## Insulator to metal transition by bandwidth reduction

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In a classic Mott system, the transition from the insulating to the metallic state occurs when the lattice spacing is reduced such that the increase in bandwidth W overcomes the local Coulomb repulsion U. In this picture, both W and the ratio W/U become larger with decreasing the distance between correlated sites. Here we show that, in transition-metal compounds, there is another path to metallization. It is associated with an expansion (instead of a compression) of the lattice and thus with a decrease in W. In this scenario W/U grows with increasing distance—the opposite of the classical Mott case. Such a surprising inversion can be obtained by decoupling the main factors controlling W and screening. This mechanism explains the electronic behavior of the  $t_{2p}^2$  triangular lattice series LiV $X_2$  (X = O, S, and Se) with increasing anionic size.

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*Introduction*. The insulator to metal transition (IMT) occurs when the bandwidth of correlated electrons, W, increases beyond a critical value, overcoming the local Coulomb repulsion U. In the classical Mott picture, W decreases with increasing lattice spacing, and the ratio W/U follows; the metallic state thus emerges by lattice compression [1]. This scenario is believed to apply to most correlated transitionmetal compounds (TMCs). In real materials, complexity arises because of multiorbital effects. The latter take place within the correlated-electrons space and have various causes: orbital degeneracy and its reduction, the Hund's rule coupling, the crystal-field splitting, the spin-orbit coupling, and the subtle interplay of all that [2-9]. This does not change the overall picture, however. The reason is that W and U are typically controlled via the same effects-e.g., the hybridization with the anionic p states—but W is more sensitive than screening to lattice changes. The metallic phase is thus expected to be stabilized by lattice compression.

Here we show that, surprisingly, there is an alternative path to the IMT, which goes via lattice expansion. This requires that the main factors controlling bandwidth and screening are decoupled. The picture is illustrated in Fig. 1 in a possible realization. In TMCs, lattice expansion eventually leads to a decrease of W, everything else staying the same. There are different ways of expanding the lattice, however. A possible path is the chemical substitution of the anions with larger ions in the same group  $(2p \rightarrow 4p)$ . This choice, in specific structures, can indeed disentangle bandwidth and screening. In this situation, while W itself decreases, screening becomes rapidly more effective. Under certain circumstances, one can then reach a paradoxical situation: by expanding the lattice, while W decreases, the ratio W/U increases, and the system turns metallic (inverse Mott transition).

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This mechanism explains the puzzling trends in the electronic properties of the  $t_{2g}^2$  LiV $X_2$  series (X = O, S, and Se). These materials have attracted attention because of the unusual formation of complex molecular clusters and valencebond solid (VBS) states at low temperature [10-25]. The

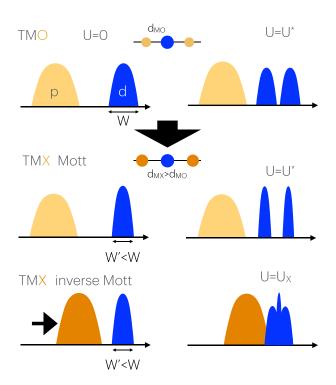


FIG. 1. Insulator to metal transition by lattice expansion. Top: Spectral functions of an insulating transition-metal oxide, for U=0(left) and for a U sufficiently large to open a gap (right). Correlated bands: blue. Anion bands: orange. Center: O is replaced by a larger anion X and the lattice spacing increases; the bandwidth W (left) is reduced. In a classic Mott picture, this enlarges the gap (right). Bottom: As in the central panel, but, in addition, the distance between anionic and correlated bands decreases (left), making screening more efficient. This eventually leads to metallization (right).

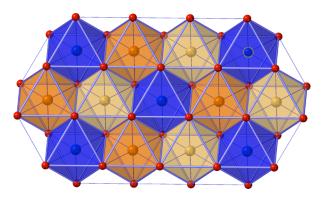


FIG. 2. Part of a VO<sub>6</sub> layer in LiVO<sub>2</sub>. Layers are made by edge-sharing VO<sub>6</sub> octahedra forming a triangular lattice (O: red spheres; V: remaining spheres); they are divided by triangular Li lattices, not shown.

crystal structure is made by layers of edge-sharing  $VX_6$  octahedra, separated by Li atoms forming themselves a triangular lattice. The V  $t_{2g}$  bands crossing the Fermi level are narrow and the systems are correlated. Experimentally, LiVO2 is a small gap insulator; from resistivity measurements the gap is 0.1–0.2 eV [26,27]. The system exhibits a phase transition to a VBS upon cooling, with critical temperature  $T_c \approx 500$  K. Instead, LiVS<sub>2</sub> and LiVSe<sub>2</sub> are metallic above  $T_c$ ; more specifically, LiVS<sub>2</sub> is a metal down to 300 K and LiVSe<sub>2</sub> is a metal down to 2 K [18]. The question we address here is why, for  $T > T_c$ , an insulator to metal transition is observed by increasing the anion size. In early works, this was seen as a consequence of an increase in W due to chemical pressure [18], hence to a classical Mott mechanism. Here we show that this explanation does not hold, however. Instead, the behavior is well captured by the different path to metallization described in the last panel of Fig. 1.

Theoretical method. We adopt the density functional theory + dynamical mean-field theory (DFT + DMFT) approach to construct and solve the  $t_{2g}$  Hubbard model describing the correlated V bands close to the Fermi level [28–30]. The noninteracting part of the model is obtained from linearized augmented plane-wave band-structure calculations based on the WIEN2K code [31]. From the latter we construct Wannier functions spanning the  $t_{2g}$  bands and obtain the associated Hubbard model:

$$\begin{split} \hat{H} &= -\sum_{ii'\sigma} \sum_{mm'} t_{mm'}^{i,i'} c_{im\sigma}^{\dagger} c_{i'm'\sigma} + U \sum_{im} \hat{n}_{im\uparrow} \hat{n}_{im\downarrow} \\ &+ \frac{1}{2} \sum_{i\sigma\sigma'} \sum_{m \neq m'} (U - 2J - J \delta_{\sigma,\sigma'}) \hat{n}_{im\sigma} \hat{n}_{im'\sigma'} \\ &- J \sum_{im \neq m'} \left( c_{im\uparrow}^{\dagger} c_{im\downarrow}^{\dagger} c_{im'\uparrow} c_{im'\downarrow} + c_{im\uparrow}^{\dagger} c_{im\downarrow} c_{im'\downarrow}^{\dagger} c_{im'\uparrow} \right). \end{split}$$
(1)

Here m and m' are V  $t_{2g}$  orbitals and i site indices; the parameters  $t_{mm'}^{i,i'}$  are the hopping integrals  $(i \neq i')$  or crystalfield matrix elements (i = i'). The lattice structure is shown in Fig. 2. The site symmetry at the V site is  $D_{3d}$ , so that the  $t_{2g}$  states split into an  $a_{1g}$  singly degenerate level and an  $e_g^\pi$  doublet; for ideal regular octahedra,  $a_{1g}: \frac{1}{\sqrt{3}}|xy-yz+xz\rangle$  and  $e_g^\pi: \{\frac{1}{\sqrt{6}}|2xy+yz-xz\rangle, \frac{1}{\sqrt{2}}|yz+xz\rangle\}$ . The bare CF splitting

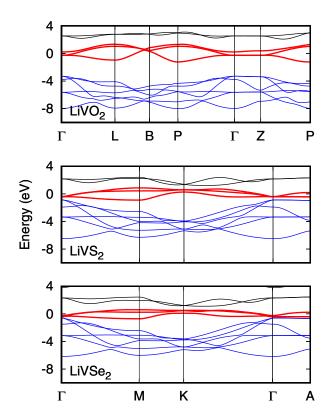


FIG. 3. LiV $X_2$  band structure with increasing anion size. From top to bottom: The  $t_{2g}$  bandwidth W (red) decreases, and the anion bands (blue) move closer to the Fermi level.

between  $a_{1g}$  and  $e_g^\pi$  states is  $\varepsilon_{\text{CF}} = 126$ , 104, and 96 meV for X = O, S, and Se, respectively, with the  $a_{1g}$  state having higher energy in all cases. We solve the correlated quantum-impurity problem via the weak-coupling continuous-time quantum Monte Carlo method [32], as implemented in Refs. [33–36]. Finally, we calculate the screened Coulomb parameters U and J by the constrained random-phase approximation method [37] via the GAP2 code [38], obtaining  $U \approx 4.3$  eV and  $J \approx 0.49$  eV for X = O,  $U \approx 2.2$  eV and  $J \approx 0.38$  eV for X = S, and  $U \approx 1.9$  eV and  $J \approx 0.36$  eV for X = Se [39]. The experimental crystal structure data for LiVO<sub>2</sub>, LiVS<sub>2</sub>, and LiVSe<sub>2</sub> are from Refs. [10–12,25]. The average Coulomb repulsion is given by  $U_{av} = U - 2J$ .

Results. Figure 3 shows the evolution of the band structure when X changes from O to Se. The figure shows that the  $t_{2g}$  bandwidth decreases from  $W \approx 2.64$  eV for X = O to  $W \approx 1.82 \text{ eV}$  for X = Se. The origin of the bandwidth reduction can be traced down to the increase of the V-V distance, which strongly suppresses direct V-V hopping integrals. The difference in the staking of layers  $(R\bar{3}m)$  space group for LiVO<sub>2</sub> and  $P\bar{3}m1$  for the other two systems) has instead a small role; in fact, an idealized calculation for LiVO2 with the  $P\bar{3}m1$  structure even yields a slightly larger  $W\approx 2.72$  eV. The reduction of W with increasing anionic size, with everything else staying the same, suggests that LiVSe<sub>2</sub> should be more correlated than LiVO<sub>2</sub>. Intra  $t_{2g}$  multiorbital effects, such as the reduction in CF splitting,  $\varepsilon_{CF}$ , could of course affect this conclusion, however. In fact, a decrease in  $\varepsilon_{CF}$  can enhance orbital fluctuations, reducing the strength of correlations [8].

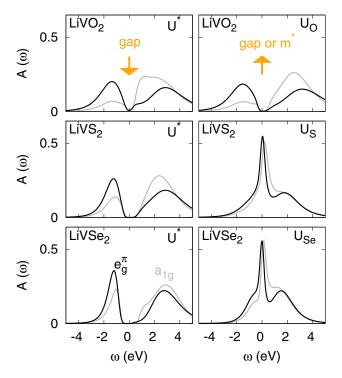


FIG. 4. Spectral functions for constant  $U_{\rm av}=U^*$  (left) and including X-dependent screening (right) [39]. Left column: The gap increases with increasing anionic radius. Right column: The opposite behavior is observed. Calculations are for  $\beta=10~{\rm eV}^{-1}$  (above  $T_c$ ). Orange arrows: Direction in which the strength of correlation increases in each column.

Furthermore, in the  $t_{2g}^2$  configuration, the Hund's rule coupling can also modify the strength of correlations [4–6], and cRPA calculations show that J decreases (from 0.49 to 0.36 eV) with increasing ionic size. Hence, the metallic behavior could be the result of changes in orbital physics *within* the space of correlated  $t_{2g}$  states. This, as we discussed in the Introduction, remains inside the classic Mott picture.

In order to put this hypothesis to the test, we first perform DFT + DMFT calculations with constant  $U_{av} = U^* \approx 4$  eV, using in each case the cRPA J values. The results are shown in Fig. 4, left-hand side. For LiVO<sub>2</sub>, top left, we obtain an insulating solution with a gap close to the experimental value. Also the other two systems, however, turn out insulators in the calculations; furthermore, the gap increases with increasing the size of the anion. This behavior could arise from suppression of orbital fluctuation, bandwidth reduction, or both combined. To disentangle these effects, we calculate the orbital polarization  $p(X) = \frac{1}{2}n_{e_g^{\pi}} - n_{a_{1g}}$ . The latter ranges from p=-2.0 [orbital ordering with the  $a_{1g}^2(e_g^\pi)^0$  configuration] to p=1.0 [orbital ordering with the  $a_{1g}^{0.5}(e_g^{\pi})^2$  configuration]; stong orbital fluctuations yield  $p\approx 0$ . Here we find  $p(O)\approx 0$ 0.52,  $p(S) \approx 0.35$  and  $p(Se) \approx 0.28$  for  $\beta = 10 \text{ eV}^{-1}$ . Everything else being the same, this suggests [2,8] that the correlation strength is weaker for X = Se than for X = O. Thus, the main cause of the gap increase appears to be the reduction of W. Either way, the conclusion is that, from the classic Mott viewpoint, all LiV $X_2$  systems should be insulators, with the largest gap in LiVSe<sub>2</sub> [40].

Next we consider the alternative mechanism illustrated in Fig. 1. To this end, we perform additional DFT + DMFT calculations, but this time taking into account materialsdependent screening effects, as obtained in the cRPA. Remarkably we find an unexpectedly strong decrease in  $U_{av}$ with increasing anion size: for X = Se,  $U_{av}$  becomes about one half of the X = O value. The resulting spectral functions are shown in the right-hand column of Fig. 4; one can see that the trends with X are the *opposite* of those obtained in the left-hand column. The effective masses  $m^*$  increase going from the bottom to the top of the figure (orange arrow); a gap only opens in the case of LiVO<sub>2</sub>. This is in line with experimental findings. The last question that remains to be addressed is what reduces  $U_{av}$  so drastically, sufficiently to reverse the trend. Our results show that the main cause is the reduction of the charge-transfer energy between anionic p and transition-metal 3d bands (Fig. 3), rather than the expansion of the wave functions. In fact, the bare direct Coulomb repulsion  $U_{\text{bare}}$  decreases from 16.6 to 13.5 eV going from X = Oto X = Se; the ratio between screened and bare Coulomb repulsion,  $r_X = U/U^{\text{bare}}$ , however, takes the value  $r_0 = 0.26$ for LiVO<sub>2</sub>,  $r_S = 0.16$  for LiVS<sub>2</sub>, and  $r_{Se} = 0.14$  for LiVSe<sub>2</sub>. This is exactly the mechanism described in Fig. 1.

Conclusion and outlook. We have shown that, surprisingly, in correlated TMCs, it is possible to obtain an insulator to metal transition by lattice expansion, i.e., decreasing the bandwidth W—as opposed to the classic Mott picture, in which lattice compression is key, so that the bandwidth increases. This unusual behavior can be realized by decoupling the main factors that control bandwidth and screening. In TMCs this disentanglement can be achieved, e.g., replacing O with larger anions of the same group. We have shown that this mechanism finds a realization in the  $LiVX_2$  series. The ideas outlined in this Letter are not confined to these compounds, however; they are likely to be at play in other TMCs, in particular, edge-sharing systems, currently the subject of intense research [41–45], since their structure promotes sizable direct-hopping integrals. The results obtained can also provide guidelines to tune correlations in devices based on these materials, e.g., energy-efficient batteries [46,47] or neuromorphic computers [48].

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Data availability. The data that support the findings of this article are not publicly available upon publication because it is not technically feasible and/or the cost of preparing, depositing, and hosting the data would be prohibitive within the terms of this research project. The data are available from the authors upon reasonable request.

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