on average 3-4 times smaller in the (3+2)-*d* model then in the (3+1)-*d* model.

The diffraction image simulations based on the refined (3+2)-*d* model using harmonic waves in principle do generate extremely weak mixed satellite reflections. It explains why these satellites could not be detected in our data, in which the first-order satellites are already very weak. If the modulations were stronger, the satellite formation would indeed exist. This, of course, is not a direct evidence that the structure is (3+2)-*d*, but it confirms that such a model does not contradict what was observed in this study. Consequently, the following discussion will mainly focus on the tetragonal (3+2)-*d* model (Table S1 and Figures S5-S7 corresponding to the (3+1)-*d* model are provided in the supporting information).

In both models of the modulated structure of K2V3O8 at 100 K the geometries of bothVO4 and VO5 building units are basically rigid. The V-O distances and O-V-O angles in both polyhedra are constant within their errors as a function of the internal parameter *t* and *u*. In addition, K-O distances below 3 Å are also only very weakly affected by the modulation and it is mainly the intermediate K-O distances (3.0-3.75 Å) that show a variation with the internal parameter(s) (Figure 8, see also Figure S5 in the supporting information). In contrast to the earlier investigated modulated structures of fresnoites, where the bond valence sums (BVS) of the A-cations are significantly affected by the modulation, the bond valence sum of the K atom in K2V3O8 is hardly affected by the varying distances and oscillates in the range BVSK = 0.947(1) - 0.949 (1) v.u. throughout the modulated structure.

The main influence of the modulation is on the interpolyhedral angles of the tetrahedra of the pyrovanadate group, O3-O3-O3, and on the angles between the tetrahedra and the square pyramids, O3-O4-O3 (Figures 9 and 10, see also Figures S6 and S7 in the supporting information). The variation of these angles leads to a slightly varying shape of the five membered rings surrounding the K-atoms.

*The influence of the radiation dose on the stability of the incommensurate structure of K2V3O8*

A close inspection of the whole series of 360 scans, which were measured on the second crystal, reveals that the first-order satellites were only visible when the initial 40 to 50 scans at 100 K were binned together. Binning frames corresponding to higher scan numbers did not show the presence of satellites and did therefore not contain any information about the modulation. Binning the first 40-50 scans in consecutive groups of 10 or 20 showed that the magnitude  of the vectors **q**1 = (, , 0.5) and **q**2 = (-, , 0.5) in the (3+2)-*d* model or of the vector **q** = (0, , 0.5) in the (3+1)-*d* model are constant within the estimated standard deviations, i.e.,  ≈ 0.313(2) or  ≈ 0.626(2). The satellites just became weaker and subsequently completely disappeared for higher scan numbers or, in other words, for increased exposure time or radiation dose. To elucidate the effect of the radiation dose on the average structure of K2V3O8 below 100 K, we refined all the 360 scans with the cyclic option in Jana2020 (Petricek *et al*., 2014). It is worth noting that the *a* lattice parameter is constant up to about the scan No. 50, above which it decreases (Figure 11). This coincides with the disappearance of the satellites. On the other hand, the *c* lattice parameter increases, while the unit-cell volume is constant. The response of the crystal structure of K2V3O8 to the high-brilliance synchrotron beam is also visible in the increase of the isotropic displacement parameters for all the atoms (Figure 12). Both agreement factors Robs and Rall decrease with the radiation dose (Figure 13).

*Temperature and radiation dose dependence of the structure of Rb2V3O8*

A similar approach as the one described above for K2V3O8 was followed to determine the temperature-dependent behaviour of Rb2V3O8. The data with one scan were recorded on cooling in the temperature range 108-350 K and then refined in a cyclic way using the program Jana2020 (Petricek *et al*., 2014). Based on the analysis of the reciprocal space at all temperatures, there is no evidence for a change of space group symmetry *P*4*bm* of the average structure. It is striking that the anomalies in the *a* and *c* lattice parameters occur in Rb2V3O8 at about 180 K (Figure 3). The bulk thermal contraction also changes its slope at this temperature indicating higher thermal contraction below 180 K. The volume contraction above and below 180 K is V(T) = 438.32(1) + 0.0197(1)\*T and V(T) = 437.61(2) + 0.0238(2)\*T, respectively. We do not see any anomaly in the temperature evolution of the lattice parameters, unit-cell volumes, and geometrical parameters at 270 K, which would correspond to the occurrence of the previously reported modulated structure (Withers *et al*., 2004; Höche, 2005). On the other hand, the isotropic displacement parameters become constant or increase below about 180 K (Figure 14). This would possibly indicate the presence of a transition to a phase with a different symmetry than *P*4*bm*. However, neither magnetic measurements by Liu and Greedan (1995) nor thermal analysis by Withers *et al*. (2004) showed any evidence for a phase transition at this temperature.

To check whether there is indeed a new phase of Rb2V3O8 below 180 K, we collected 20 consecutive scans at 100 K on the same crystal at the end of the series of measurements in the range 108-350 K. It turned out that in the binned data set at this temperature there are no additional weak reflections violating the extinction rules for space group *P*4*bm* of the average structure. We also do not see any weak features that could possibly be interpreted as satellites or superstructure reflections.

A cyclic refinement of the 20 scans collected on the first crystal of Rb2V3O8 at 100 K with the program Jana2020 (Petricek *et al*., 2014) as a function of the scan number (i.e., the exposure time or radiation dose) revealed that the *a* and *c* lattice slightly decrease and increase, respectively (Figure 15). In addition, the isotropic displacement parameters for all atoms monotonically increase with the constant unit-cell volume (Figure 16).

An additional data set with 40 scans was collected on a second crystal of Rb2V3O8 at 250 K, i.e., below the phase transition to the incommensurate phase reported by Withers *et al*. (2004), without any prior long exposure to the synchrotron radiation. The crystal was in the beam for one scan at room temperature to check its quality. No satellites are seen in the combined data sets from either first 20 or all 40 binned scans at 250 K. Both the *a* lattice parameter and unit-cell volume remain essentially constant while the *c* lattice parameter slightly increases as a function of the scan number (Figure 15). The response of the crystal structure to the prolonged synchrotron beam exposure is also reflected in the increase of the isotropic displacement parameters for all the cations (Figure 16). These trends at both 100 K and 250 K are in principle analogous to those observed upon cooling below 180 K and to those in K2V3O8 at 100 K upon long exposure to the synchrotron radiation beam. Unlike in K2V3O8 at 100 K (Figure 13), the agreement factors Robs and Rall for the data on Rb2V3O8 with limited number of scans at 100 K and 250 K (20 and 40, respectively) are essentially constant.

**Conclusions**

Our conclusions are:

1. In the temperature range, in which the commensurate fresnoite structures of K2V3O8 and Rb2V3O8 (*P*4*bm*, *Z* = 2) are stable, the *a* lattice parameter is constant and the thermal volume contraction is solely due to the change in the *c* lattice parameter. Such a behaviour is very similar to that of (NH4)2V3O8 and Cs2V3O8 upon compression at room temperature (Grzechnik *et al*., 2011; Grzechnik *et al*., 2016). Up to the pressure-induced phase transitions to the centrosymmetric structures *P*4/*mbm,* which are observed in these two vanadate fresnoites, the *a* lattice parameters are also nearly constant and it is mainly the contraction perpendicular to the stacking of the V3O8 slabs that accounts for the bulk compressibility.
2. Due to the absence of mixed satellite reflections it is not possible to unambiguously decide whether the modulated structure of K2V3O8 below 115 K is better described in (3+2) or (3+1) dimensional space. However, the slightly better agreement factors and the smaller number of parameters hint towards the (3+2)-*d* model as the correct choice. The geometries of the VO4 and VO5 building units stay constant as a function of the modulation and it is mainly slight rotations of these polyhedra (reflected in varying interpolyhedral O-O-O angles) that lead to a small variation of the intermediate K-O distances. In contrast to other fresnoites, the effect of the modulation on the bond valence sums of the slightly “underbonded” K+ are minimal.
3. Since no bulk thermal expansion is observed in K2V3O8 and Rb2V3O8 when they are in the high-brilliance synchrotron beam for an extended time, we interpret our observations on their behaviour to be caused from the exposure to radiation. Any effect from heating the material by the beam would otherwise lead to the bulk expansion of the lattice. Our results imply that the detection of weak satellites in incommensurate phases (e.g., the mixed satellites in K2V3O8 below 115 K) is difficult as the prolonged exposure to a high-brilliance synchrotron beam could lead to the disappearance of subtle modulations. The onset of the damage in Rb2V3O8 could already be seen at a temperature as high as 250 K. The postulated phase transition to the incommensurate phase, which is supposed to occur at 270 K (Withers *et al*., 2004), is not observed in our synchrotron data. The reason could be either that the intense radiation suppresses the modulation or that the electron beam, to which the sample was exposed in the study by Withers *et al*. (2004), shows an interaction with the structure that leads to the development of a modulated structure.
4. Radiation damage manifests as a structural disorder in the crystal, that suppresses scattering at high angles and can heat the crystal and/or modify the equilibrium between various polymorphic modifications. Notable examples in the field of small molecule crystallography were presented by Christensen *et al*. (2019), Lawrence Bright *et al*. (2021), Bogdanov *et al*. (2021), Chernyshov *et al*. (2022). However, none of these examples deals with modulated structures.
5. The newly developed cyclic refinement option in Jana2020 (Petricek *et al*., 2014) permits a convenient treatment of many subsequent data sets collected on one single crystal under similar conditions. We could show that this option is not only suitable for temperature dependent measurements but also for the study of the effects due to the radiation damage.
6. We finally conclude with a note on synchrotron data collection strategies suitable to measure coexisting strong and weak diffraction features. Bright synchrotron light and modern low-noise pixel area detectors make diffraction experiments very fast, but there is a limitation on the linear detector response of the dynamic range. One strategy is to collect data with different attenuations of the incoming beam and then merge the datasets, taking the strong intensities from the attenuated data and weak intensities from the un-attenuated measurement. Another strategy, the one applied here, is to perform the same data collection many times with an attenuated beam. These data can then be summed partially or totally into a single dataset with improved statistics. Such binning of the data is a technique perfectly suited to the zero-readout noise pixel area detectors, like Pilatus 2M used in our case. Individual datasets can be analysed separately to see the effect of the radiation dose as each dataset accounts for a specific dose accumulation proportional to the exposure time. This collection strategy gives greater flexibility for data analysis over more traditional strategies and allows the maximum data statistics to be extracted from each crystal.

**Figure 1** Crystal structure of the A2V3O8 vanadate fresnoites at room temperature.

**Figure 2** Reciprocal space sections *hk0* (left) and *hk0.5* (right) at 100 K.

**Figure 3** Normalized lattice parameters and unit-cell volumes to the values at 298 K as a function of temperature. The open and solid symbols are for Rb2V3O8 and K2V3O8, respectively.

**Figure 4** Ueq atomic displacement parameters as a function of temperature in K2V3O8. The estimated standard deviations are smaller than the size of the symbols.

**Figure 5** K-O interatomic distances as a function of temperature in K2V3O8. The standard deviations are drawn when larger than the size of the symbols.

**Figure 6** O-O interatomic distances as a function of temperature in K2V3O8.

**Figure 7** Interpolyhedral angles as a function of temperature in K2V3O8.

**Figure 8** K-O distances below 4.25 Å in the (3+2)-*d* structure of K2V3O8 at 100 K as a function of the internal coordinate *t* (*u* = 0.0) (*top*) and *u* (*t* = 0.0) (*bottom*).

**Figure 9** O3-O3-O3 interpolyhedral angles in the (3+2)-*d* model as a function of the internal coordinate *t* for different values of the internal coordinate *u*.

**Figure 10** O3-O4-O3 interpolyhedral angles in the (3+2)-*d* model as a function of the internal coordinate *u*. The modulation of the angles does not depend on the internal coordinate *t*.

**Figure 11** Lattice parameters and unit-cell volumes as a function of the scan number in K2V3O8 at 100 K.

**Figure 12** Ueq atomic displacement parameters as a function of the scan number in K2V3O8 at 100 K. Estimated standard deviations for the oxygen atoms are not plotted for clarity.

**Figure 13** Agreement facrors Robs (full symbols) and Rall (open symbols) as a function of the scan number in K2V3O8 at 100 K.

**Figure 14** Ueq atomic displacement parameters as a function of temperature in Rb2V3O8. The estimated standard deviations are smaller than the size of the symbols.

**Figure 15** Lattice parameters and unit-cell volumes as a function of the scan number in Rb2V3O8 at 100 K (full symbols) and 250 K (open symbols). Estimated standard deviations are shown when larger than the size of the symbols.

**Figure 16** Ueq atomic displacement parameters as a function of the scan number in Rb2V3O8 at 100 K (full symbols) and 250 K (open symbols). Estimated standard deviations for the cations are shown when larger than the size of the symbols. Estimated standard deviations for the oxygen atoms are not plotted for clarity.

**Table 1** Experimental data for the measurement on the (3+2)-*d* structure of K2V3O8 at 100 K ( = 0.69127 Å).

----------------------------------------------------------------------------------

Superspace group *P*4*bm*(*αα*1/2)(*-αα*1/2)0*gg*

*a* (Å) 8.942(2)

*c* (Å) 5.224(2)

V (Å3) 417.6

*Z* 2

Dc (g cm-3) 2.855

Giso 0.208(8)

Modulation vector **q**1 0.313(2), 0.313(2), 0.499(1)

Modulation vector **q**2 -0.313(2), 0.313(2), 0.499(1)

*Data collection*

No. measured refl. 8245

No. main refl. 1463

No. satellite refl. ± (1,0) 3794

Range of *hkl* -10 < *h* < 10

-11 < *k* < 11

 -6 < *l* < 6

 -1 < *m* < 1

 -1 < *n* < 1

No. observed refl. (all, main, satellites)a 445, 793

Rint(obs/all) 1.68/1.70

*Refinement* b

Robs/all (all refl.) 1.56/2.07

wRobs/all (all refl.) 1.93/2.00

Robs/all (main refl.) 1.31/1.31

wRobs/all (main refl.) 1.69/1.69

Robs/all (satellite refl. ± (1,0)) 4.00/7.17

wRobs/all (satellite refl. ± (1,0)) 4.10/4.58

No. parameters 96

----------------------------------------------------------------------------------

a Criterion for observed reflections is |*F*obs| > 3

b All agreement factors are given in %, weighing scheme 1/[2(*F*obs) + (0.01 *F*obs)2]

1. Brian C. Sales provided the samples for this study. Sophia Kurig assisted us with the preparation of the diffraction measurements. We also thank Xiaoping Wang for discussions.

References

References

Basu, S.S., Rahut, S., Bisht, A.S. & Basu, J.K. (2022). *Mater. Sci. Semiconduct. Process.* **147**, 106681.

Baur, W.H. (1974). *Acta Cryst*. B**30**, 1195-1215.

Bindi, L., Dusek, M., Petricek, V. & Bonazzi, P. (2006). *Acta Cryst*. B**62**, 1031-1037.

Blatov, V.A., Shevchenko, A.P. & Proserpio, D.M. (2014). *Cryst. Growth Des*. **14**, 3576-3586.

Bogdanov, N.E., Zakharov, B.A., Chernyshov, D., Pattison, P. & Boldyreva, E.V. (2021). *Acta Cryst.* B**77**, 365-370.

Bury, C.S., Brooks-Bartlett, J.C., Walsh, S.P. & Garman, E.F. (2018). *Protein Science* **27**, 217-228.

Chakoumakos, B.C., Custelcean, R., Kamiyama, T., Oikawa, K., Sales, B.C. & Lumsden, M.D. (2007). *J. Solid State Chem*. **180**, 812-817.

Chernyshov, D., Dyadkin, V. & Törnroos, K.W. (2022). *Acta Cryst*. B**78**, 392-396.

Choi, J., Zhu, Z.T., Musfeldt, J.L., Ragghianti, G., Mandrus, D., Sales, B.C. & Thompson, J.R. (2001). *Phys. Rev*. B**65**, 054101.

Choi, K.-Y., Lemmens, P., Gnezdilov, V.P., Sales, B.C. & Lumsden, M.D. (2012). *Phys. Rev*. B**85**, 144434.

Christensen, J., Horton, P.N., Bury, C.S., Dickerson, J.L., Taberman, H., Garman, E.F. & Coles, S.J. (2019). *IUCrJ* **6**, 703-713.

Dyadkin, V., Pattison, Ph., Dmitriev, V. & Chernyshov, D. (2016). *J. Synchrotron Rad*. **23**, 825-829.

Galy, J. & Carpy, A. (1975). *Acta Cryst*. B**31**, 1794-1795.

Grzechnik, A., Ren, T.-Z., Posse, J.M. & Friese, K. (2011). *Dalton Trans*. **40**, 4572-4577.

Grzechnik, A., Yeon, J., Zur Loye, H.-C. & Friese, K. (2016). *J. Solid State Chem*. **238**, 252-258.

Ha-Eierdanz, M.-L. & Müller, U. (1992). *Z. Anorg. Allg. Chem*. **613**, 63-66.

Hoche, T., Esmaeilzadeh, S., Uecker, R., Lidin, S. & Neumann, W. (2003). *Acta Cryst*. **59**, 209-216.

Höche, T. (2005). Habilitation, Universität Leipzig, Germany.

Jo, J.H., Hwang, J.Y., Choi, J.U., Kim, H.J., Sun, Y.K. & Myung, S.T. (2019). *J. Power*

*Sources* **432**, 24-29.

Krause, L., Tolborg, K., Egede Grønbech, T.B., Sugimoto, K., Brummerstedt Iversen, B. & Overgaard, J. (2020). *J. Appl. Cryst*. **53**, 635-649.

Lawrence Bright, E., Giacobbe, C., Wright, J.P. (2021). *J. Synchrotron Rad*. **28**, 1377-1385.

Li, Z., Wu, B.K., Yan, M.Y., He, L., Xu, L., Zhang, G.B., Xiong, T.F., Luo, W. & Mai, L.Q. (2020). *ACS Appl. Mater. Interfaces* **12**, 10420-10427.

Li, J.S., Li, Q.N., Pang, H., Yu, Z.Z. & Yan, D.L. (2022). *Mater. Lett*. **320**, 132301.

Liu, G. & Greedan, J.E. (1995). *J. Solid State Chem*. **114**, 499-505.

Lu, M., Wang, K.F., Ke, H.D., Hu, Q., Liu, Z.H. & Wu, H.R. (2018). *Mater. Lett*. **232**, 224-227.

Lumsden, M.D., Nagler, S.E., Sales, B.C., Tennant, D.A., McMorrow, D.F., Lee, S.-H. & Park, S. (2006). *Phys. Rev*. B**74**, 214424.

Petricek, V., Dusek, M. & Palatinus, L. (2014). *Z. Kristallogr*. **229**, 345-352.

Rai, R.C., Cao, J., Musfeldt, J.L., Singh, D.J., Wei, X., Jin, R., Zhou, Z.X., Sales, B.C. & Mandrus, D. (2006). *Phys. Rev*. B**73**, 075112.

Rigaku Oxford Diffraction (2021). *CrysAlis PRO*. Rigaku Oxford Diffraction, Yarnton, England.

Sales, B.C., Lumsden, M.D., Nagler, S.E., Mandrus, D. & Jin, R. (2002). *Phys. Rev. Lett*. **88**, 095901.

Stoe & Cie (2011). *X-Area*. Stoe &Cie, Darmstadt, Germany.

Takeda, H., Yasuoka, H., Yoshida, M., Takigawa, M., Ghimire, N.J., Mandrus, D. & Sales, B.C. (2019). *Phys. Rev*. B**100**, 054406.

Theobald, F.R., Theobald, J.-G., Vedrine, J.C., Clad, R. & Renard, J. (1984). *J. Phys. Chem. Solids* **45**, 581-587.

Wang, X. & Petricek, V. (2019). *Workshop on Symmetry and Superspace Approach to Modulated Crystal Structures*, Oak Ridge, Tennessee, USA

Withers, R., Höche, T., Liu, Y., Esmaeilzadeh, S., Keding, R. & Sales, B.C. (2004). *J. Solid State Chem*. **177**, 3316-3323.

Yang, Y.S., Liu, Z.F., Deng, L.Q., Tan, L.L., Niu, X.G., Sanad, M.M.S., Zeng, L., Zhu, Z.P. & Zhu, Y.J. (2019). *Chem. Commun*. **55**, 14988-14991.

Yeon, J., Sefat, A.S., Tran, T.T., Halasyamani, P.S. & zur Loye, H.-C. (2013). *Inorg. Chem*. **52**, 6179-6186.

Supporting information

**Table S1** Experimental data for the measurement on the (3+1)-*d* structure of K2V3O8 at 100 K ( = 0.69127 Å).

-----------------------------------------------------------------------------

Superspace group *Cmm*2(0*β*1/2)*s*00

*a* (Å) 12.6452(9)

*b* (Å) 12.6452(9)

*c* (Å) 5.2235(5)

V (Å3) 835.2

*Z* 4

Dc (g cm-3) 2.855

Twin volume (1,2) 0.469(2), 0.531(2)

Giso 0.214(6)

Modulation vector **q** 0, 0.626, 0.5

*Data collection*

No. measured refl. 8351

No. main refl. 1575

No. satellite refl. ± (1,0) 6776

Range of *hkl* -16 < *h* < 15

-16 < *k* < 16

 -6 < *l* < 6

 -1 < *m* < 1

No. observed refl. (all, main, satellites)a 1929, 762, 1167

Rint(obs/all) 1.58/1.61

*Refinement* b

Robs/all (all refl.) 1.58/2.40

wRobs/all (all refl.) 1.98/2.09

Robs/all (main refl.) 1.34/1.39

wRobs/all (main refl.) 1.80/1.83

Robs/all (satellite refl. ± (1,0)) 4.09/8.48

wRobs/all (satellite refl. ± (1,0)) 4.49/5.31

No. parameters 144

-----------------------------------------------------

a Criterion for observed reflections is |*F*obs| > 3

b All agreement factors are given in %, weighing scheme 1/[2(*F*obs) + (0.01 *F*obs)2]



**Figure S1** Volume of the tetrahedra and square pyramids as a function of temperature in K2V3O8.



**Figure S2** V1-O interatomic distances in the VO5 square pyramid as a function of temperature in K2V3O8.



**Figure S3** V2-O interatomic distances in the VO4 tetrahedra as a function of temperature in K2V3O8.The standard deviations are drawn when larger than the size of the symbols.



**Figure S4** V2-O bond-length and O-O distance tetrahedral distortions in K2V3O8.



**Figure S5** K-O distances in the (3+1)-*d* structure of K2V3O8 at 100 K as a function of the internal coordinate *t*.



**Figure S6** O31-O31-O32 (black line) and O31-O32-O32 (red line) interpolyhedral angles in the (3+1)-*d* model as a function of the internal coordinate *t*.



**Figure S7** O31-O41-O31 (black line), O32-O42-O32 (red line), and O32-O42-O32 (blue line) interpolyhedral angles in the (3+1)-*d* model as a function of the internal coordinate *t*.

1. When the Fourier amplitudes of the modulation function of the displacement parameters were smaller than 3 times the standard deviation, the corresponding parameters were set to 0.0 in both models. This did not lead to a significant increase of the overall agreement factors. [↑](#footnote-ref-1)