

Compact extreme ultraviolet source for laboratory-based photoemission spectromicroscopy

Christoph Schmitz, Daniel Wilson, Denis Rudolf, Carsten Wiemann, Lukasz Plucinski, Sally Riess, Martin Schuck, Hilde Hardtdegen, Claus M. Schneider, F. Stefan Tautz, and Larissa Juschkin

Citation: *Applied Physics Letters* **108**, 234101 (2016); doi: 10.1063/1.4953071

View online: <http://dx.doi.org/10.1063/1.4953071>

View Table of Contents: <http://scitation.aip.org/content/aip/journal/apl/108/23?ver=pdfcov>

Published by the [AIP Publishing](#)

Articles you may be interested in

[Note: Thickness determination of freestanding ultra-thin foils using a table top laboratory extreme ultraviolet source](#)

Rev. Sci. Instrum. **84**, 056109 (2013); 10.1063/1.4807153

[Plasma-oxidation of Ge\(100\) surfaces using dielectric barrier discharge investigated by metastable induced electron spectroscopy, ultraviolet photoelectron spectroscopy, and x-ray photoelectron spectroscopy](#)

J. Appl. Phys. **110**, 033302 (2011); 10.1063/1.3611416

[Tunable ultrafast extreme ultraviolet source for time- and angle-resolved photoemission spectroscopy](#)

Rev. Sci. Instrum. **81**, 073108 (2010); 10.1063/1.3460267

[Properties of p - n heterojunction diode based on Ge 2 Sb 2 Te 5 and its application for phase change random access memory](#)

J. Appl. Phys. **105**, 061627 (2009); 10.1063/1.3055417

[Potential of discharge-based lithium plasma as an extreme ultraviolet source](#)

Appl. Phys. Lett. **89**, 031503 (2006); 10.1063/1.2227560

The banner features a blue background with a molecular structure of spheres and sticks on the left. On the right, the text 'NEW Special Topic Sections' is written in large, white, sans-serif font. Below this, the text 'NOW ONLINE' is in yellow, followed by 'Lithium Niobate Properties and Applications: Reviews of Emerging Trends' in white. The AIP Applied Physics Reviews logo is in the bottom right corner. On the left, there is a small inset image of a book cover for 'AIP Applied Physics Reviews' showing a technical diagram of a device structure.

NEW Special Topic Sections

NOW ONLINE
Lithium Niobate Properties and Applications:
Reviews of Emerging Trends

AIP Applied Physics Reviews

Compact extreme ultraviolet source for laboratory-based photoemission spectromicroscopy

Christoph Schmitz,^{1,a)} Daniel Wilson,^{1,2} Denis Rudolf,^{1,2} Carsten Wiemann,¹ Lukasz Plucinski,¹ Sally Riess,¹ Martin Schuck,¹ Hilde Hardtdegen,¹ Claus M. Schneider,¹ F. Stefan Tautz,¹ and Larissa Juschkin^{1,2}

¹Peter Grünberg Institut, Forschungszentrum Jülich, JARA-FIT, 52425 Jülich, Germany

²Experimental Physics of EUV, RWTH Aachen University, JARA-FIT, 52074 Aachen, Germany

(Received 31 December 2015; accepted 22 April 2016; published online 6 June 2016)

We report on the combination of a state-of-the-art energy-filtering photoemission electron microscope with an intense yet compact laboratory-based gas discharge extreme ultraviolet (EUV) light source. Using a photon energy of 71.7 eV from oxygen plasma (O^{5+} spectral line), we demonstrate element-selective photoelectron imaging in real space and band structure mapping in reciprocal space. Additionally, the high surface sensitivity of the EUV light was used to study the surface oxidation on islands of the phase-change material $Ge_1Sb_2Te_4$. The EUV light source allows the extension of spectromicroscopy, previously only feasible at synchrotron beamlines, to laboratory-based work.

Published by AIP Publishing. [<http://dx.doi.org/10.1063/1.4953071>]

With the upsurge of functionalized microstructures in various academic disciplines, the spatially resolved characterization of the chemical and electronic properties on the sub-micron scale becomes progressively more important. If a photoemission electron microscope is combined with an energy analyzer, the imaging of energy-filtered electrons becomes possible (EF-PEEM). Such instruments are known as spectromicroscopes. To exploit the full potential of spectromicroscopy, an intense, monochromatic yet tunable light source of defined polarization in the extreme ultraviolet (EUV) and soft X-ray spectral range is required. Therefore, spectromicroscopy is commonly carried out at undulator beamlines of 3rd generation synchrotron sources.¹ However, this limits its use to a small scientific community. If an intense laboratory-based light source tunable in the suitable spectral range was available, the application potential of spectromicroscopy could be vastly enhanced, allowing for both electronic structure (valence band spectroscopy) and chemical composition (core level spectroscopy) imaging of many samples on a routine basis. In this context, EUV radiation plays an important role, as photoemission with EUV light offers several advantages. First, the photoionization cross section for many elements is maximized in the EUV ensuring high photoemission signals.^{2,3} Second, the electron inelastic mean free path (IMFP) has a minimum of around 70 eV of electron kinetic energy with an average value below 1 nm maximizing the surface sensitivity.^{4,5} Unfortunately, however, the currently available standard laboratory-based light sources for photoemission⁶ do not cover the EUV spectral gap between 40.8 eV (UPS) and 1.235 keV (X-ray photoelectron spectroscopy, XPS). This calls for the development of a suitable EUV light source.

In this letter, we report on proof-of-concept experiments combining an intense, yet compact gas discharge plasma EUV light source and a state-of-the-art PEEM (based on a

Focus NanoESCA^{7–11}). The EUV source emits in the spectral range between vacuum ultraviolet (20 eV) and soft X-rays (600 eV), thus covering the lower part of the spectral gap and allowing for electronic and chemical analyses in a single experiment. To demonstrate the feasibility of the approach, we show examples of real space spectromicroscopy and reciprocal space band imaging at a photon energy of $h\nu = 71.7$ eV.

Initially designed for EUV lithography,¹² the EUV source was already applied in EUV microscopy^{13,14} and spectroscopy.¹⁵ Using a high current discharge to ignite a gas plasma, the source generates a spectrum of narrow-bandwidth ($\Delta E/E = 10^{-3} - 10^{-5}$) spectral lines with highest intensity between 20 eV and 600 eV photon energy emitted by multiple ionized atoms, e.g., O^{2+} – O^{6+} , etc. The abundance and the intensity of the observed spectral lines—corresponding to the emitted photon energies—strongly depends on the plasma temperature, plasma density, and gas composition. A typical spectrum of an oxygen plasma, which is used in this study, is shown in the inset of Fig. 1. The EUV source operates in a pulsed charge-discharge mode with a tunable repetition frequency in the range of 1.5–2.5 kHz, whereby each discharge generates an EUV pulse with an average pulse length of several tens of nanoseconds. Due to the high brightness of the source, the setup is expected to fulfill the requirements for live image acquisition in (EF-) PEEM. The challenge in combining this light source with a photoemission setup mainly lies in the construction of the transfer optics that is necessary to monochromatize and focus the EUV light onto the sample. To minimize the EUV intensity losses caused by absorption, the full setup is operated under vacuum, and only reflecting optics are used. Additionally, to maintain the required UHV conditions in the analyzer (10^{-9} mbar) while having plasma operation conditions (10^{-1} mbar) in the discharge region of the source, a differential pumping scheme and a thin, almost EUV transparent Al(150 nm)/C(27 nm) foil are used for vacuum separation. Since the emitted EUV spectrum consists of

^{a)}Author to whom correspondence should be addressed. Electronic mail: c.schmitz@fz-juelich.de

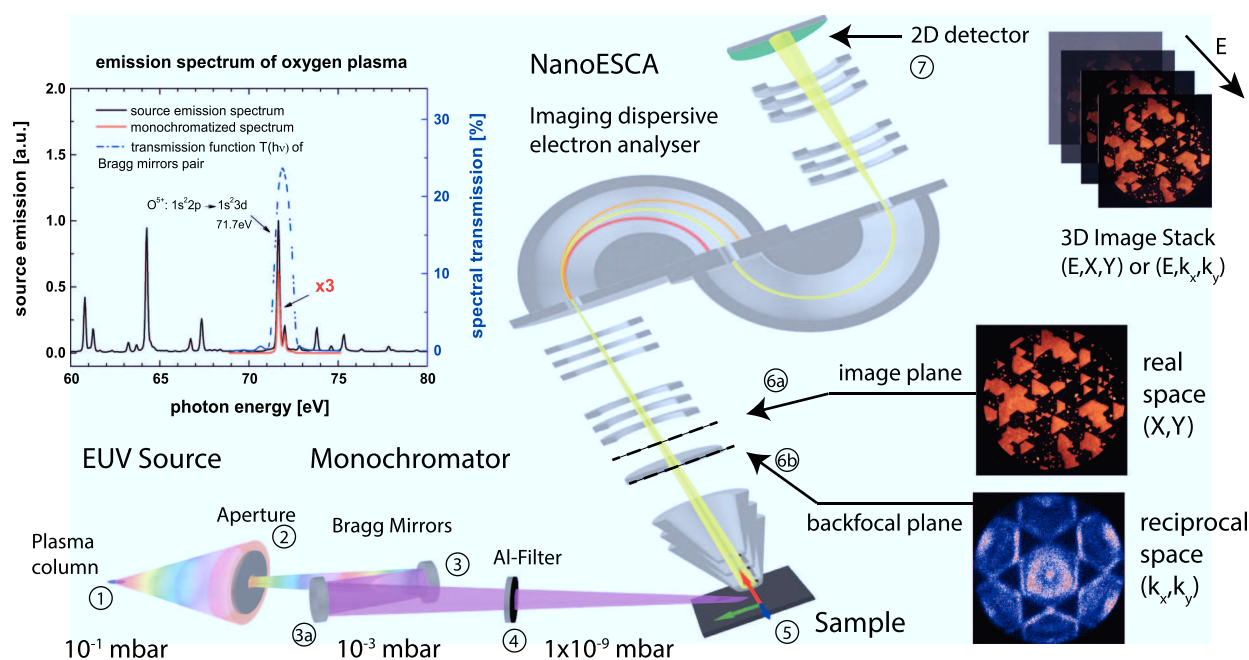


FIG. 1. Experimental setup: A discharge plasma within the EUV source generates a multi-wavelength EUV pulse (1). The radiation is collimated by a small aperture (2) and monochromatized (3) by a pair of narrow bandwidth Bragg mirrors (FWHM = 1.0 eV). The second mirror (3a) additionally focuses the monochromatized light onto the sample. An EUV-transparent Al filter of 150 nm thickness (4) is used for vacuum differentiation of the source and the UHV PEEM vacuum vessels. Photo-emitted electrons from the sample surface (5) are imaged by the EF-PEEM onto a 2D detector (7) behind two hemispherical analyzers. The microscope has two modes of operation: In **real space mode** (6a), the lateral intensity distribution $I(x, y, E)$ of photoelectrons for different kinetic energies E is projected. In **reciprocal space mode** (6b), the energy-dependent angular distribution of the electrons and momentum maps $I(k_x, k_y, E)$ can be derived. Inset: Measured emission spectrum of oxygen plasma before monochromatization (black) and calculated spectrum behind the Bragg mirror pair (red). The blue line shows the spectral transmission of the Bragg mirror pair measured at the synchrotron.¹⁶

many isolated emission lines (Fig. 1, inset), each of which has a sufficiently small intrinsic linewidth ($\Delta E < 10$ meV due to Doppler broadening in the case of oxygen ions), no wide-range high-resolution monochromator is required. Instead, a set of Al/Mo-based Bragg mirrors consisting of a flat and a concave mirror (1 m radius) is used as narrow spectral bandwidth filter (FWHM = 1.0 eV [Ref. 16]). The Bragg mirrors are optimized to select and focus only the spectral lines emitted by Li-like O^{5+} ions around $h\nu = 71.7$ eV photon energy ($1s^2 2p \rightarrow 1s^2 3d$ doublet transition) onto the sample. The final spot size on the sample is 3.0×1.7 mm², and the total photon flux after spectral discrimination of 71.7 eV photon energy was measured by a calibrated photodiode to be approximately 2×10^{11} photons/s with a relative spectral resolution of about 1200. The photon flux and spectral resolution have the same order of magnitude as, e.g., the BaD EIPh vacuum ultraviolet photoemission beamline at Elettra.¹⁷ The performance of the setup using EUV excitation was tested by determining the spatial resolution with the help of a calibration sample. The calibration sample exhibits checkerboard structures consisting of neighboring Au and Si squares having a size of 1×1 μ m². PEEM images have been acquired with the EUV source (71.7 eV), and, for reference, with a conventional Hg lamp (5.2 eV) (see Fig. 2). The lens settings and the contrast aperture size have been kept the same for both measurements. The spatial resolution in the indicated region [green line in Fig. 2(a)] was determined by fitting a convolution of a Gaussian function and a step function to the measured line profile and extracting the full width at half maximum. For both photon energies, a sub-micron resolution was achieved. Specifically, the resolution amounts to 348.2 ± 110 nm

(71.7 eV) and 151.3 ± 9 nm (5.2 eV), respectively. The resolution derived for Hg excitation is in agreement with the instrumental resolution of 50–150 nm and the sample feature accuracy of <200 nm. The loss of spatial resolution at 71.7 eV as compared to 5.2 eV illumination may be attributed to (a) a reduced contrast for this specific photon energy and material combination, (b) chromatic aberrations due to a larger spread of electron kinetic energies, and (c) to the onset of space and surface mirror charges; for pulsed light sources, they are commonly observed to reduce the spatial resolution.¹⁸ All three effects have been observed in the course of our experiments. Further experiments are required to specifically quantify the origin and the impact of space charge effects. Next, we demonstrate the capability of the EUV source to perform chemically resolved spectromicroscopy based on core-level photoemission. To this end, we turned to a chemically

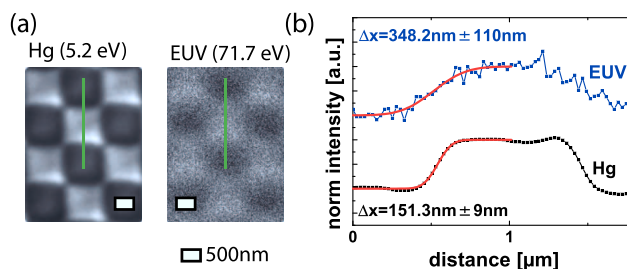


FIG. 2. PEEM images of a structured calibration sample with $1 \mu\text{m} \times 1 \mu\text{m}$ Au and Si squares at 5.2 eV (left) and 71.7 eV (right) photon energy. Both images show an average of 80 images at 10 s of integration time per image. The integrated intensity along the green lines together with fits of convolved Gaussian and step functions (red lines) are shown in (b). The spatial resolution is denoted by Δx .

complex and microscopically structured, ternary material system. Specifically, we investigated crystalline hexagonal GeSb-Te (GST, 1:2:4) islands epitaxially grown on Si (111)¹⁹ by metal organic vapor phase epitaxy (MOVPE). A particular focus of these experiments was on surface oxidation. GST is a prototype material for resistive switching based on phase change processes and it is well known from optical data storage in Compact Disc ReWritable (CD-RW) and Digital Versatile Disc ReWritable (DVD-RW).²⁰ Prior to our photoemission studies, the *ex-situ* prepared sample was carefully cleaned by mild sputtering (Ar ions, 500 eV) and annealing at 250 °C. Due to the high surface sensitivity of the method, this preparation step is mandatory. Fig. 3 shows EF-PEEM images of a GST sample recorded at $h\nu = 5.2$ eV (a) and $h\nu = 71.7$ eV (b). In both images, triangular-shaped islands ranging in size from several hundreds of nanometers up to 50 μm and more are visible. The contrast in Fig. 3(a) arises from a difference in work function Φ between GST and Si. The bright regions correspond to GST, whereas the dark regions are the bare Si substrate. Orientation, alignment, and shape of the GST islands confirm the expected epitaxial growth on Si(111). Fig. 3(b) displays the spatially resolved tellurium concentration based on spectromicroscopy images acquired over the energy range of the spin-orbit split Te 4d core level. In this case, the contrast is of chemical origin and stems from the presence or absence of Te atoms in the specific regions. Orange areas correspond to high Te concentrations and blue areas to low or zero Te concentrations. The Te intensity distribution clearly follows the shape of the GST islands as observed with the Hg lamp. Similar intensity distributions were acquired for Sb and Ge and their corresponding oxides

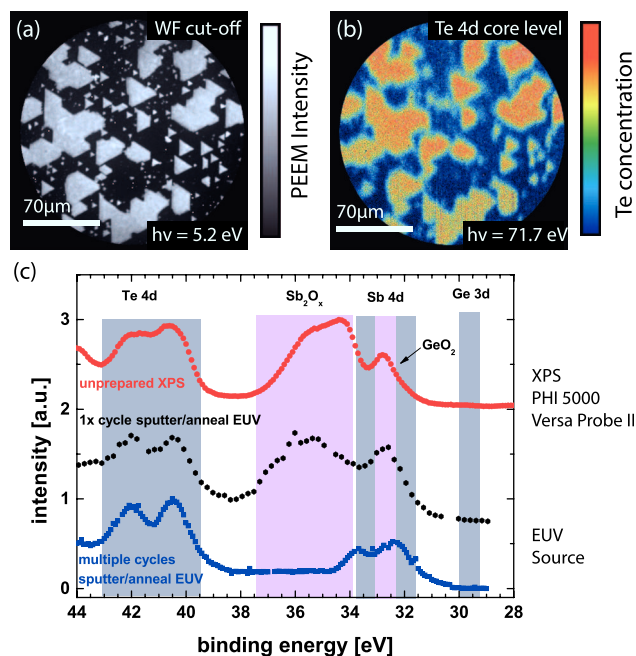


FIG. 3. EF-PEEM image of GST islands recorded at $h\nu = 5.2$ eV (a) and at $h\nu = 71.7$ eV (b) photon energy. Image (a) was acquired at the work function (WF) cut-off. Image (b) consists of integrated images around the Te 4d core level binding energies (39 eV–41 eV). The image is slightly shifted with respect to (a) and shows space charge induced image deterioration. In panel (c), a binding energy spectrum of gently sputtered GST islands acquired with the EUV source (black) is compared to XPS reference data of a pristine sample (red). The oxide components of Ge and Sb are clearly visible. Additional preparation cycles lead to a purely metallic Sb/Te termination (blue).

(not shown). As all core-level distribution maps resemble the same intensity profile on the μm scale without variations between different islands, we assume the chemical composition of the islands to be homogeneous and independent of the island's size. A photoemission spectrum including only the signal from GST islands was extracted from the 3D image stack and is presented in Fig. 3(c) (black data points). We note that the acquisition time for each data point in the spectrum is only 4 s. Still, the spin-orbit splitting of the Te 4d core level can be clearly identified.²¹ In addition, we can identify the energy-shifted Sb 4d peak of Sb_2O_3 as a fingerprint of the oxidized surface of *ex-situ* prepared GST. Very likely, the Ge 3d peak of GeO_2 at about 33 eV is also present in the spectrum, but obscured by the Sb 4d peak. The observed oxide components confirm the anticipated surface oxidation. To compare our data with a well-established method, we performed XPS with Al K_{α} radiation on the same sample in a standard XPS instrument (PHI 5000 VersaProbe II). Despite the different inelastic mean free paths ($\lambda_{\text{IMFP}} < 1$ nm (Ref. 5) at $h\nu = 71.7$ eV compared to a few nm at $h\nu = 1486.6$ eV), the EUV (black) and X-ray photoemission (red) spectra are very similar (Fig. 3(c)). Application of additional cycles of sputter/annealing, however, completely removed the oxide layer in favor of a metallic Te/Sb termination (blue line in Fig. 3(c)).

Finally, to demonstrate the capability of the setup to perform angle-resolved photoemission spectroscopy (ARPES) and band structure mapping, the NanoESCA is used in its reciprocal (“k-space”) mode to image the energy-dependent angular distribution of photoelectrons emitted from a Au (111) single crystal surface. The crystal was cleaned by multiple cycles of Ar ion sputtering and subsequent annealing to 500 °C. For the experiment, the energy resolution of the analyzer was set to $\Delta E = 400$ meV in order to achieve the best compromise between transmission, energy resolution, and acquisition time. Here, the acquisition time ranges from 1 min per energy step for the high-intensity Au 5d states (1–8 eV binding energy) in the standard mode up to 10 min per image when approaching the Fermi level E_F , where a special, single electron event counting mode¹⁰ was used. The calibration of the reciprocal field of view was performed by fitting a parabola to the secondary electron emission cone as described by Escher *et al.*⁹ At 71.7 eV photon energy, the maximum observable field of view (FoV) is approximately 5.6 \AA^{-1} . Within this FoV, the first as well as parts of the neighboring surface Brillouin zones can be imaged simultaneously (Fig. 4(a)). The symmetry of the (111) surface with indicated high symmetry points $\bar{\Gamma}$, \bar{M} and \bar{K} is clearly visible. In addition, the Shockley surface state (SS)²² located around $\bar{\Gamma}$ (starting 500 meV below the Fermi edge, see inset) can be identified in Fig. 4(b) as a central bright spot. The measured k-radius of the state at E_F is around 0.13 \AA^{-1} . Valence band photoemission spectra of the Au (111) surface were extracted by integrating over the full acquired k-space region (≈ 2.5 BZ) at different binding energies. The resulting spectrum is displayed in Fig. 4(e). The observed double peak structure at 3 eV and 6 eV binding energy arises from the two Au 5d bands, while the Fermi edge is mainly defined by the Au 6s state. For a qualitative interpretation of the acquired k-maps, *ab-initio* band structure calculations (GGA-PBE) of bulk Au (111) including spin-orbit coupling

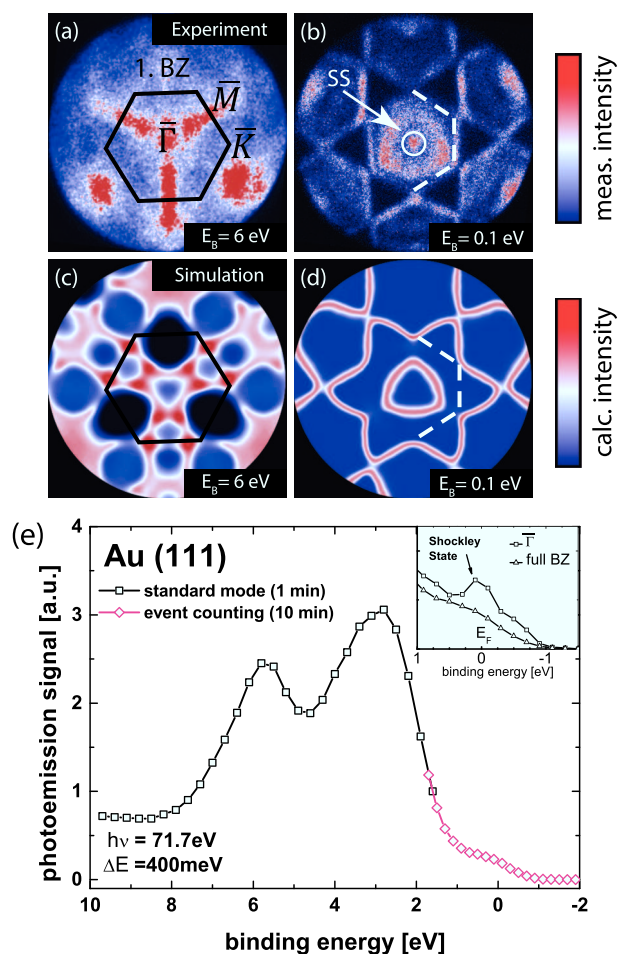


FIG. 4. Experimental k-space maps of Au 5d band (a) and Au(111) Fermi surface (b) showing a range of 5.6 \AA^{-1} in reciprocal space. The high symmetry points $\bar{\Gamma}$, \bar{M} and \bar{K} of the hexagonal surface Brillouin zone are marked within (a). Corresponding band structure calculations (WIEN2K) are shown in (c) and (d) for a qualitative comparison. The spectrum (e) was derived from integrated k-space images recorded at different binding energies in the standard (black squares) as well as in the event counting mode¹⁶ (red diamonds) Inset: magnified view of the Fermi edge integrated over a k-space region around $\bar{\Gamma}$ only and integrated over the full Brillouin zone (BZ).

were performed using WIEN2K²³ (Figs. 4(c) and 4(d)). The simulated raw data was artificially broadened by 400 meV to match the energy resolution of the experiment. The main features and symmetries of the experiment can be reproduced by the simulation with a good agreement. However, it must be mentioned that some additional features, such as the additional hexagon located in the center of the Fermi surface (dashed line Fig. 4(b)), cannot be explained by those basic simulations. We attribute this feature to a second final state transition.

To conclude, we presented highly surface-sensitive spectromicroscopy with a laboratory-based gas-discharge EUV light source and a state-of-the-art energy filtered PEEM. Both real and reciprocal space imaging were demonstrated showing that EUV radiation with $h\nu = 71.7 \text{ eV}$ from this source can be used to investigate the electronic valence structure as well as the chemical composition of a given sample in a single experiment. To put our results into perspective, only a few photoemission experiments (ARPES and PEEM) employing a laboratory-based EUV light source beyond He I (21.2 eV) and He II (40.8 eV) have been reported so far, mostly using

laser-generated high harmonics (HHG)^{24–26} and laser-produced plasma light sources.^{27–29} In future experiments, we plan to use a collector in combination with a grating monochromator to provide easy access to the different photon energies emitted by the source allowing for 3D molecular orbital tomography or depth-probing in XPS, and to further improve the energy resolution. Ultimately, the energy resolution of the source will be limited by Doppler broadening (about 10 meV) of the dominant spectral lines.

We acknowledge financial support from a JARA-FIT Seed Fund Project funded through the German Excellence Initiative. L.J. acknowledges financial support from the Helmholtz Association for a Helmholtz Professorship as a part of the Initiative and Networking Fund. We also thank B. Küpper and A. Bremen for their technical assistance throughout the project as well as A. Besmehn for XPS measurements. S.R., M.S., and H.H. as well as C.S., C.W., and C.M.S. acknowledge funding within the Collaborative Research Center 917 “Nanoswitches” by the Deutsche Forschungsgemeinschaft (DFG).

- ¹C. Wiemann, M. Patt, I. P. Krug, N. B. Weber, M. Escher, M. Merkel, and C. M. Schneider, *J. Surf. Sci. Nanotechnol.* **9**, 395 (2011).
- ²J. J. Yeh, *Atomic Calculation of Photoionization Cross-sections and Asymmetry Parameters* (Gordon & Breach Science, 1993).
- ³J. J. Yeh and I. Lindau, *Atomic Data and Nuclear Data Tables* (Elsevier Science, 1985), Vol. 32, pp. 1–155.
- ⁴C. J. Powell and A. Jablonski, NIST Electron Inelastic-Mean-Free-Path Database, Version 1.2, National Institute of Standards and Technology, Gaithersburg, MD, 2010.
- ⁵M. P. Seah and W. A. Dench, *Surf. Interface Anal.* **1**, 2 (1979).
- ⁶UPS: Ar: 11.6 eV, Kr: 10 eV, Ne: 16.848 eV and 16.671 eV, He I: 21.22 eV, He II: 40.8 eV, Hg: 4.8 eV and XPS: Mg K₂: 1253.5 eV, Al K₂: 1486.5 eV.
- ⁷M. Escher, N. Weber, M. Merkel, C. Zietzen, P. Bernhard, G. Schönhense, S. Schmidt, F. Forster, F. Reinert, B. Krömker, and D. Funnemann, *J. Phys.: Condens. Matter* **17**, 1329 (2005).
- ⁸N. Weber, M. Escher, M. Merkel, A. Oelsner, and G. Schönhense, *J. Phys.: Conf. Ser.* **100**, 72031 (2008).
- ⁹M. Escher, K. Winkler, O. Renault, and N. Barrett, *J. Electron Spectrosc. Related Phenom.* **178–179**, 303 (2010).
- ¹⁰M. Patt, C. Wiemann, N. Weber, M. Escher, A. Gloskovskii, W. Drube, M. Merkel, and C. M. Schneider, *Rev. Sci. Instrum.* **85**, 113704 (2014).
- ¹¹C. Wiemann, M. Patt, S. Cramm, M. Escher, M. Merkel, A. Gloskovskii, S. Thiess, W. Drube, and C. M. Schneider, *Appl. Phys. Lett.* **100**, 223106 (2012).
- ¹²S. Danylyuk, H. Kim, S. Brose, C. Dittberner, P. Loosen, T. Taubner, K. Bergmann, and L. Juschkin, *J. Vac. Sci. Technol. B* **31**, 21602 (2013).
- ¹³M. Benk, K. Bergmann, D. Schäfer, and T. Wilhein, *Opt. Lett.* **33**, 2359 (2008).
- ¹⁴L. Juschkin, R. Freiburger, and K. Bergmann, *J. Phys.: Conf. Ser.* **186**, 12030 (2009).
- ¹⁵D. Wilson, D. Rudolf, C. Weier, R. Adam, G. Winkler, R. Frömter, S. Danylyuk, K. Bergmann, D. Grützmacher, C. M. Schneider, and L. Juschkin, *Rev. Sci. Instrum.* **85**, 103110 (2014).
- ¹⁶The Bragg mirrors were fabricated by optiX fab GmbH. The reflectivity was measured at the PTB EUV reflectometry beamline in Berlin.
- ¹⁷L. Petaccia, P. Vilmercati, S. Gorovikov, M. Barnaba, A. Bianco, D. Cocco, C. Masciovecchio, and A. Goldoni, *Nucl. Instrum. Methods Phys. Res.* **606**, 780 (2009).
- ¹⁸A. Locatelli, T. O. Mentes, M. Á. Niño, and E. Bauer, *Ultramicroscopy* **111**, 1447 (2011).
- ¹⁹M. Schuck, S. Rieß, M. Schreiber, G. Mussler, D. Grützmacher, and H. Hardtdegen, *J. Cryst. Growth* **420**, 37 (2015).
- ²⁰M. Wuttig and N. Yamada, *Nat. Mater.* **6**, 824 (2007).
- ²¹NIST X-ray Photoelectron Spectroscopy Database, Version 4.1 (National Institute of Standards and Technology, Gaithersburg, 2012); <http://srdata.nist.gov/xps>.

- ²²M. Hoesch, M. Muntwiler, V. N. Petrov, M. Hengsberger, L. Patthey, M. Shi, M. Falub, T. Greber, and J. Osterwalder, *Phys. Rev. B* **69**, 241401 (2004).
- ²³P. Blaha, K. Schwarz, G. Madsen, D. Kvasnicka, and J. Luitz, *WIEN2K: An Augmented Plane Wave + Local Orbitals Program for Calculating Crystal Properties* (Karlheinz Schwarz, Technische Universität Wien, Austria, 2001).
- ²⁴S. Eich, A. Stange, A. V. Carr, J. Urbancic, T. Popmintchev, M. Wiesenmayer, K. Jansen, A. Ruffing, S. Jakobs, T. Rohwer, S. Hellmann, C. Chen, P. Matyba, L. Kipp, K. Rossnagel, M. Bauer, M. M. Murnane, H. C. Kapteyn, S. Mathias, and M. Aeschlimann, *J. Electron. Spectrosc. Relat. Phenom.* **195**, 231 (2014).
- ²⁵A. Mikkelsen, J. Schwenke, T. Fordell, G. Luo, K. Klünder, E. Hilner, N. Anttu, A. A. Zakharov, E. Lundgren, J. Mauritsson, J. N. Andersen, H. Q. Xu, and A. L'Huillier, *Rev. Sci. Instrum.* **80**, 123703 (2009).
- ²⁶S. H. Chew, F. Süßmann, C. Späth, A. Wirth, J. Schmidt, S. Zherebtsov, A. Guggenmos, A. Oelsner, N. Weber, J. Kapaldo, A. Gliserin, M. I. Stockman, M. F. Kling, and U. Kleineberg, *Appl. Phys. Lett.* **100**, 51904 (2012).
- ²⁷A. J. Arko, J. J. Joyce, and L. Morales, *J. Alloys Compd.* **286**, 14 (1999).
- ²⁸G. Schriever, S. Mager, A. Naweed, A. Engel, K. Bergmann, and R. Lebert, *Appl. Opt.* **37**, 1243 (1998).
- ²⁹H. Kondo, T. Tomie, and H. Shimizu, *Appl. Phys. Lett.* **72**, 2668 (1998).