

Probing microstructural origin of complex flow behavior

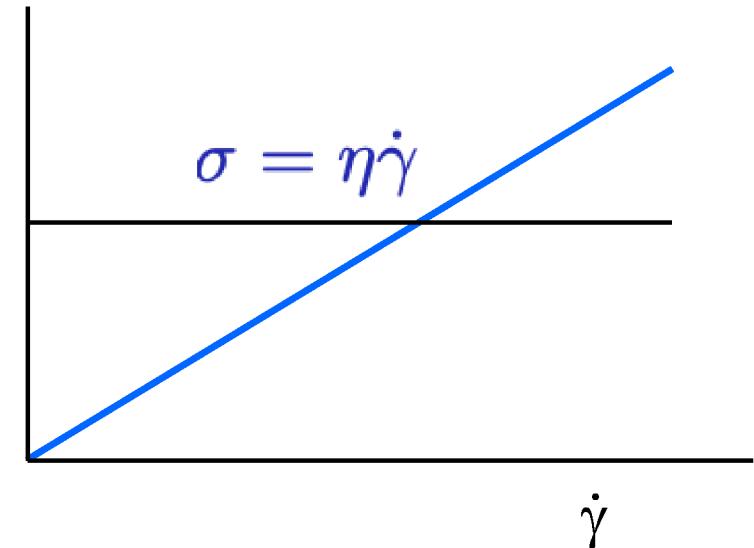
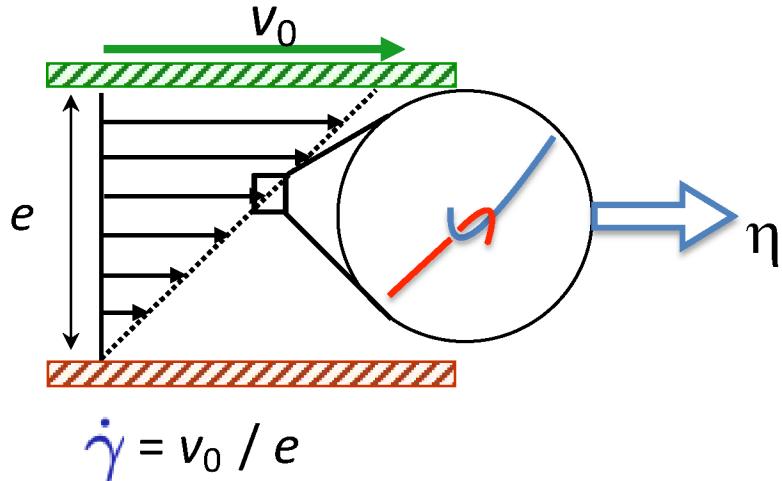
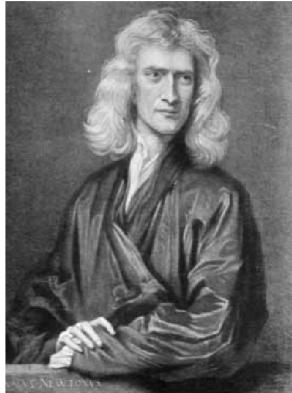
Pavlik Lettinga

UCSB, 2nd February 2018

Complex flow: high inertia

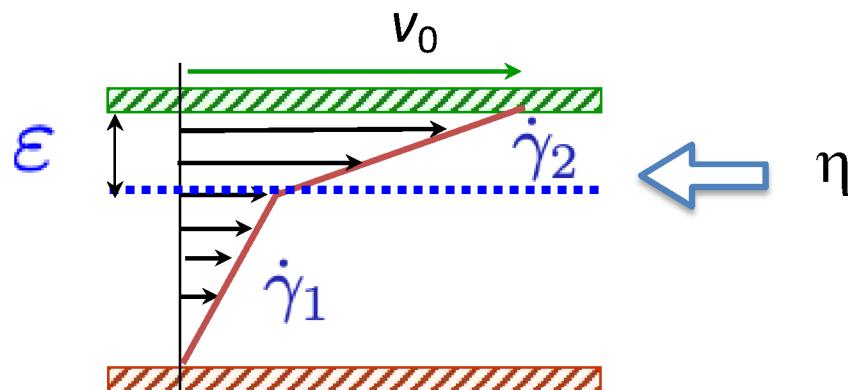


Ideal Newtonian fluids

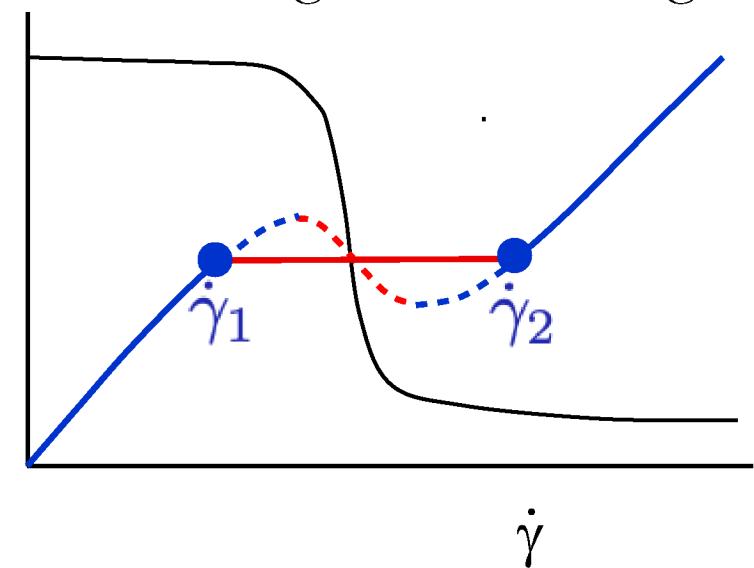


Non-linear Newton: shear thinning fluids

Flow instabilities: shear banding

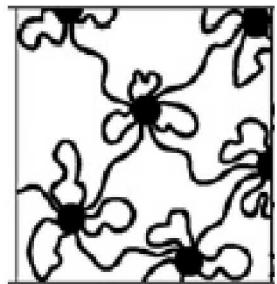


strong shear-thinning



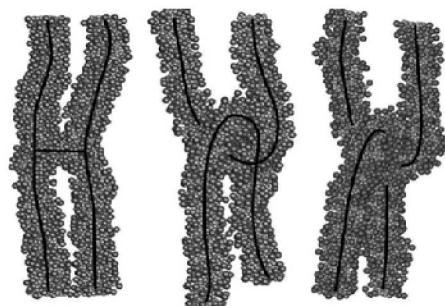
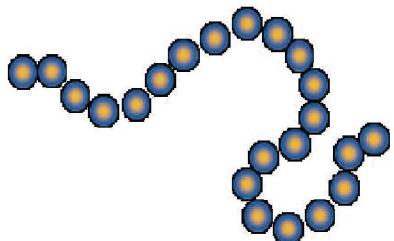
Possible shear thinners

Living gels:

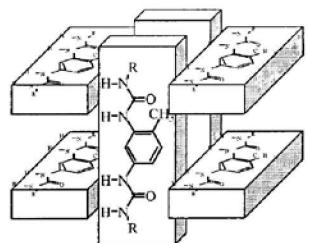


Sprakel et al, Soft Matter, 4,
(2008) 1696

Living polymers:

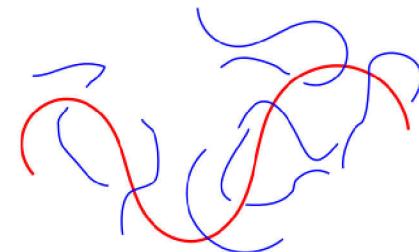


M. P. Lettinga and S. Manneville, *Phys. Rev. Lett.*, **103** 2009

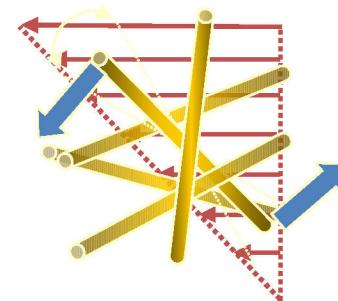


Van der Gucht et al *Phys. Rev. Lett.*, **97**, (2006) 108301

Stiff Polymers:



Rods:

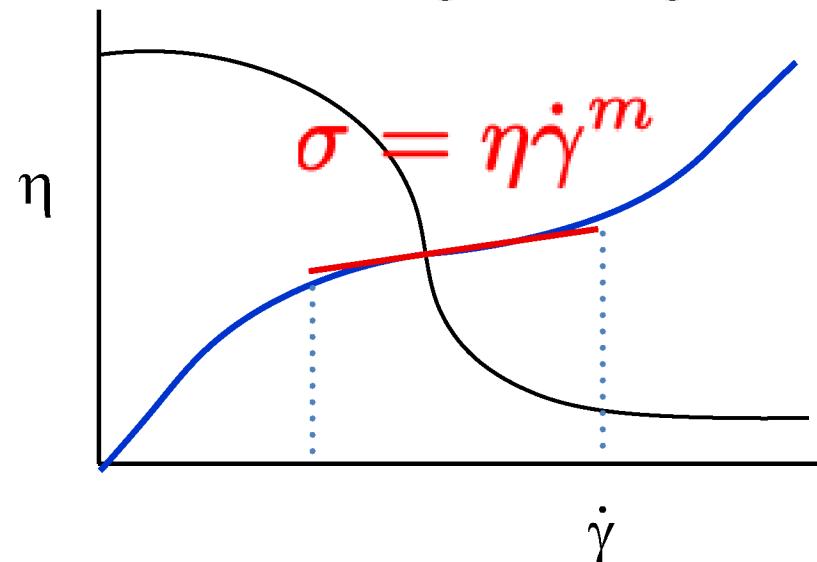


Main questions:

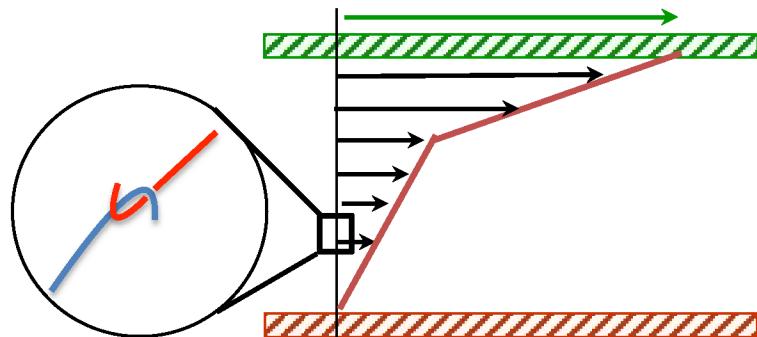
Molecular origin of shear band formation:

Can polymer shear band?
Can rods shear band?

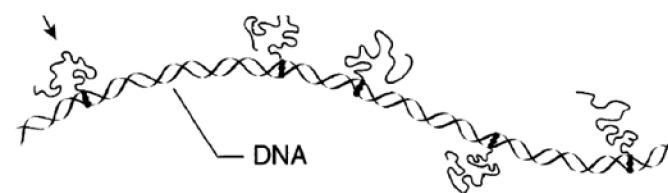
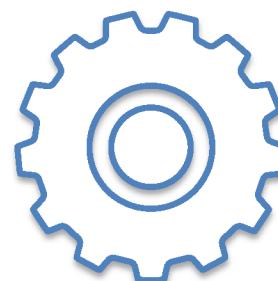
- How strong is strong?



- Always shear banding for given m , or is it system dependent?

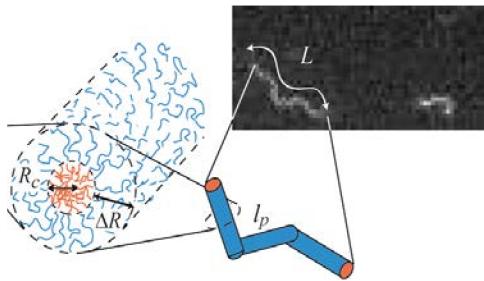


- Can we tune shear band formation?

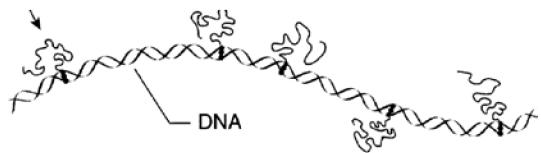


Contents: Tune stiffness, length and interaction

- Pb-Peo worm-like micelles:



- DNA



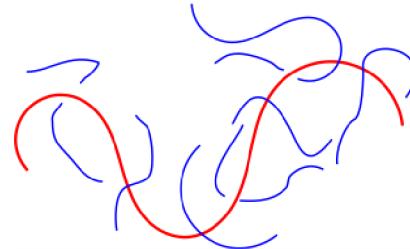
- F-actin:



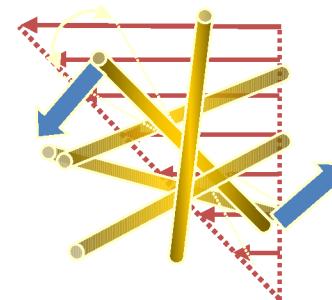
- Filamentous viruses:



Stiff Polymers:



Rods:



Smoluchowski theory for hard rods

Gives equation of motion for the orientational tensor \mathbf{S} :

$$\frac{d}{dt}\mathbf{S} = -6D_r \left\{ \mathbf{S} - \frac{1}{3}\hat{\mathbf{I}} + \frac{L}{D}\varphi \left(\mathbf{S}^{(4)} : \mathbf{S} - \mathbf{S} \cdot \mathbf{S} \right) \right\} + \dot{\gamma} \left\{ \hat{\boldsymbol{\Gamma}} \cdot \mathbf{S} + \mathbf{S} \cdot \hat{\boldsymbol{\Gamma}}^T - 2\mathbf{S}^{(4)} : \hat{\mathbf{E}} \right\}$$

→ Link with macroscopic stress

$$\Sigma_D = 2\eta_0\dot{\gamma} \left[\hat{\mathbf{E}} + \frac{(L/D)^2}{3\ln\{L/D\}}\varphi \times \left\{ \hat{\boldsymbol{\Gamma}} \cdot \mathbf{S} + \mathbf{S} \cdot \hat{\boldsymbol{\Gamma}}^T - \mathbf{S}^{(4)} : \hat{\mathbf{E}} - \frac{1}{3}\hat{\mathbf{I}}\mathbf{S} : \hat{\mathbf{E}} - \frac{1}{\dot{\gamma}} \frac{d\mathbf{S}}{dt} \right\} \right]$$

Collective slowing down: Dynamic definition spinodal point

→ $\delta S(t) = \exp(-6D_R^{eff}t)\delta S(t=0)$

$$D_R^{eff} = D_R^0 \left(1 - \frac{1}{4} \frac{L}{d_{eff}} \varphi \right) \longrightarrow \Omega_{eff} = \omega / D_R^{eff}$$

$$\longrightarrow Pe_{eff} = \dot{\gamma}_0 / D_R^{eff}$$

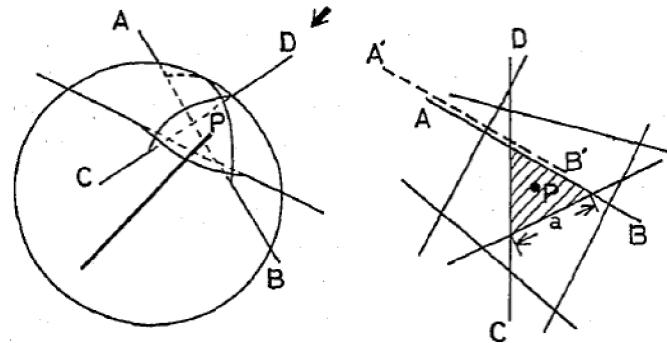
D_R^0 : rotational at *infinite* dilution



Topological slowing down:

Dois phenomenological rotational diffusion coefficient

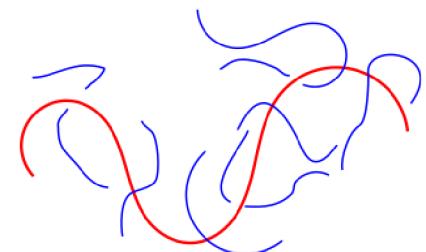
$$D_r = c D_r^0 (v L^3)^{-2}$$



Monotonic constitutive theory for polymeric liquids

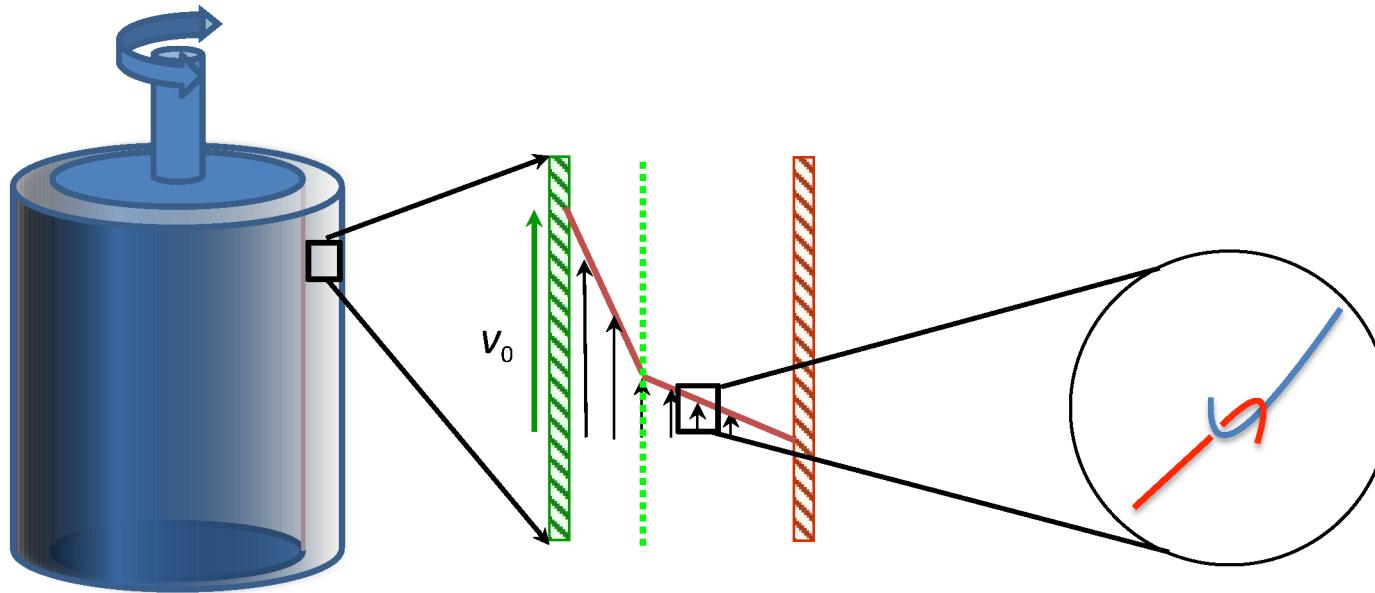
Competition of shear flow with Rouse and reptation time

non-monotonic behavior due to concentration coupling



Cromer et al, *Phys. Fluids*, 2013

Experimental input needed:

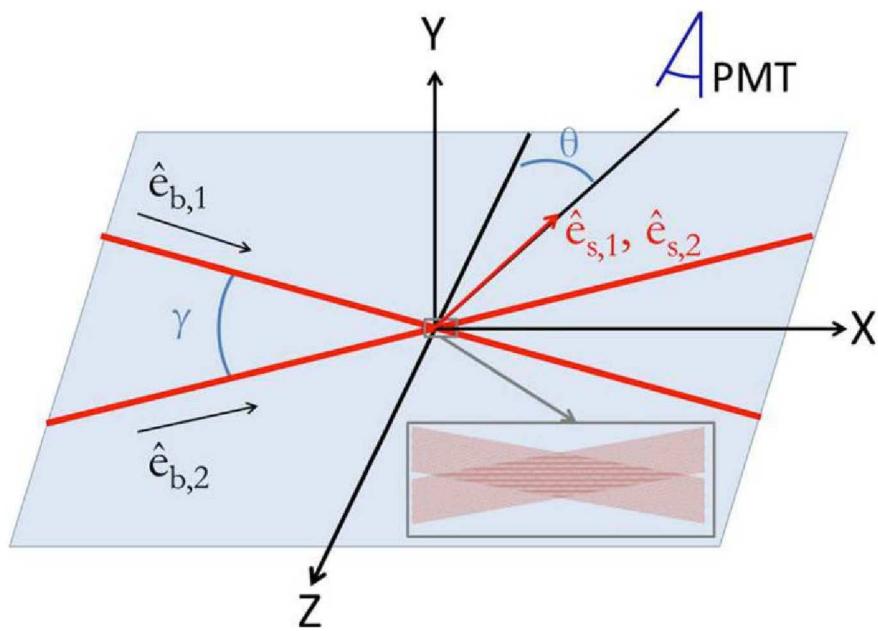
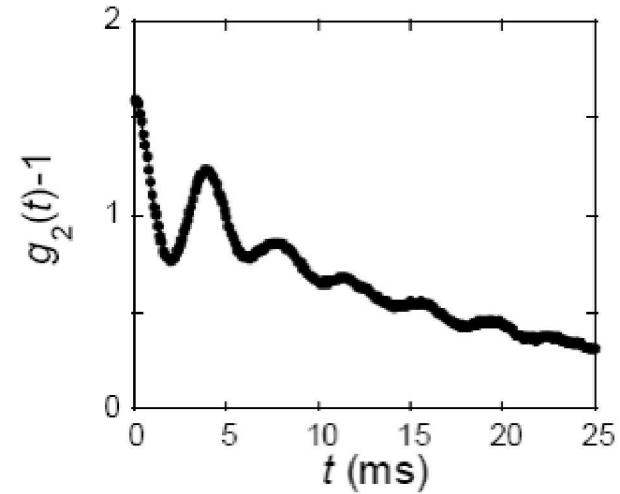
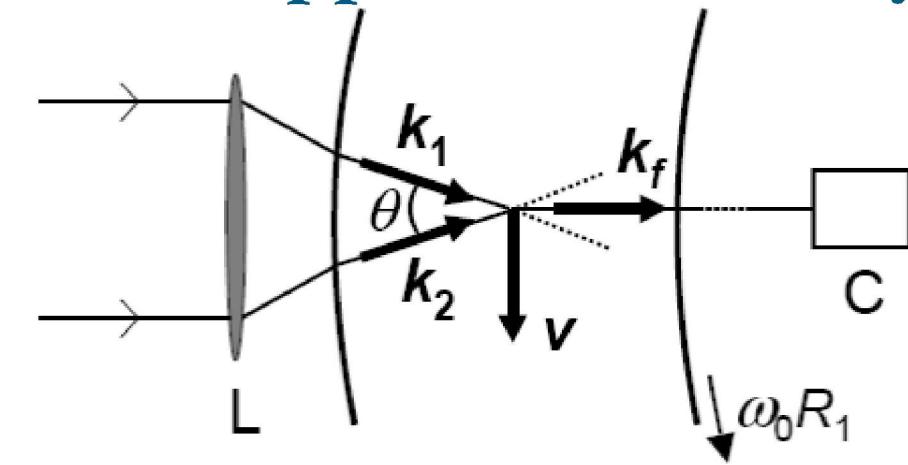


Information needed:

- Probe the mechanical response of the system.
- Probe the stability of the flow.
- Probe structure *in situ* over broad range of length-scales and time-scales.

Probe the stability of flow with

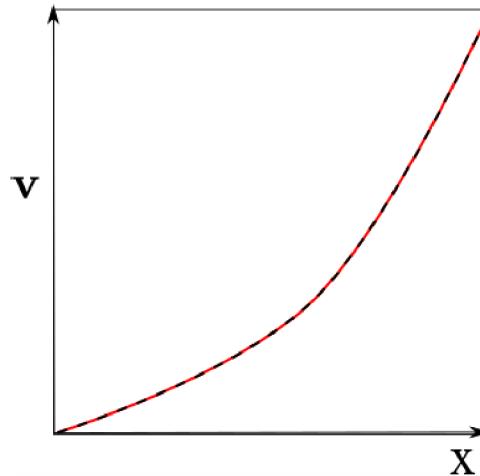
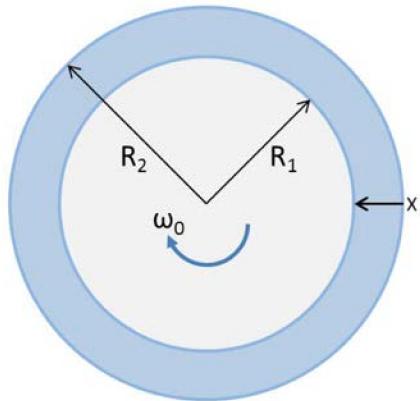
Laser Doppler Velocimetry



Analyse velocity profiles

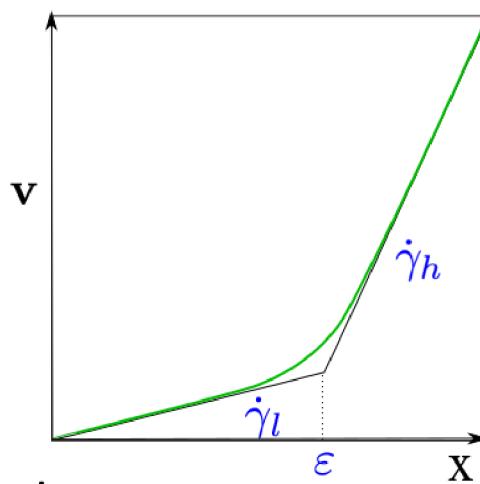
$$V(x) = \omega_0 R_1 [(R_2 - x)^{1-2/m} - R_2^{1-2/m}] / [R_1^{1-2/m} - R_2^{1-2/m}]$$

Account for curvature cell:



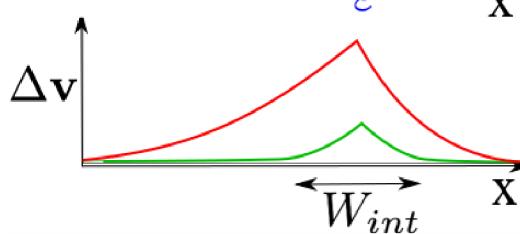
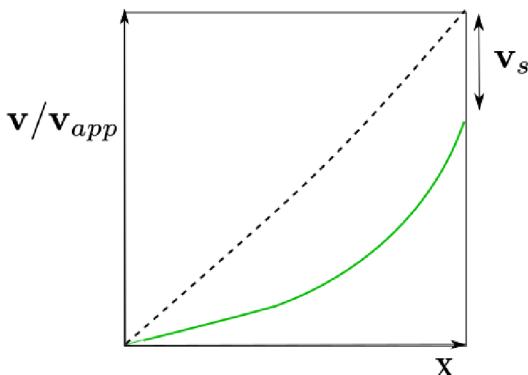
m
compare with m_{fc}

Shear banding with interface:



$\dot{\gamma}_h$ $\dot{\gamma}_l$ ε

Wall slip:

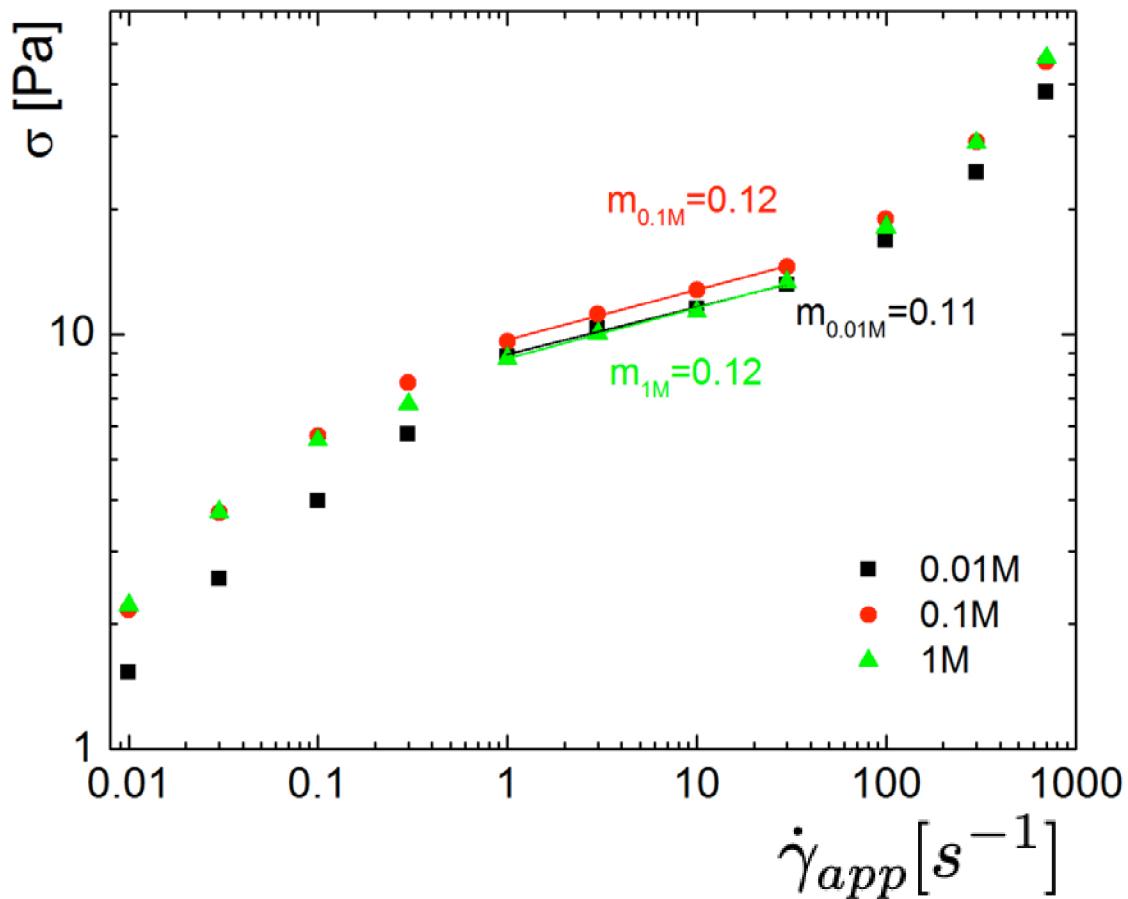
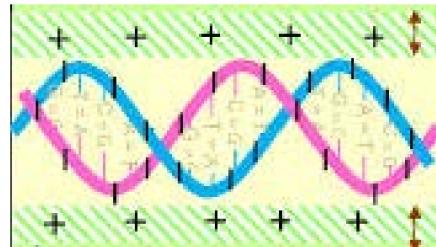


W_{int}

DNA, the tuneable polymer part I

$\langle L \rangle \approx 20 \text{ } \mu\text{m}$, $d = 7 \text{ nm}$, $l_p = 50 \text{ nm}$

Tune repulsion by adding salt:

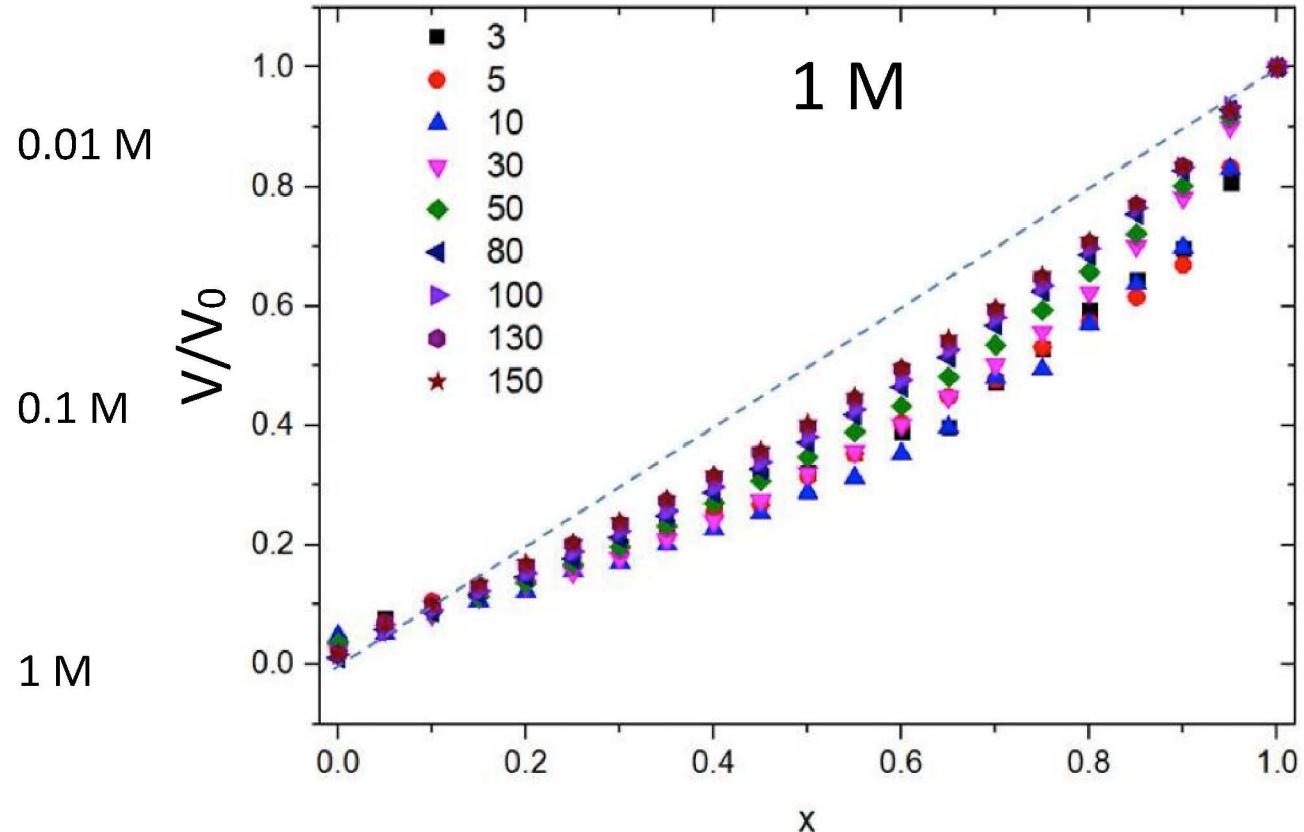


concentration: 0.7 mg/ml

Tuning by addition of salt

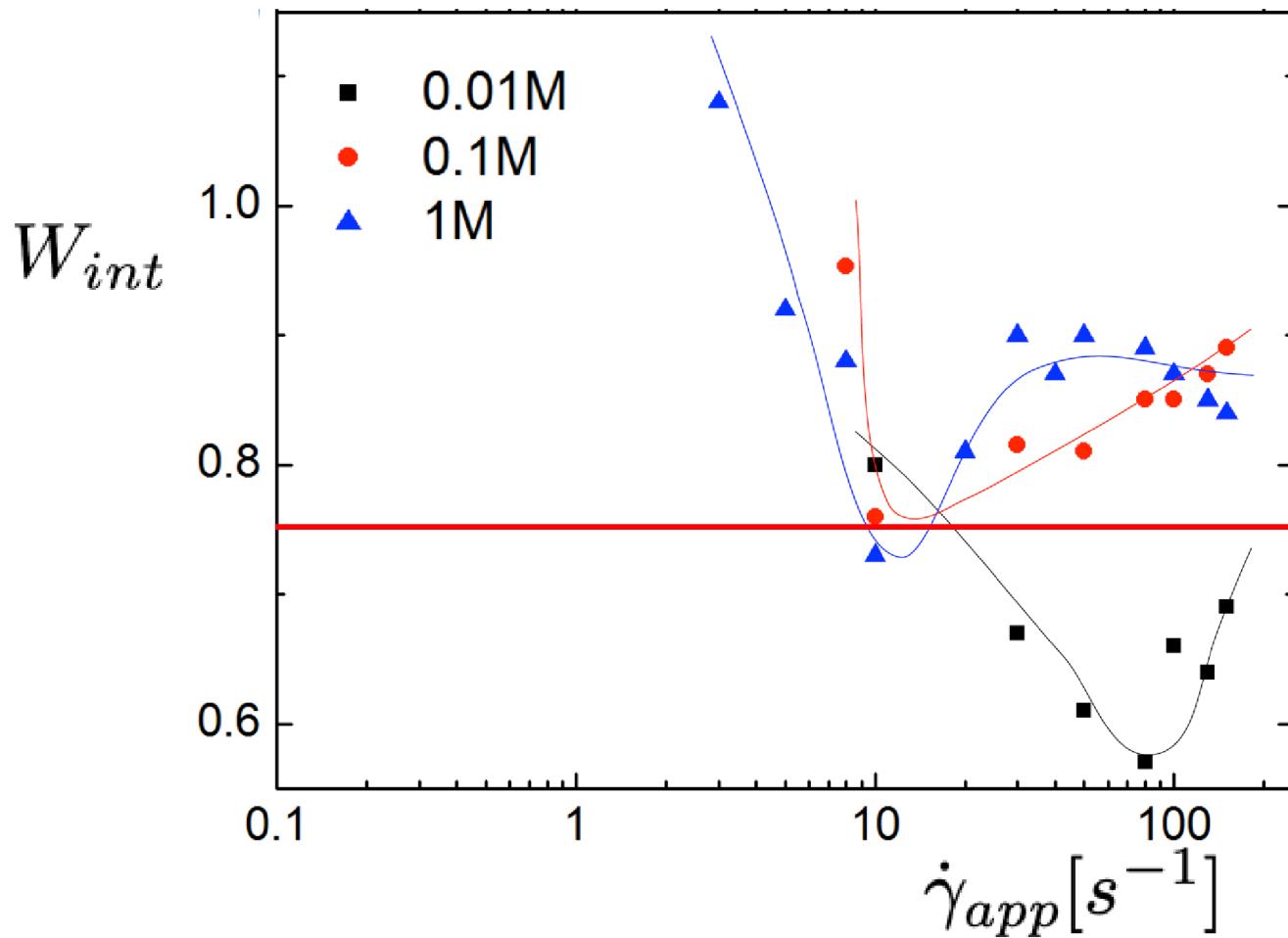


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- Bands disappear at equal thinning m_{fc}
- Birefringence disappears along with the bands

Tuning by addition of salt



- Bands disappear via widening of the interface

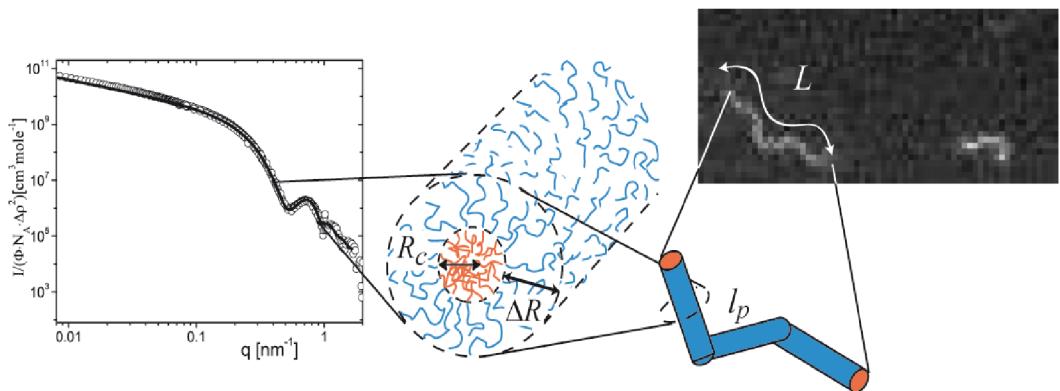
Some conclusions I

- How strong is strong?
- Always shear banding for given m , or is it system dependent? Depends on system
- Suppression shear banding via widening interface, BUT: broad shear banding can exist with broad interface
- Can we tune shear band formation? Yes, a bit

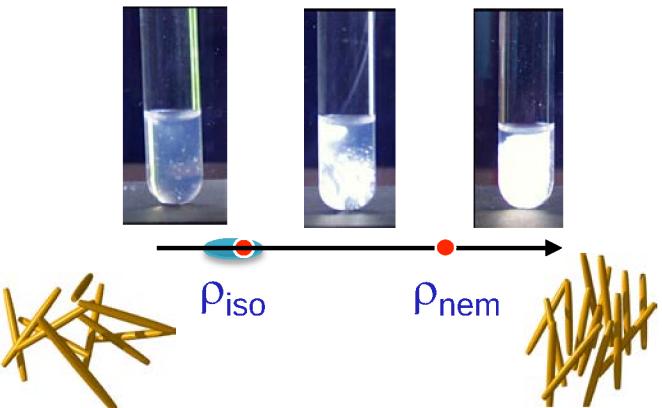
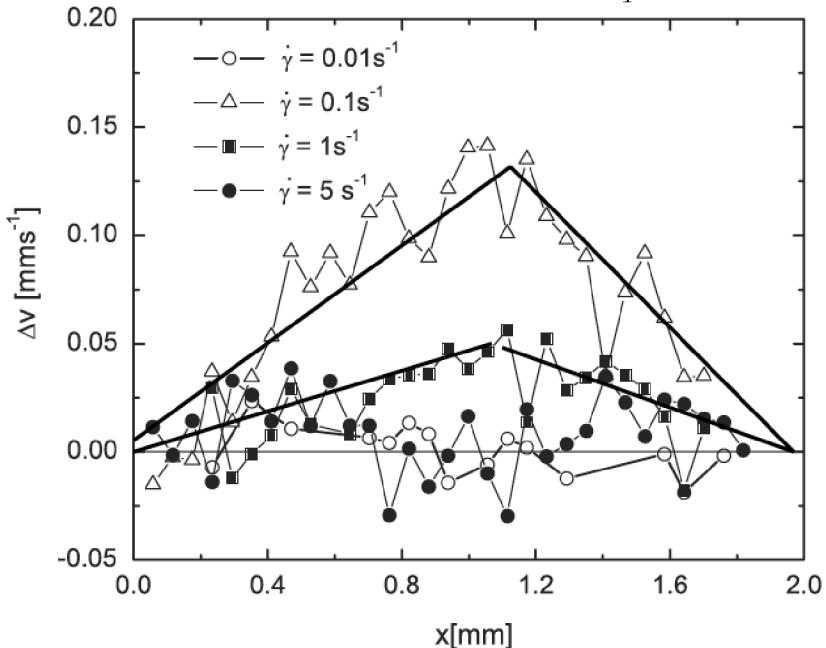
Also seen for Xanthan, with $m_{fc} = 0.21$
Tang et al, Soft Matter 2018

- New question: Is it charge or stiffness?

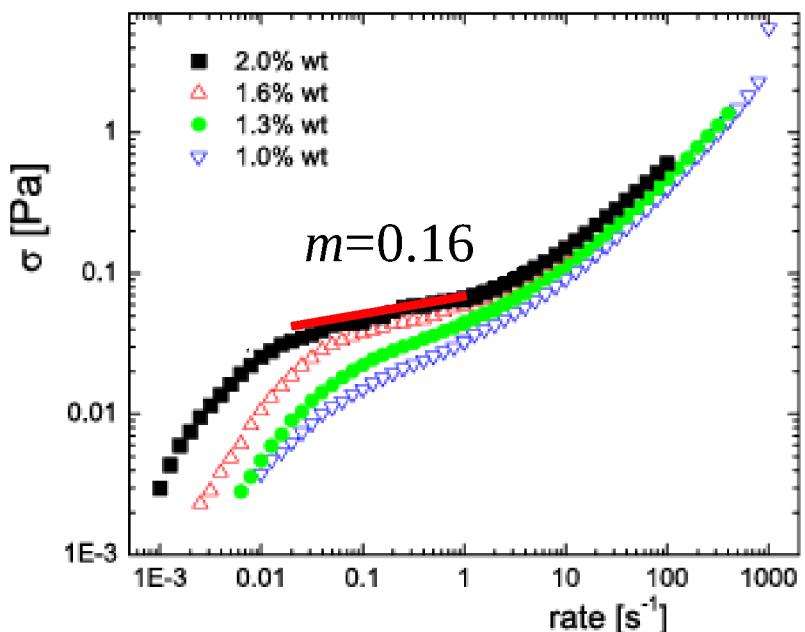
Neutral Rods close to I-N: Giant PB-PEO wormlike micelles



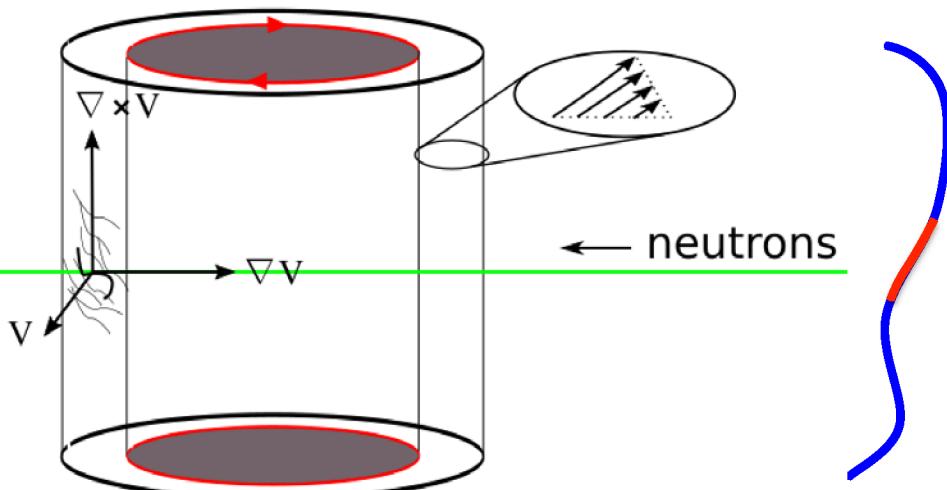
$$\langle L \rangle = 800 \text{ nm}, d = 28 \text{ nm}, l_p = 500 \text{ nm}$$



Y.-Y Won et al., *Science*, 283 1999

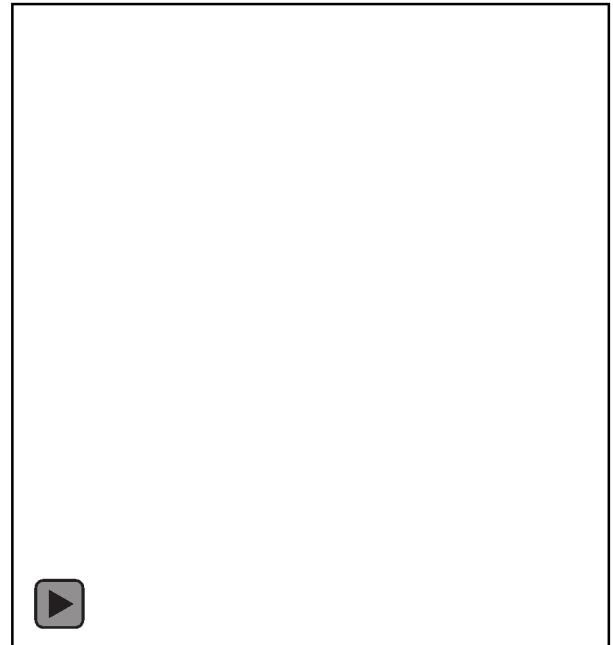


t-SANS to probe segment ordering dynamics



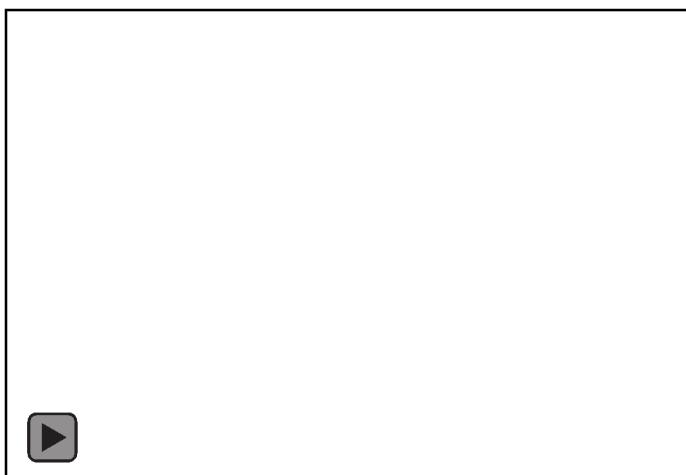
$$\langle P_2(t) \rangle = \frac{\int d\vartheta \sin(\vartheta) f(\vartheta) P_2(\vartheta))}{\int d\vartheta \sin(\vartheta) f(\vartheta)}$$

$$I(t_i, \vec{q}) = \sum_n^{N_{cycle}} I(t_i + n\Delta t, \vec{q})$$



Orientational distribution function

$$f(\theta)$$



$$\theta \text{ [rad.]}$$

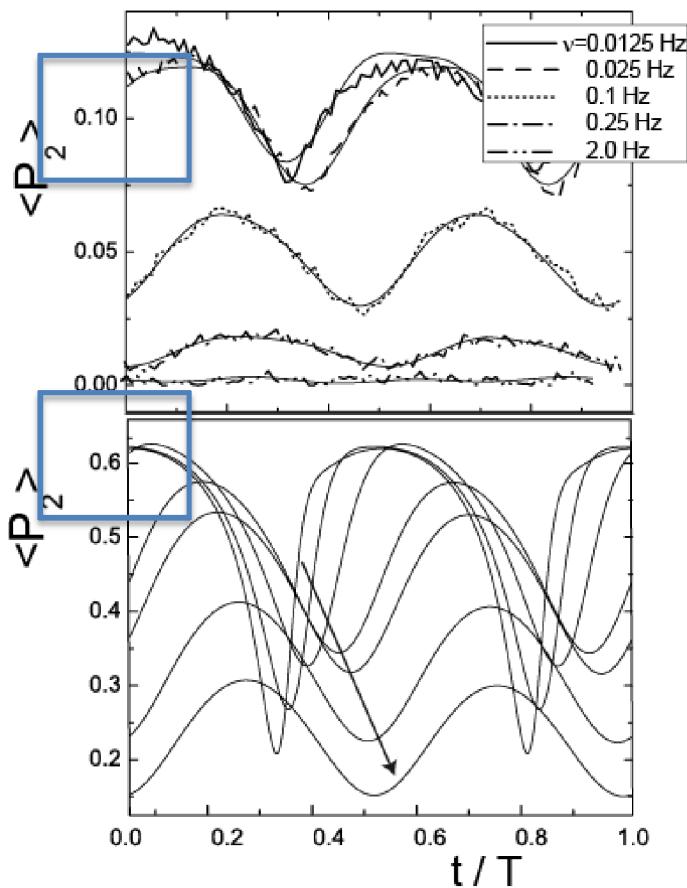
Micro

&

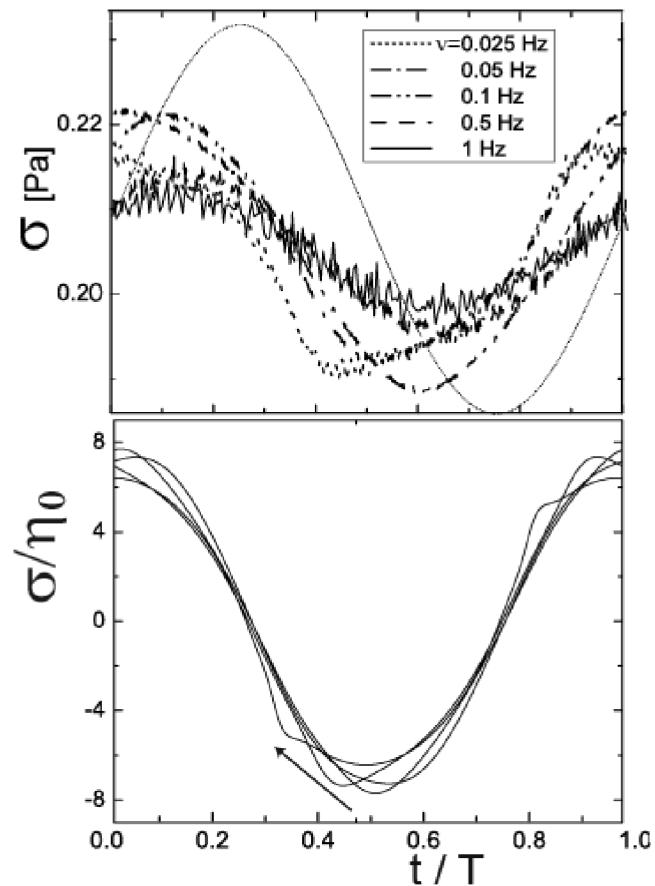
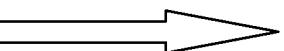
Macro



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$$\Sigma_D(\mathbf{S})$$



Scaling frequency:

$$\Omega_{eff} = \omega / D_R^{eff}$$

$$D_R^{eff} = D_R (1 - \varphi / \varphi_{IN})$$

$$D_R = 0.04 \text{ } s^{-1} \ll D_R^0$$

$$\frac{L}{d_{eff}} \varphi_{IN} = 3.0$$

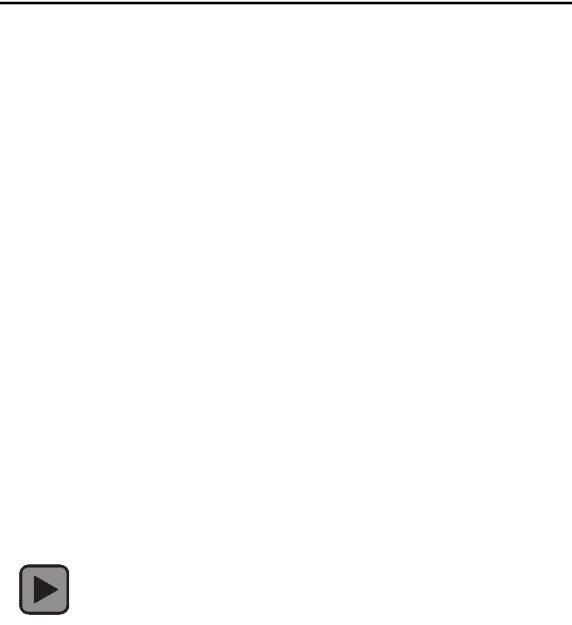
Some conclusions II

- Is it charge or stiffness? STIFFNESS

Suggestion:

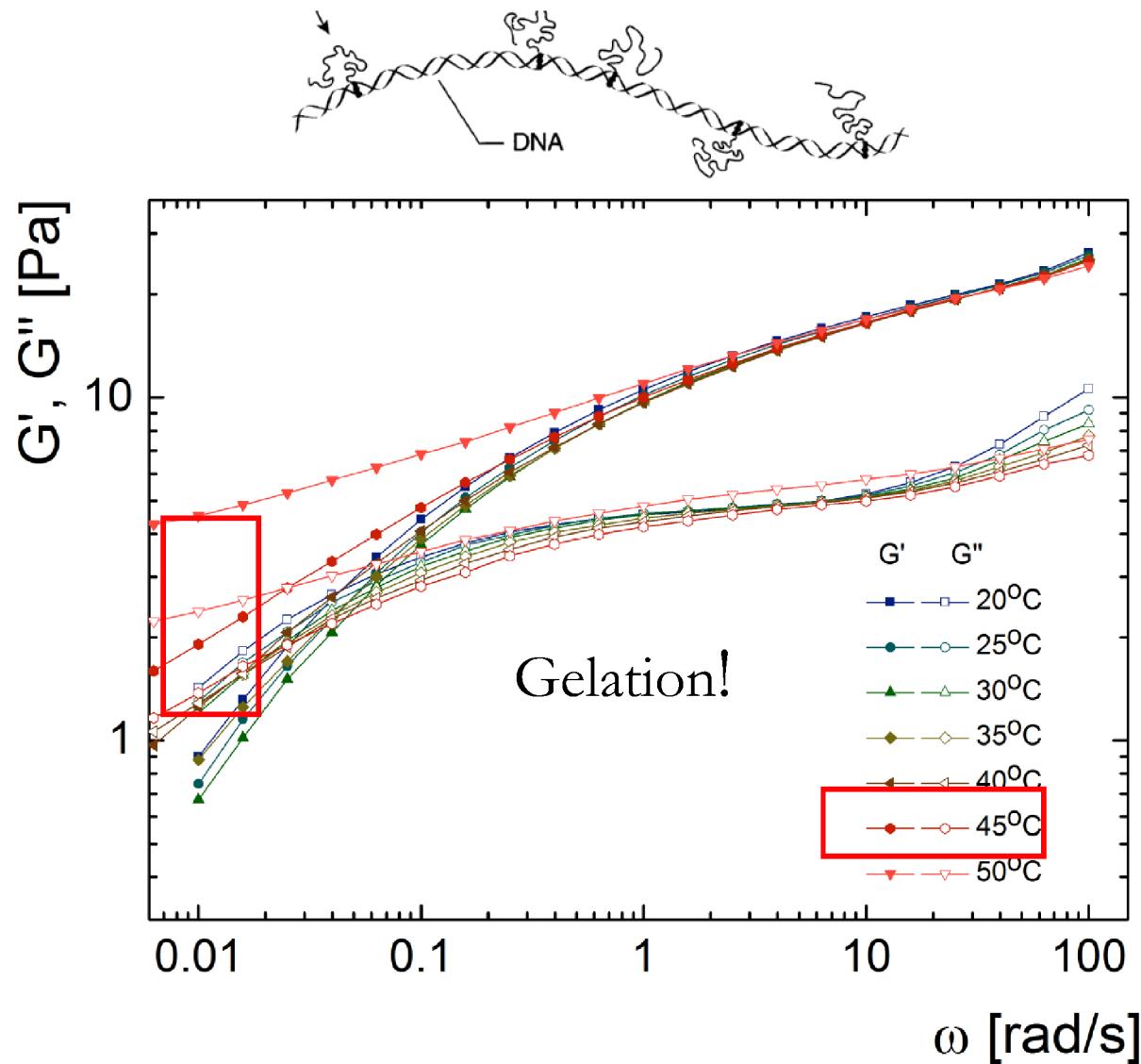
Stiff systems don't collapse once disentangled

- New question: Can we force collapse?
- New question: Can we have a better look?



DNA, the tuneable polymer part II

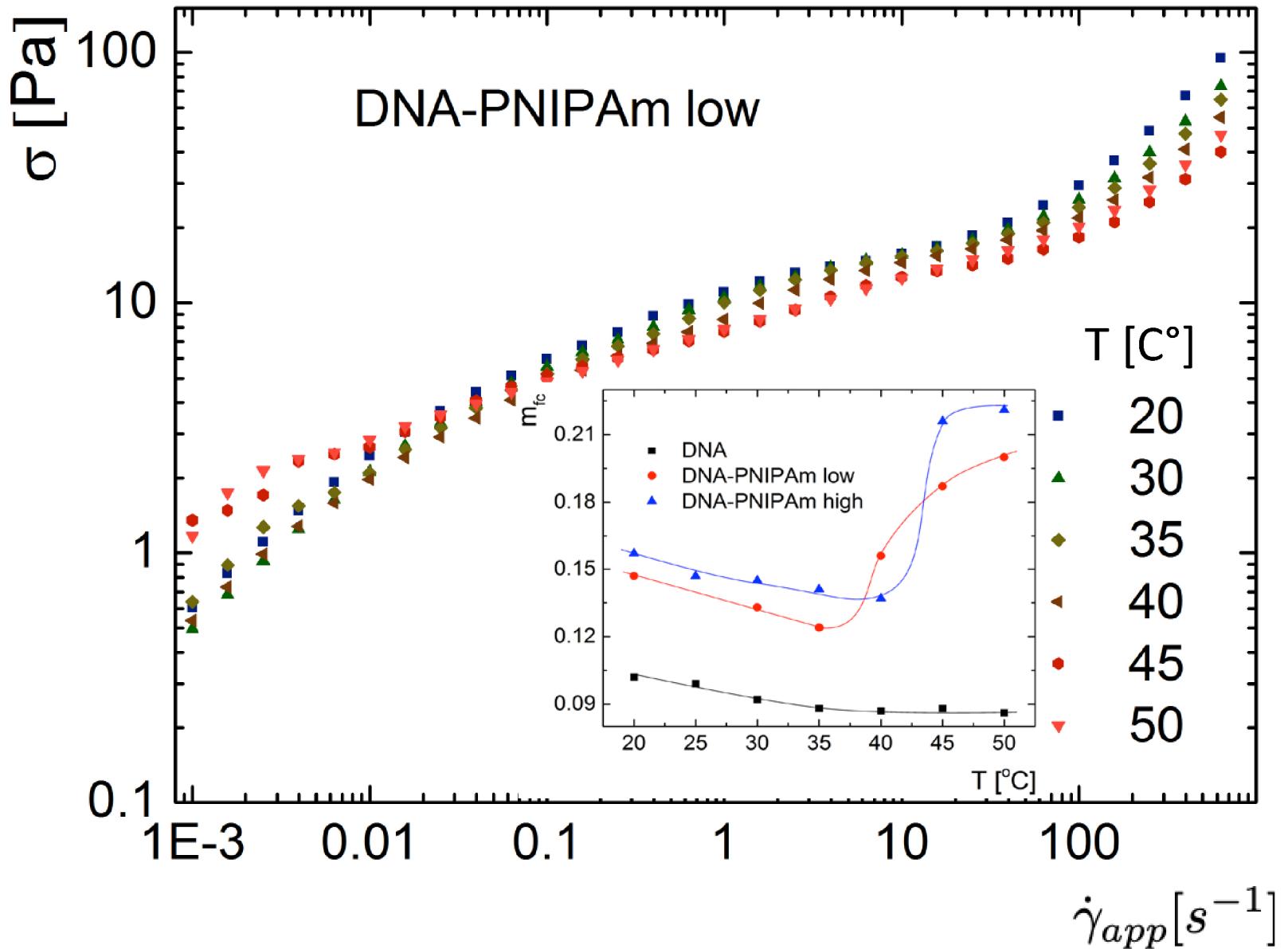
Tune attraction by adding T-sensitive brush (PNIPAm)



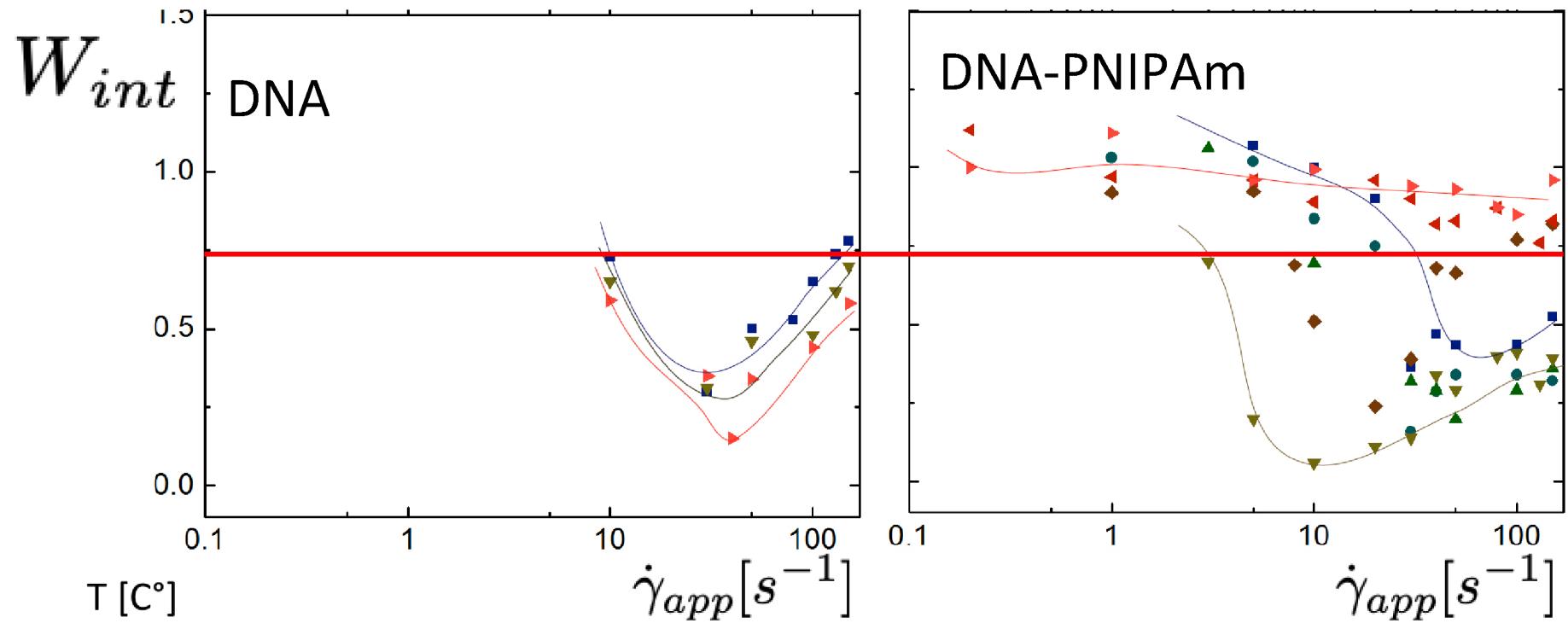
Tuning by increasing attraction



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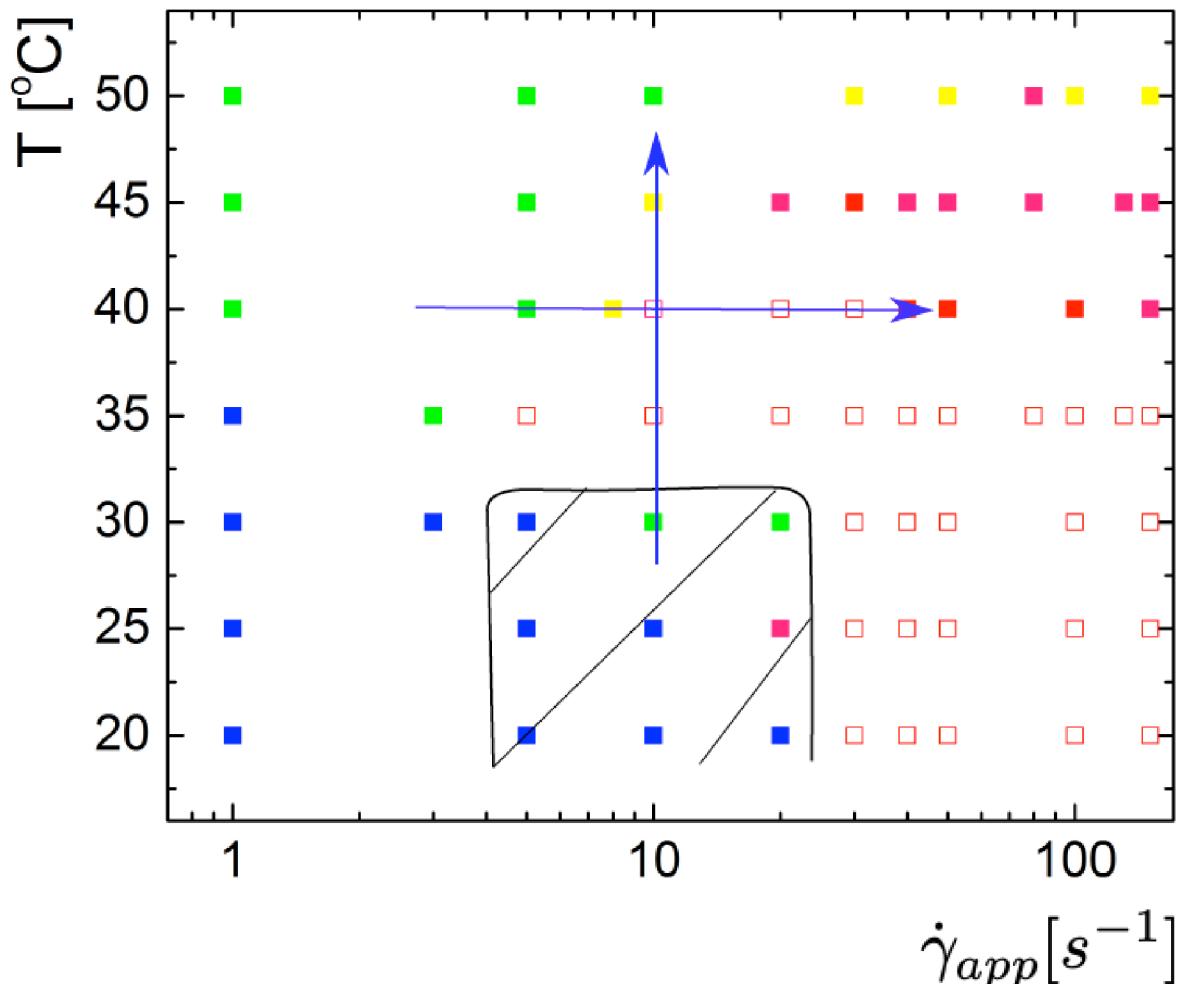


Tuning by increasing attraction



- 20
- ▲ 30
- ◆ 35
- ▼ 40
- 45
- ▼ 50

Diagram of states



- Attraction suppresses shear band formation (and orientation)
- Re-entrant behavior in two directions

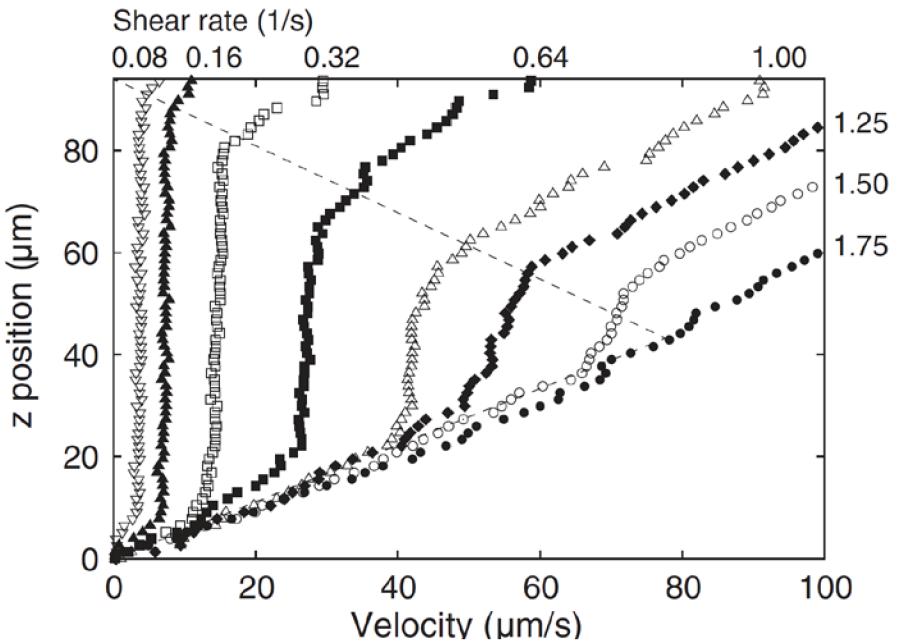
Some conclusions III

- How strong is strong?
- Always shear banding for given m ,
or is it system dependent?
Depends on system
- Is it charge or stiffness?
STIFFNESS
- Suppression shear banding via widening interface,
BUT: broad shear banding can exist with broad interface
- Can we tune shear band formation?
YES
- New question: Can we have a better look?

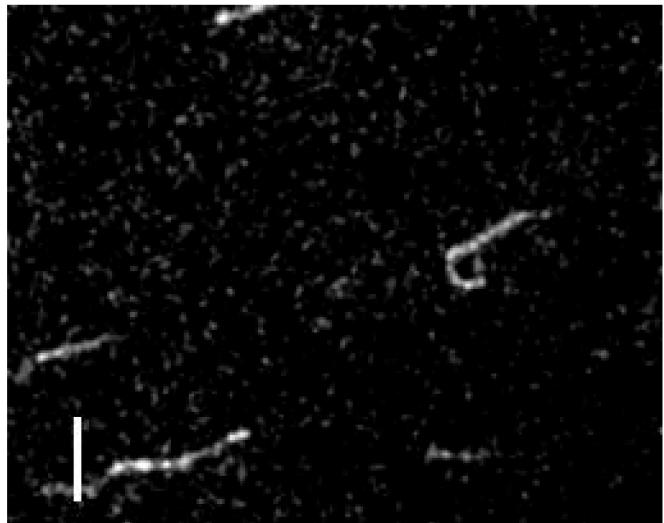
F-actin: stiffer and longer

$$\langle L \rangle \approx 20 \text{ } \mu\text{m}, d = 7 \text{ nm}, l_p = 17 \text{ } \mu\text{m}$$

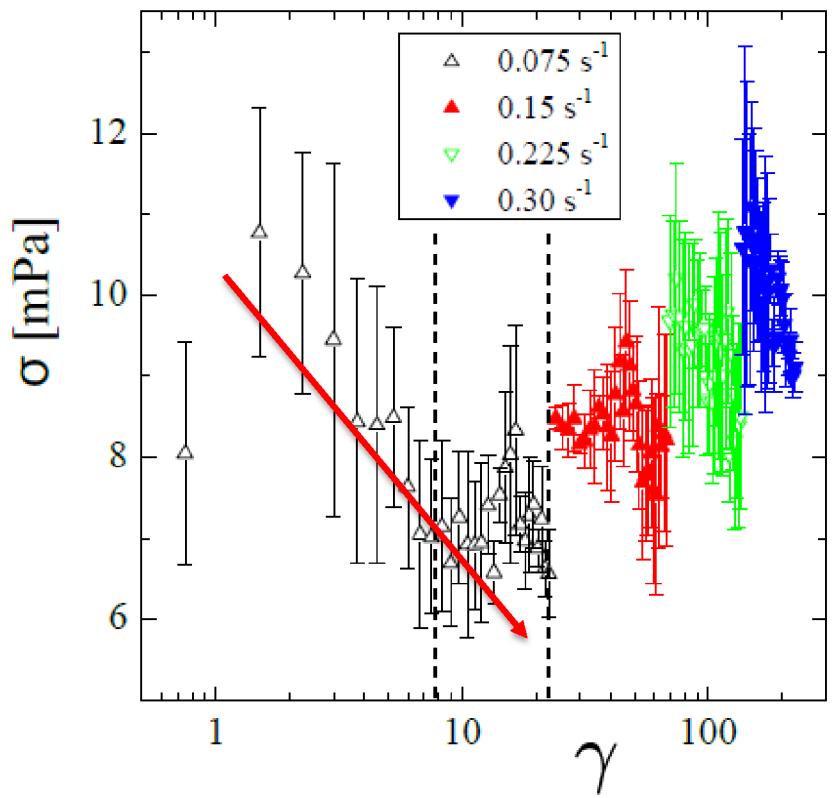
Shear banding has been identified by
 Kunita et al, PRL 109, 248303 (2012)



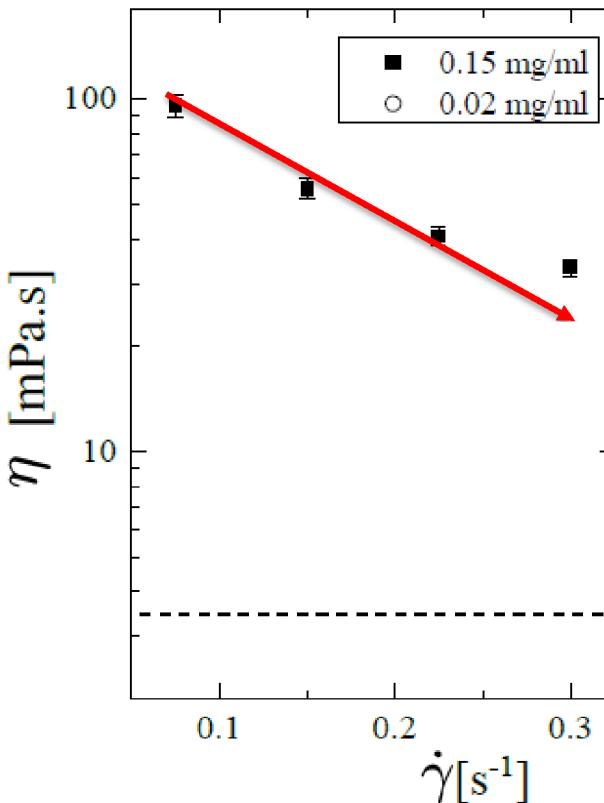
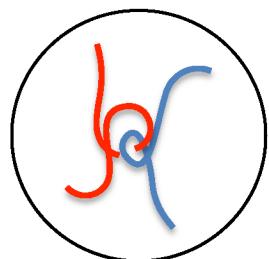
Goal: obtain 3-D structural information



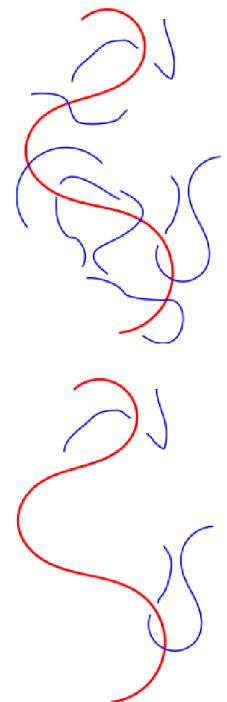
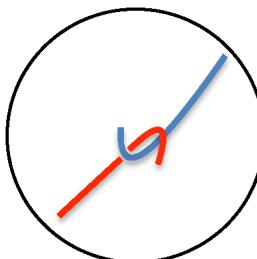
Rheological response of F-actin dispersions



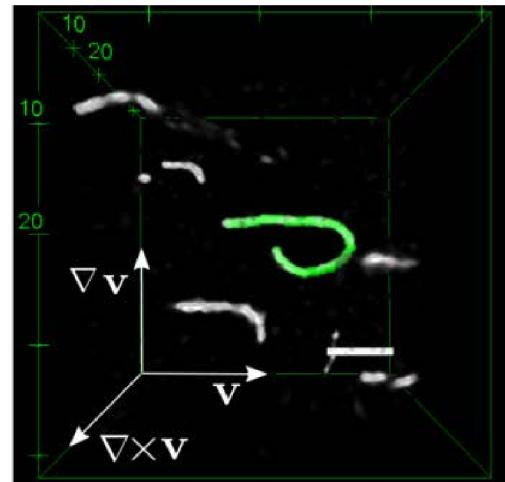
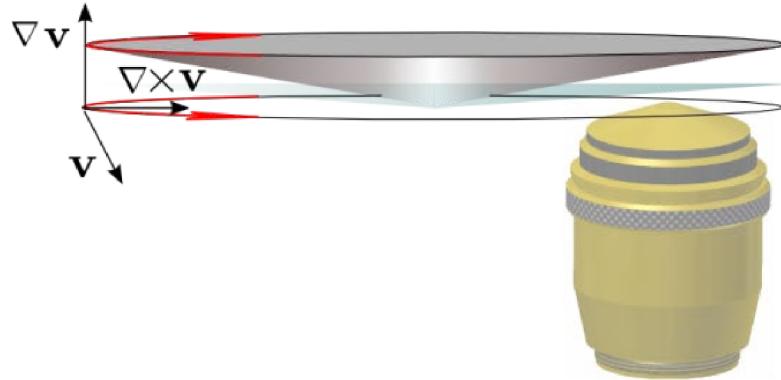
Strain softening



Shear thinning



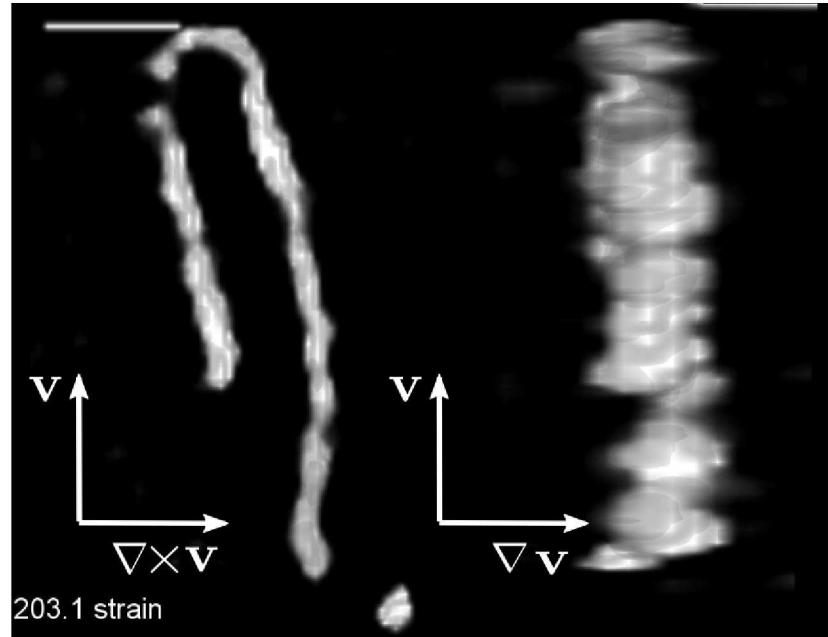
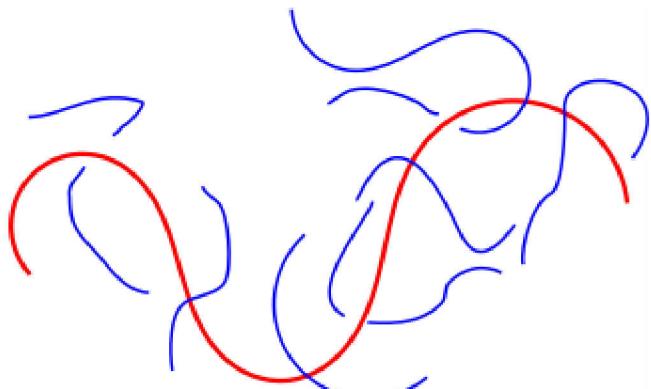
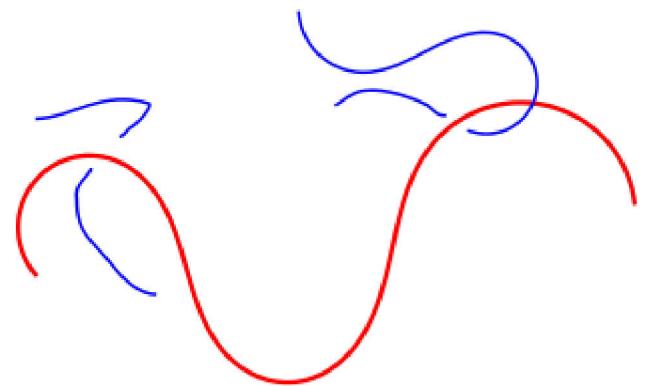
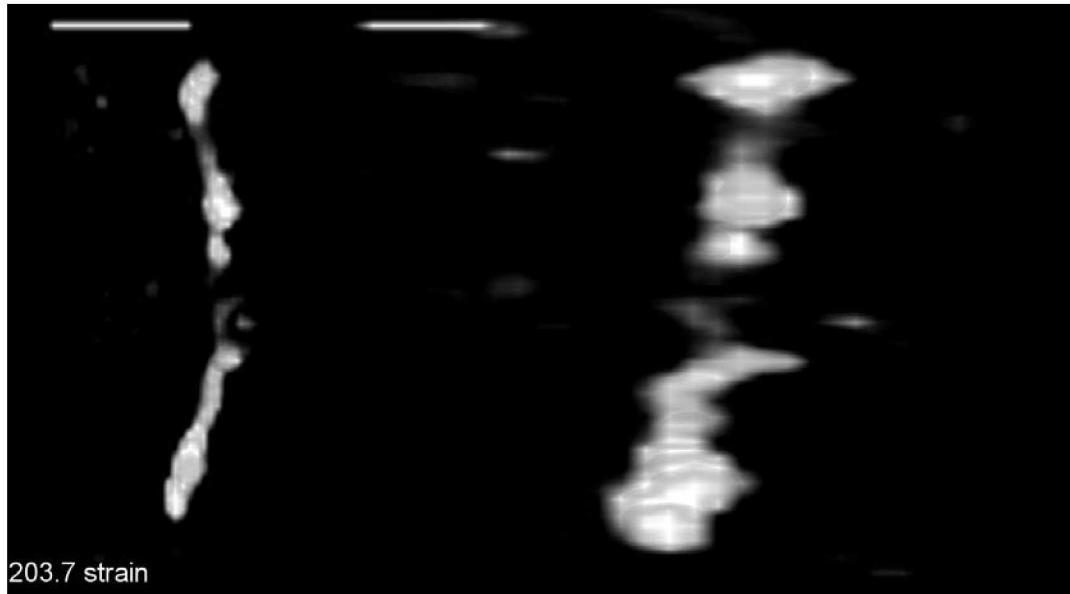
In situ confocal microscopy on entangled F-actin



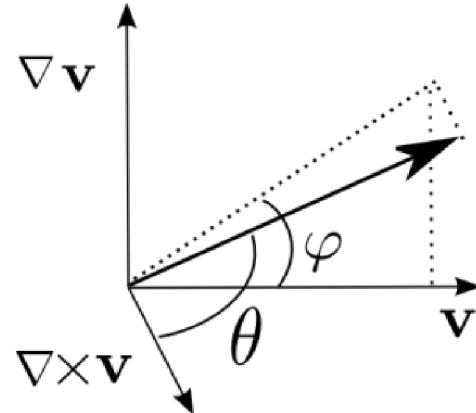
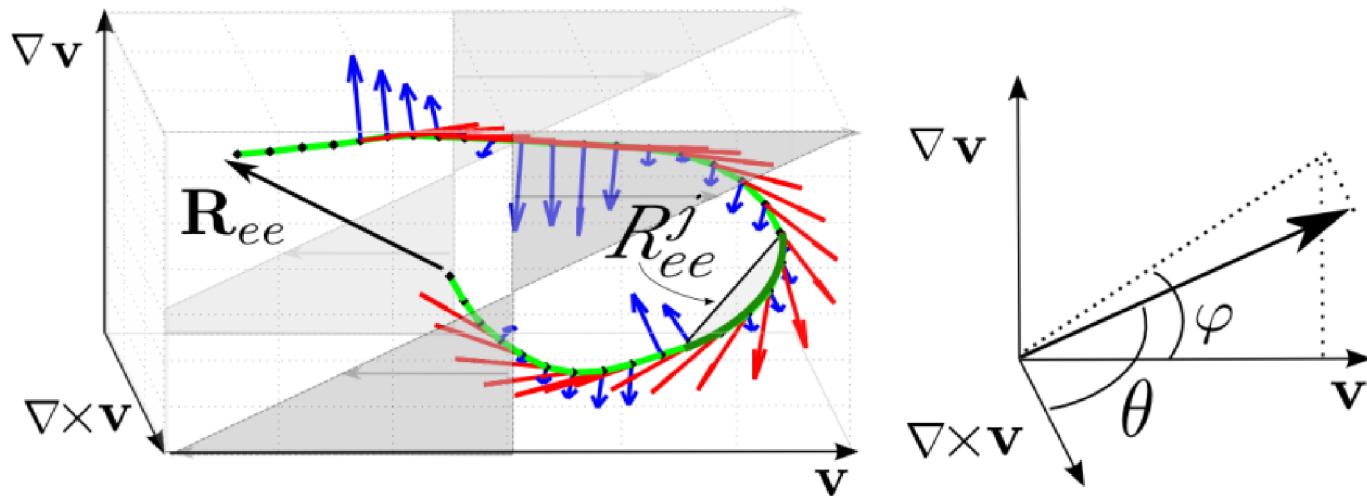
- Use three concentrations, label 1 per 100 filaments
- About 100 analyzed filaments per combination



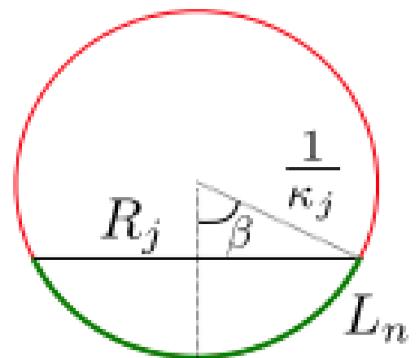
Sheared F-Actin in 3-D



Analyze local bending and stretching:

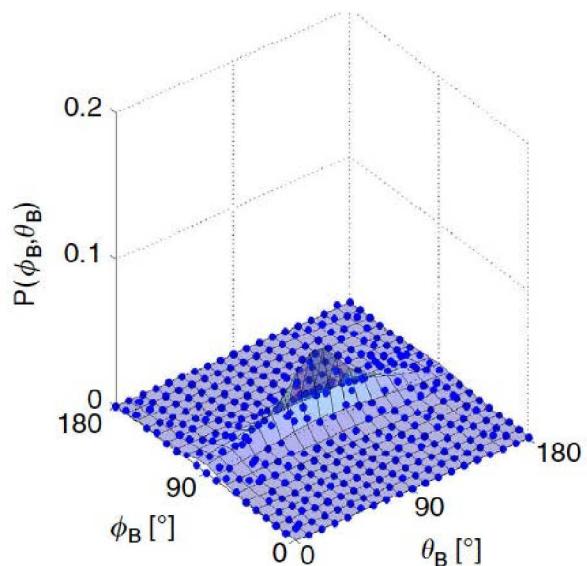
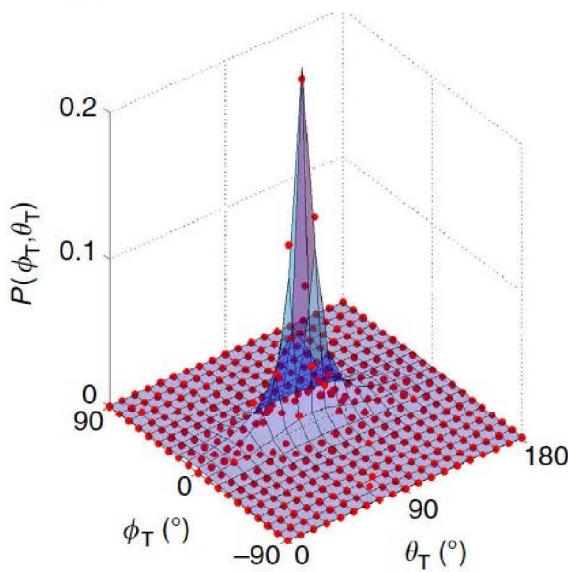
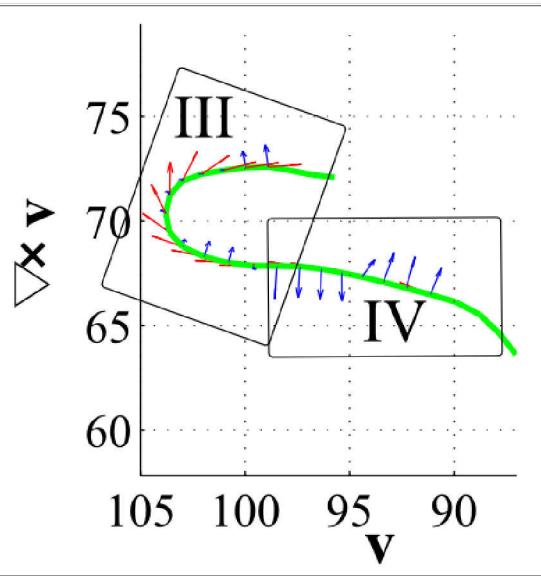


$$\hat{T}_j \equiv \frac{\dot{\mathbf{r}}_j}{|\dot{\mathbf{r}}_j|}; \hat{B}_j \equiv \frac{\dot{\mathbf{r}}_j \times \ddot{\mathbf{r}}_j}{|\dot{\mathbf{r}}_j \times \ddot{\mathbf{r}}_j|}; \kappa_j = \frac{|\dot{\mathbf{r}}_j \times \ddot{\mathbf{r}}_j|}{|\dot{\mathbf{r}}_j|^3}$$



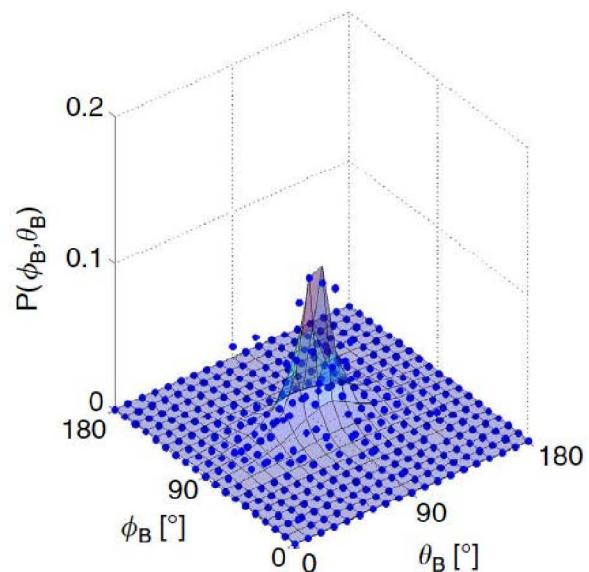
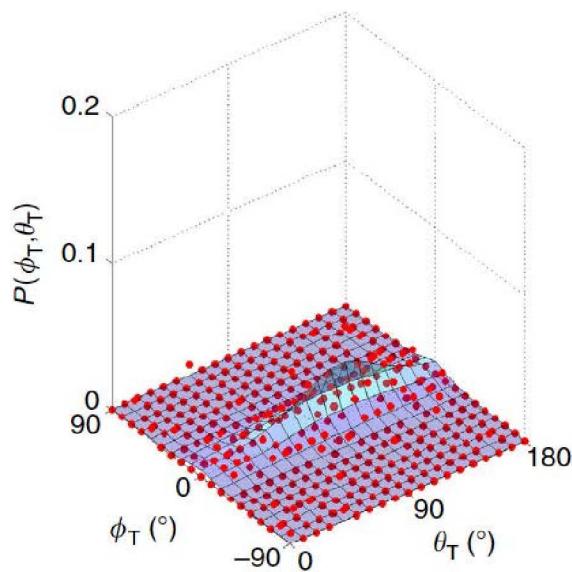
Distribution of angles

Stretched: IV



$$f(\theta, \phi) = a / \left(\left(\frac{\theta - \Delta\theta}{w_\theta} \right)^2 + \left(\frac{\phi - \Delta\phi}{w_\phi} \right)^2 + 1 \right)$$

Bent: III



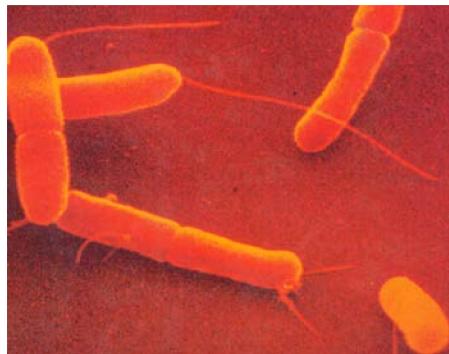
Some conclusions IV

- How strong is strong?
- Always shear banding for given m ,
or is it system dependent? Depends on system
- Can we tune shear band formation? YES
- Is it charge or stiffness? STIFFNESS
- Suppression shear banding via widening interface
- Long stiff filaments form ordered hair pins
- So what about real rods?

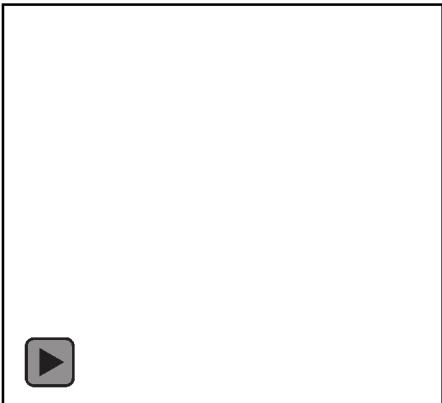
Stiff and mono-disperse rods

Materials Bio-Engineered Phage Systems (varying morphology)

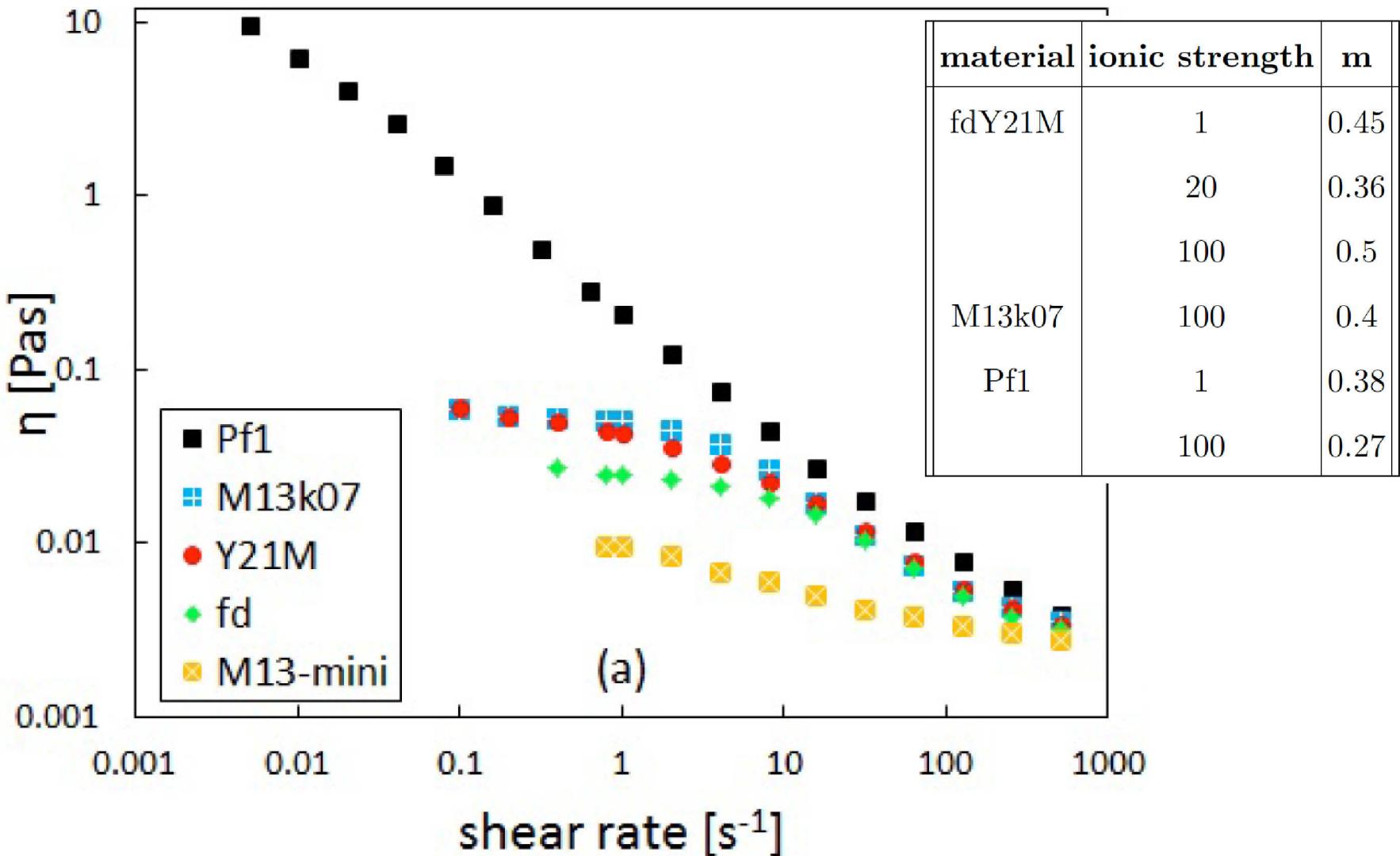
d=6.6 nm, but effective thickness depends on ionic strength



| system | L [μm] | L _p [μm] |
|--------------|--------|---------------------|
| fd wild type | 0.88 | 2.8 |
| fd Y21M | 0.91 | 9.9 |
| Pf1 | 1.96 | 2.8 |
| M13k07 | 1.2 | 2.8 |

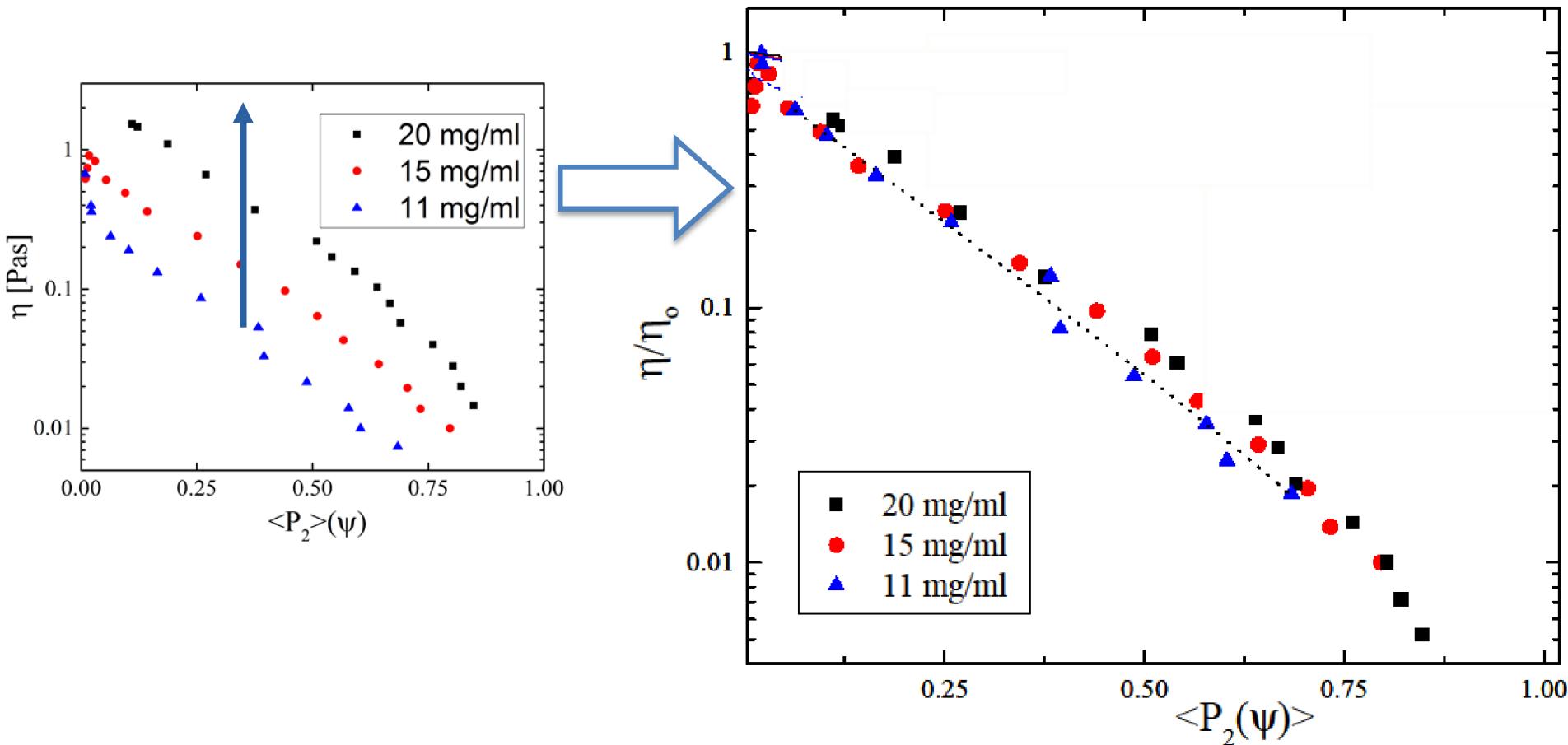


Shear thinning rods

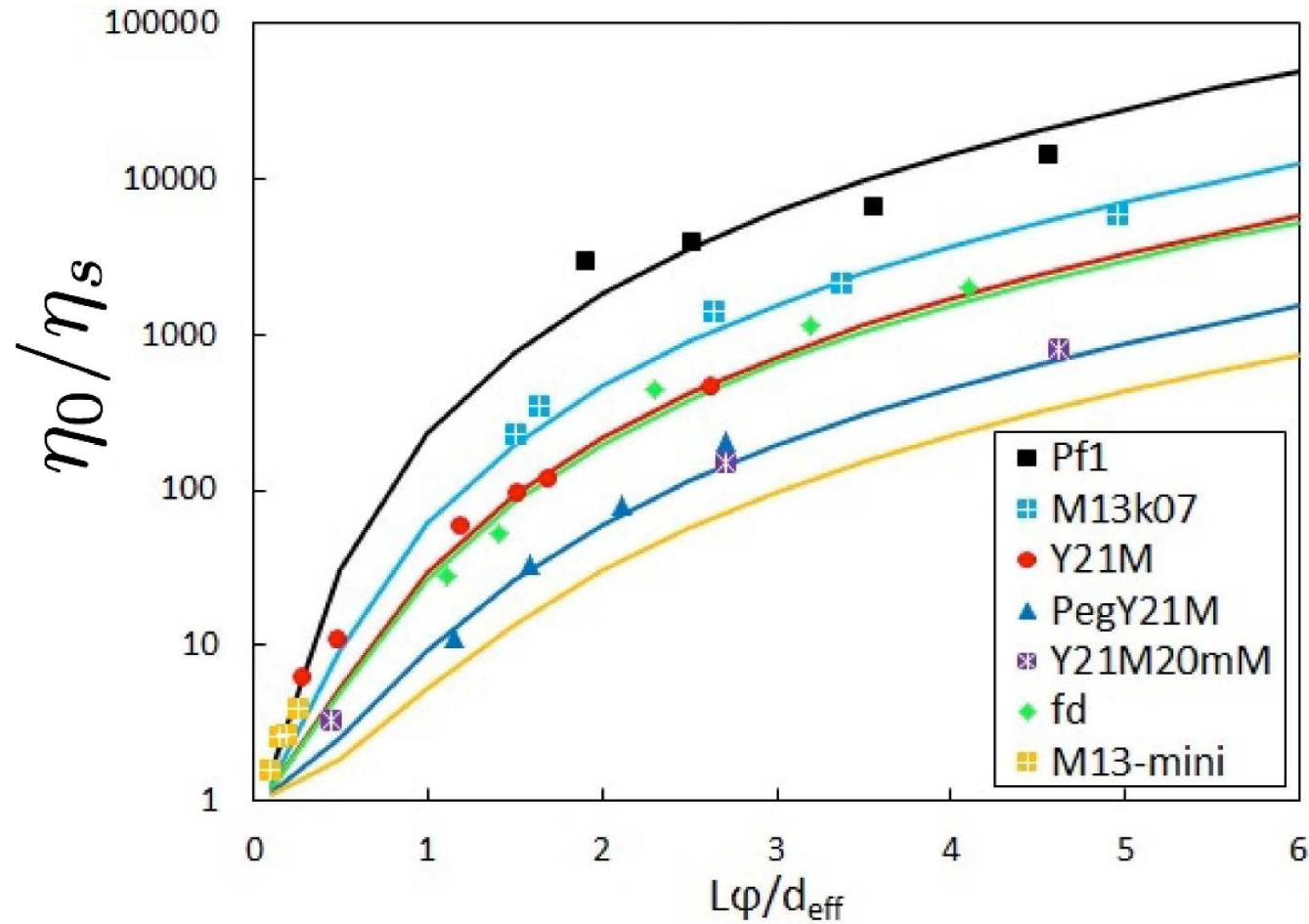


Scale viscosity

Assumption: shear thinning is caused by orientation



Zero shear viscosity of rods

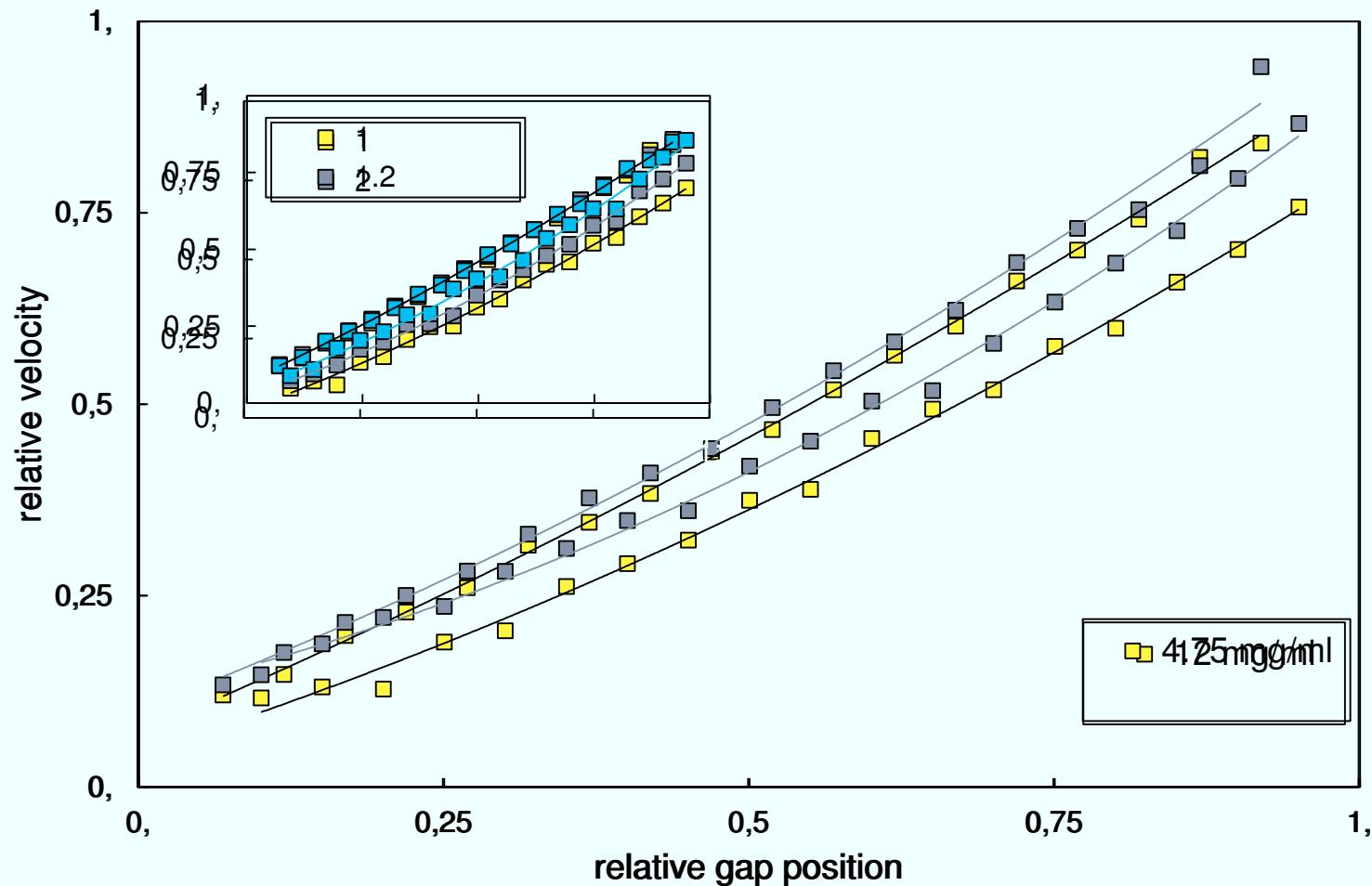


$$D_r = c D_r^0 (v L^3)^{-2}$$

$$\longrightarrow c = 3 \cdot 10^3$$

Velocity profiles of rods

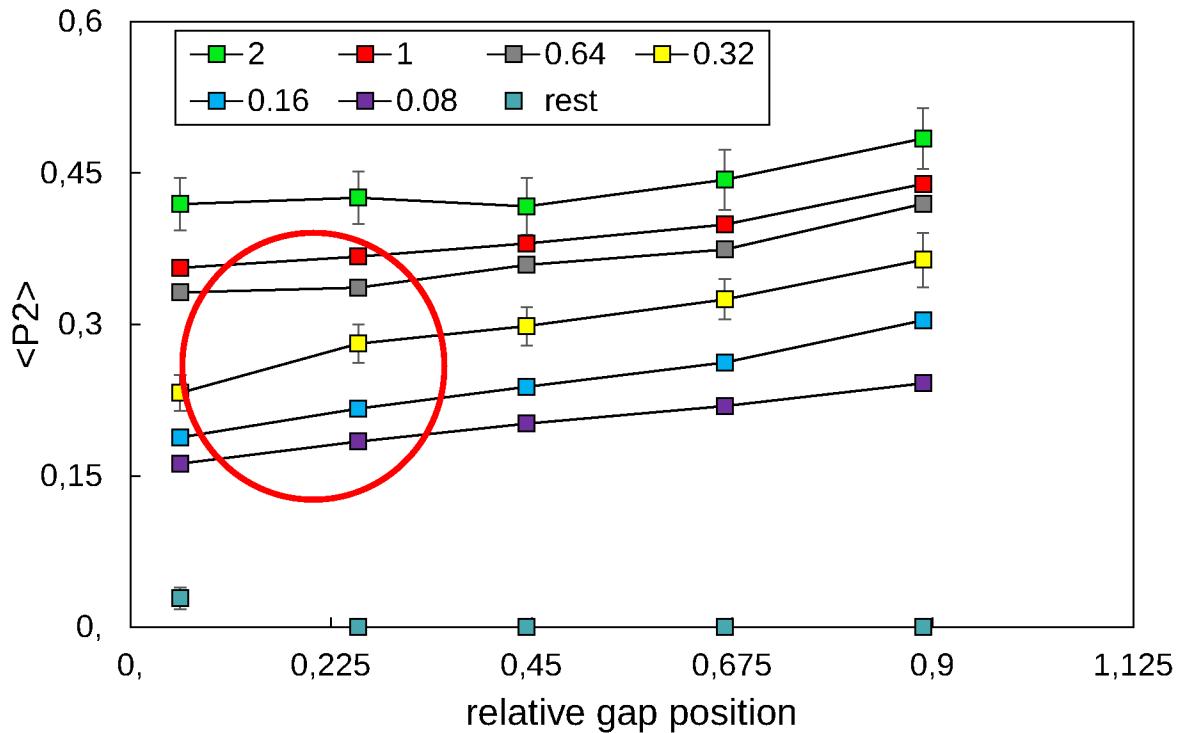
Velocity profile of M13k07 ($L=1.2 \mu\text{m}$, $L_p=2.2 \mu\text{m}$):



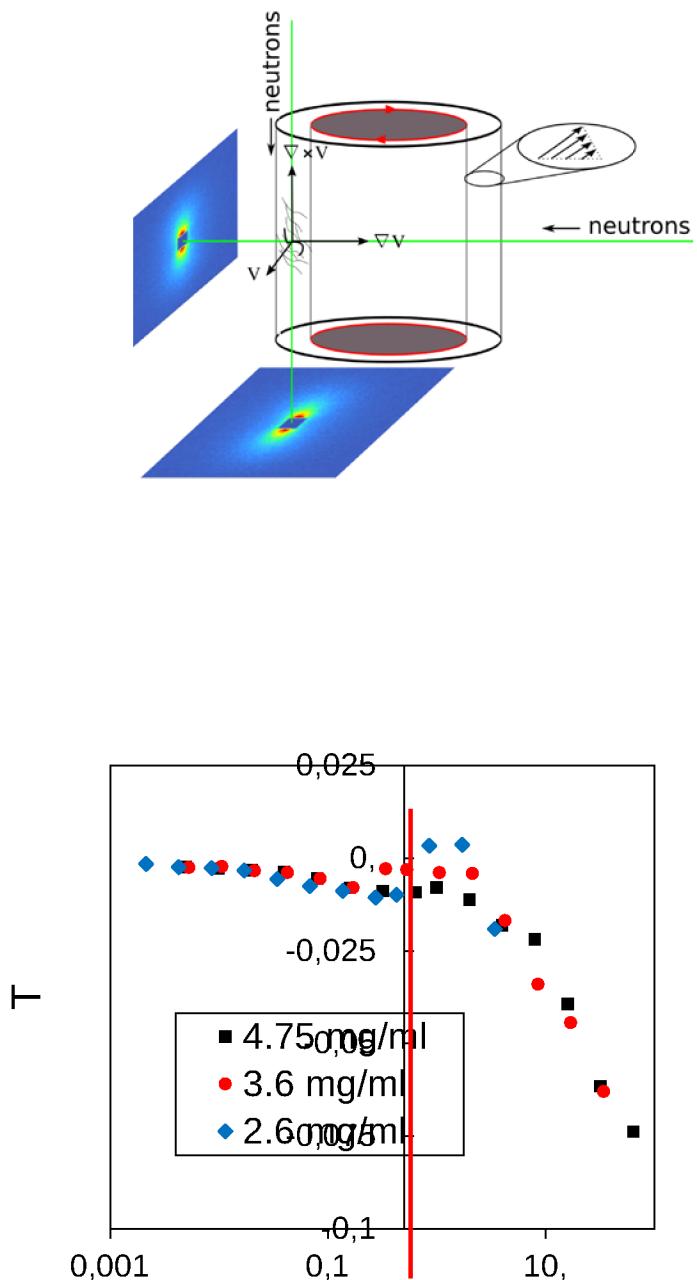
- very long and flexible rods show hints of shear banding

Zero shear viscosity of rods

Shear-banding and hairpin formation



- biaxiality reverses in a small shear rate range after “shear banding”



Final conclusion

- Always shear banding for given m ,
or is it system dependent? Depends on system
- Can we tune shear band formation? YES
- How strong is strong? $m_{fc} < 0.25$
- Suppression shear banding via widening interface,
BUT: broad shear banding can exist with broad interface

Suggestions:

- Shear banding is suppressed when chain collapses after disentanglement
- Collapse affects the shear-curvature viscosity
- Shear banding is suppressed when system is not long enough

Acknowledgements

FZ Jülich:

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Lutz Willner

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PSI, Villigen:

Joachim Kohlbrecher

ILL, Grenoble:

Lionel Porcar

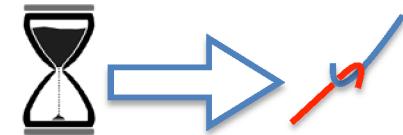
Amolf Amsterdam:

Gijsje Koenderink



Conclusions and Outlook

We find the connection between ordering and stress for *semi-flexible polymers* to *stiff rods*:



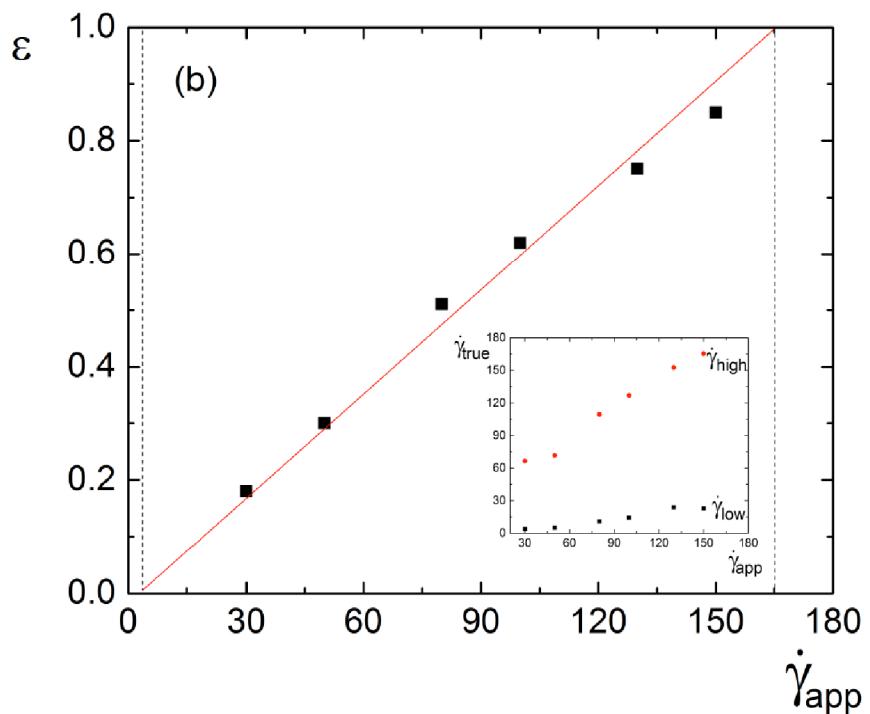
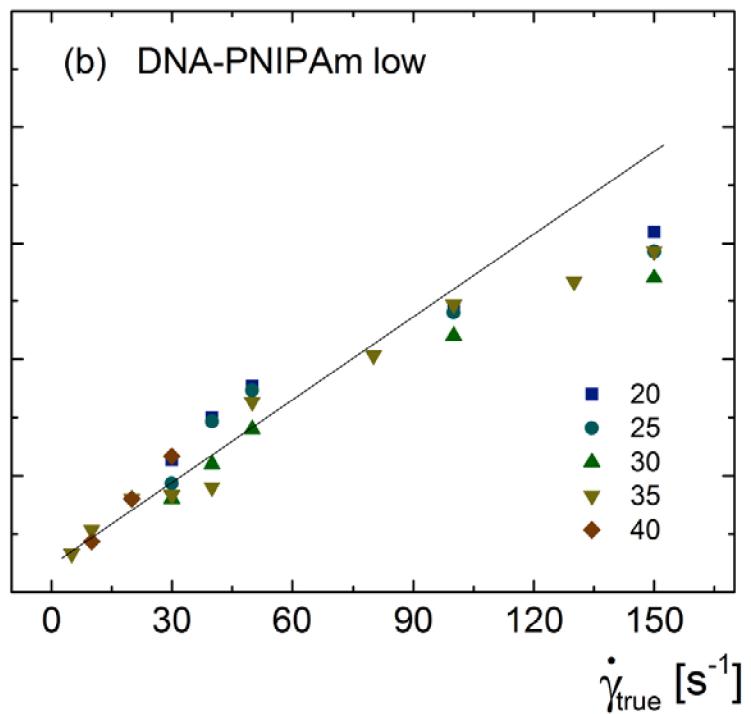
- Shear induced biaxial alignment of stiff segments

But:

- big flaws in theory for sheared rods, no non-linear theory for sheared semi-flexible polymers
- no good handle on set flow instability

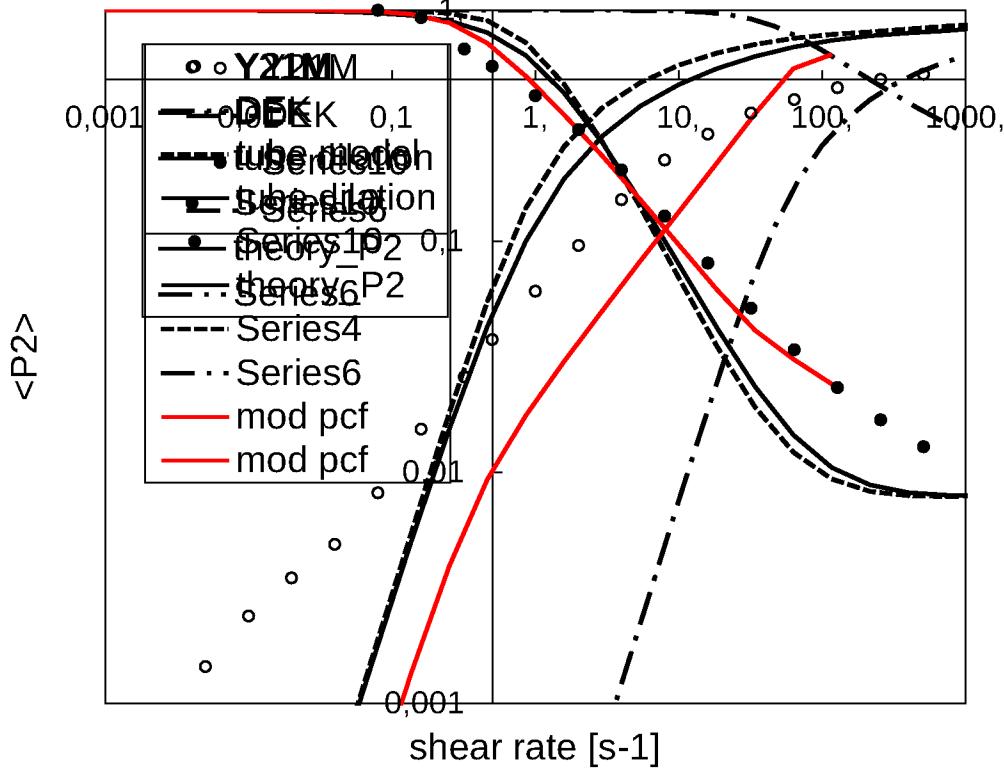
So:

- Improve theory
- Develop new systems:
 - Controlled polydispersity
 - Controlled friction with grafted DNA
 - Use labeled living stiff supra-molecular polymers



Advancements: pcf

Nonlinear viscosity and ordering of the ideal rod

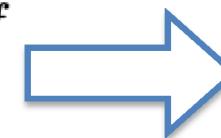




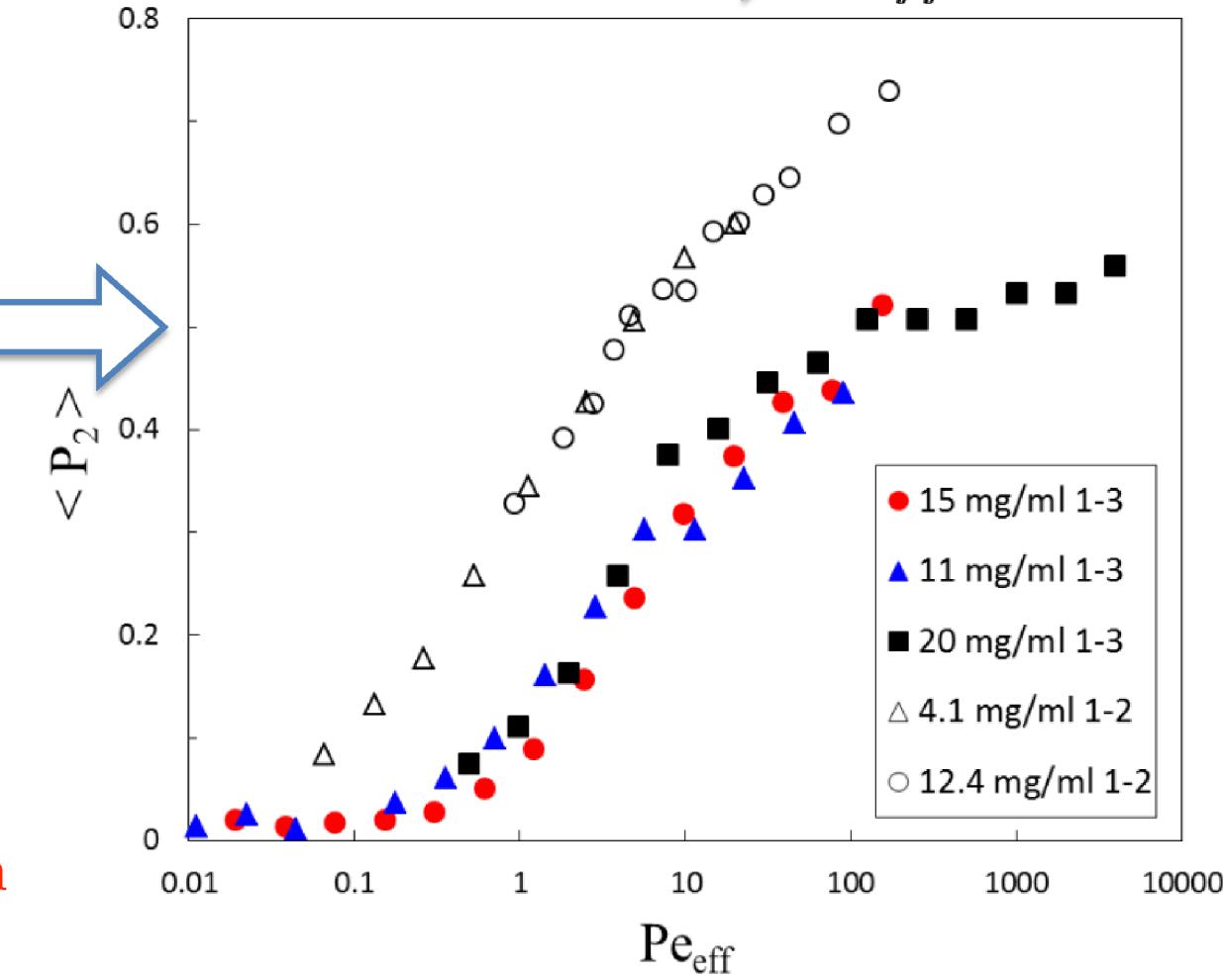
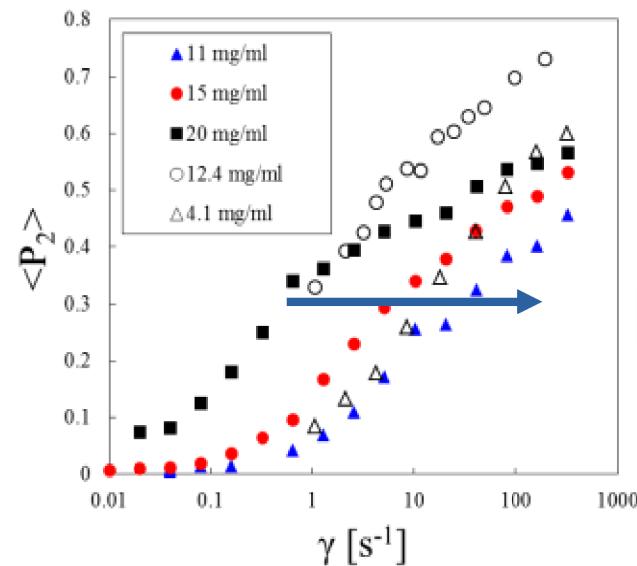
=?

Obtain the I-N spinodal point

Scale shear rate: $Pe_{eff} = \dot{\gamma}_0 / D_R^{eff}$



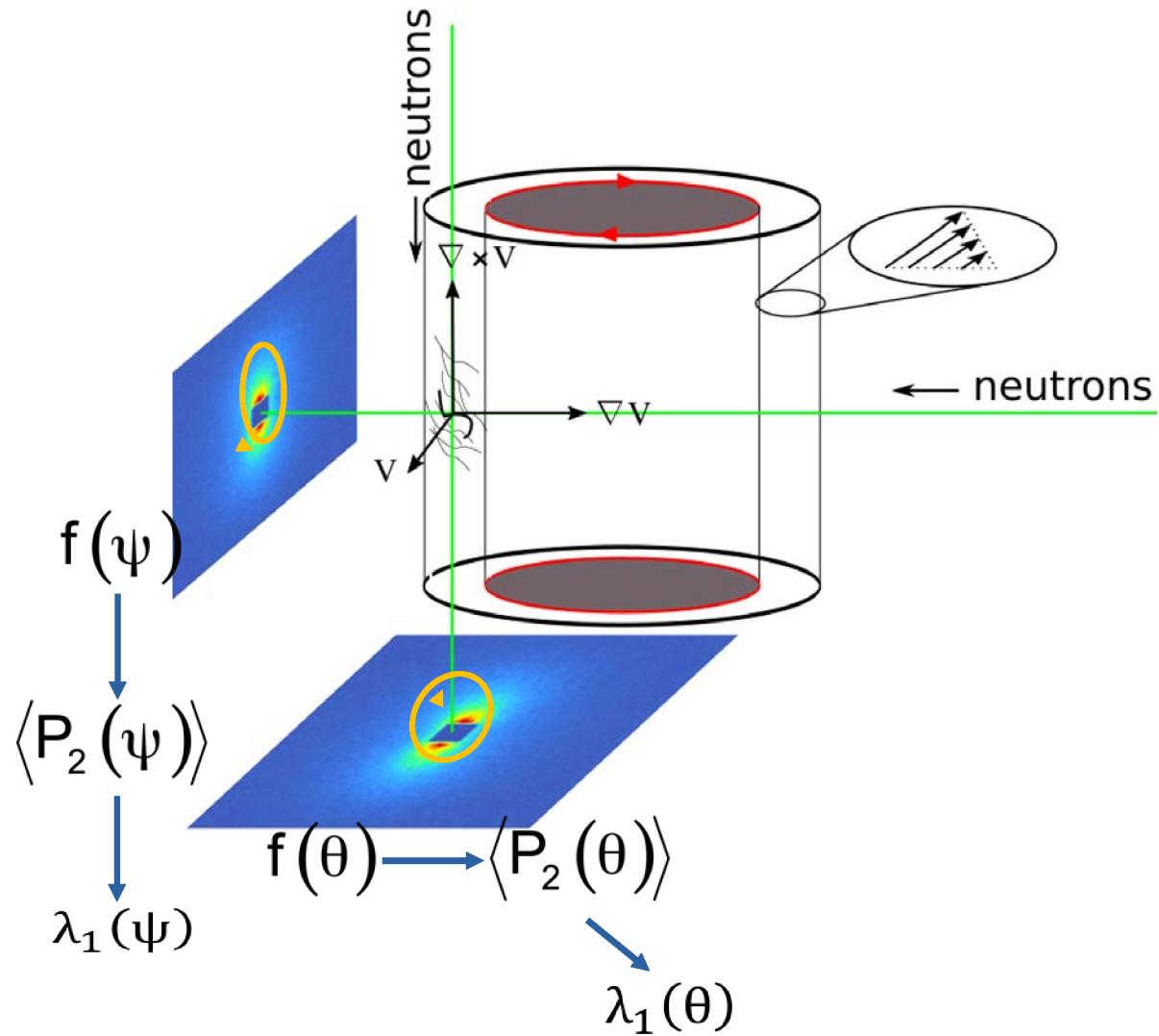
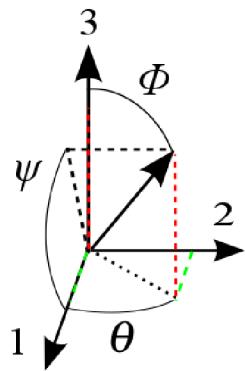
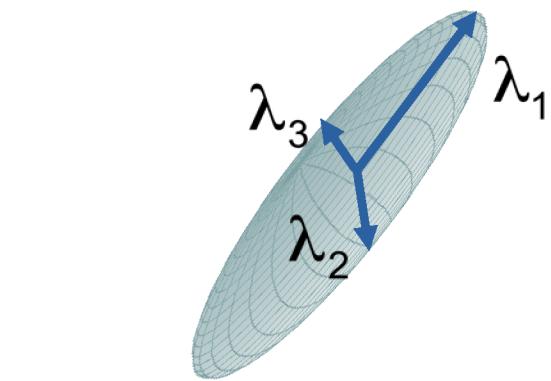
$$\frac{L}{d_{eff}} \varphi_{IN} = 4.2$$



- Collective scaling works
- different ordering in different directions:
Biaxiality!

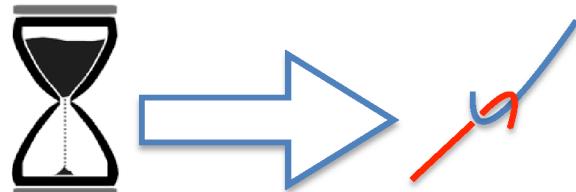
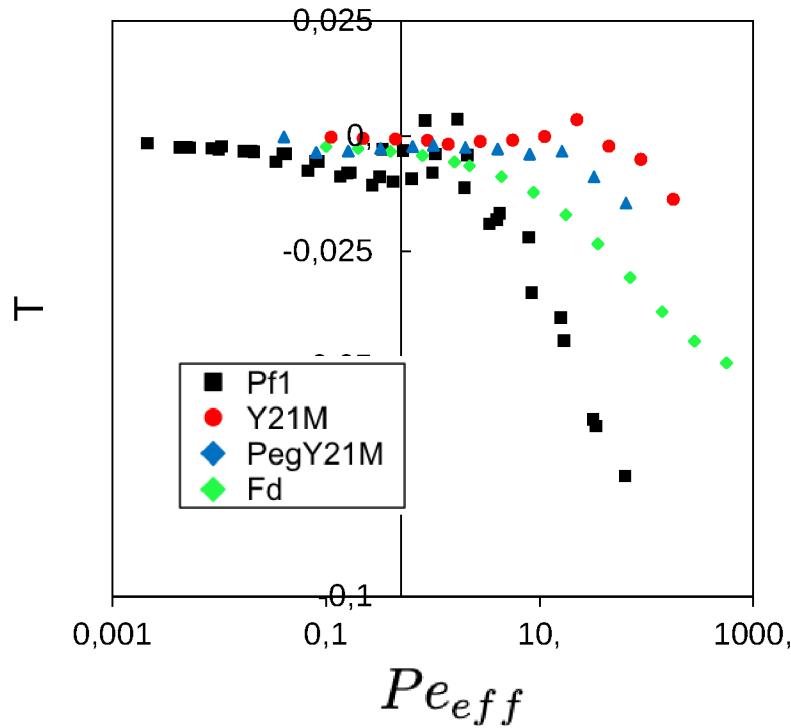
Scattering in 3-D of actual rods

3-D SANS



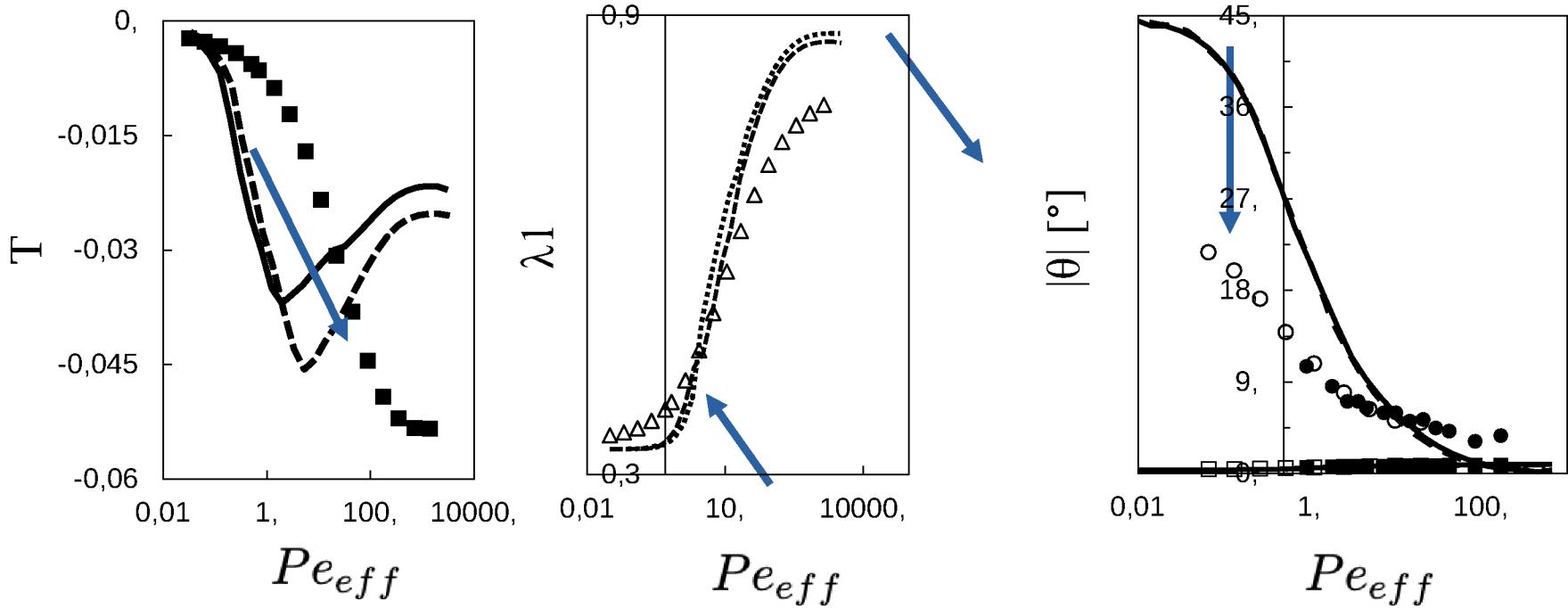
$$\lambda_1(\psi) \Rightarrow \psi_{\max} = 0$$

Biaxiality



- Collective scaling works
- But no good handle on topological effect and biaxiality.

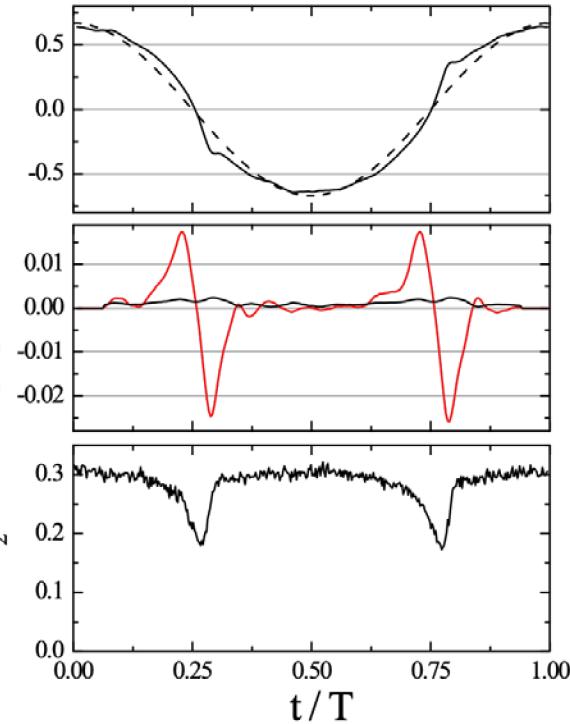
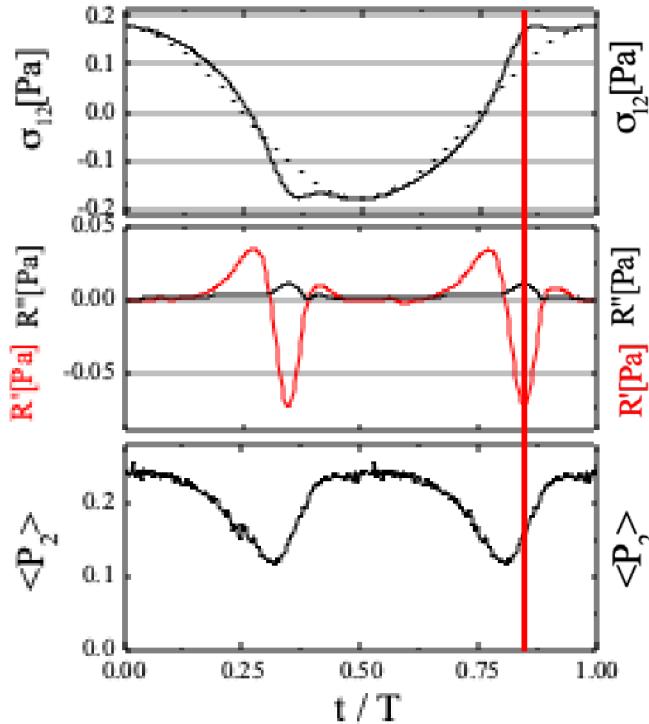
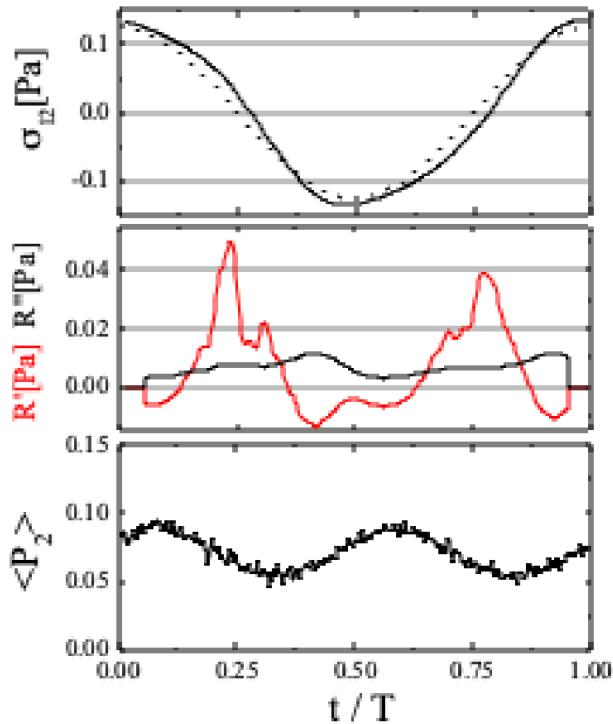
Scaling other ordering parameters



- Strong dependence at low shear rate;
weak dependence at high shear rate

Dynamic response fd virus in isotropic phase

$f = 0.01 \text{ Hz}$



$$\dot{\gamma}_0 = 3.2 \text{ s}^{-1}$$

$$\dot{\gamma}_0 = 12.8 \text{ s}^{-1}$$

$$\dot{\gamma}_0 = 102 \text{ s}^{-1}$$

Push and pull experiments

End-to-end vector \mathbf{R}_{e-e} is the relevant parameter

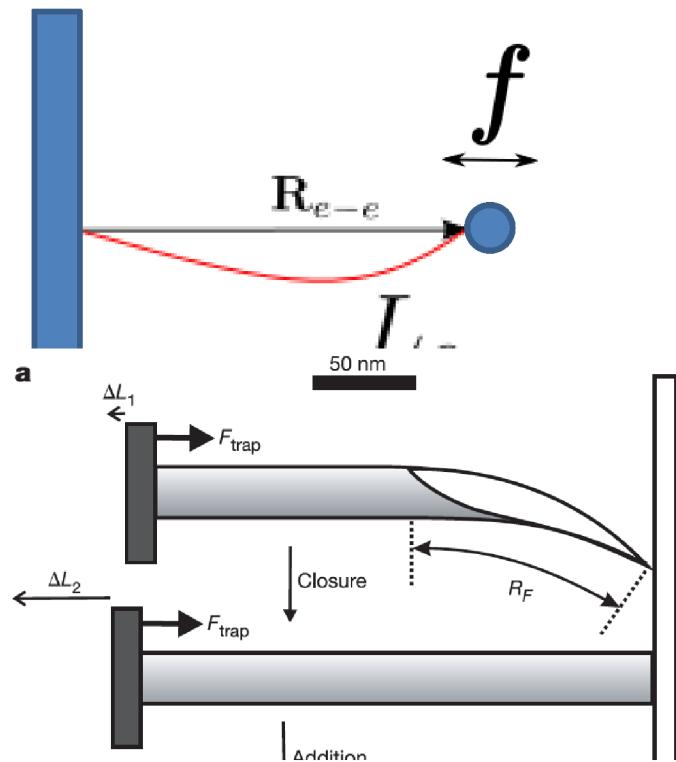
Stiff: $\mathbf{R}_{e-e} \approx L_c$

Semi-flexible:

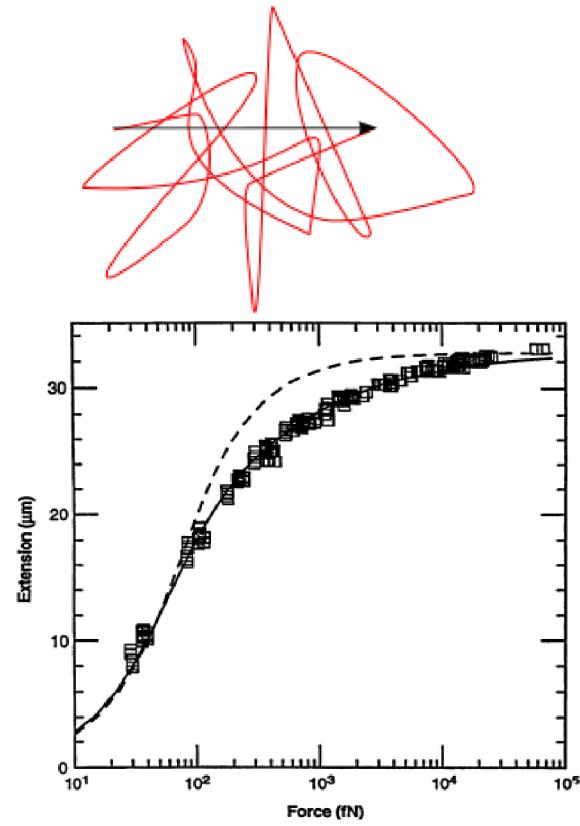
$\mathbf{R}_{e-e} < L_c$

Flexible:

$\mathbf{R}_{e-e} \ll L_c$



Assembly dynamics of microtubules at molecular resolution
Kerssemakers et al, Nature, 442 (2006)



Entropic elasticity of λ -DNA
Bustamante, Science 265(1994)

Introduction

Scaling the effective Peclet number by an apparent rotational diffusion coefficient (MPC calculation):

- effective Peclet number:
- at the I-N transition: $Pe_{eff} = \dot{\gamma}/D_r^{coll}$ (lower spinodal point)
- Smoluchowski theory: $\varphi_{IN} \Rightarrow D_r^{coll} \equiv 0$

$$\frac{\partial}{\partial t} \delta S = -6D_r^{coll} \delta S$$

$$D_r^{coll} = D_r^0 \left(\frac{L}{d} - \frac{L}{5d} \varphi \right)$$

(depending on U between 4 and 5)

- MPC:

$$D_r^{coll} = D_r^0 A \left(\frac{L}{d_{eff}} \varphi_{IN} - \frac{L}{d_{eff}} \varphi \right)^v$$

Tao et al., *J. Chem. Phys.*, 2006

- measurement:

$$\varphi_{IN} = 4.2$$

Motivation:

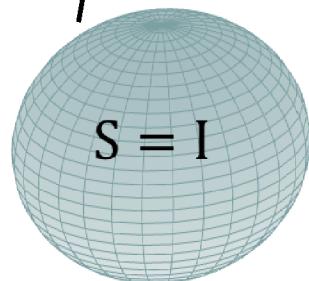
rod suspensions show strong shear thinning: microscopic reason

no shear

isotropic pdf

Brownian motion:
rotational diffusion

$$D_r \neq D_r^0$$



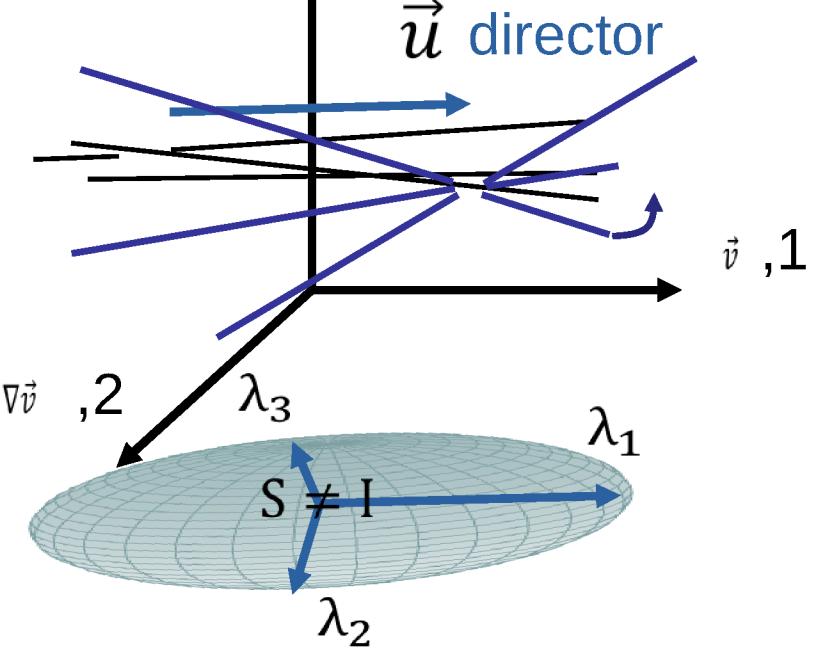
ordering tensor

high viscosity

shear
anisotropic pdf

$$\vec{\nabla} \times \vec{v}, 3$$

$$D_r = D_r^0$$



much lower viscosity



Danach Brühe

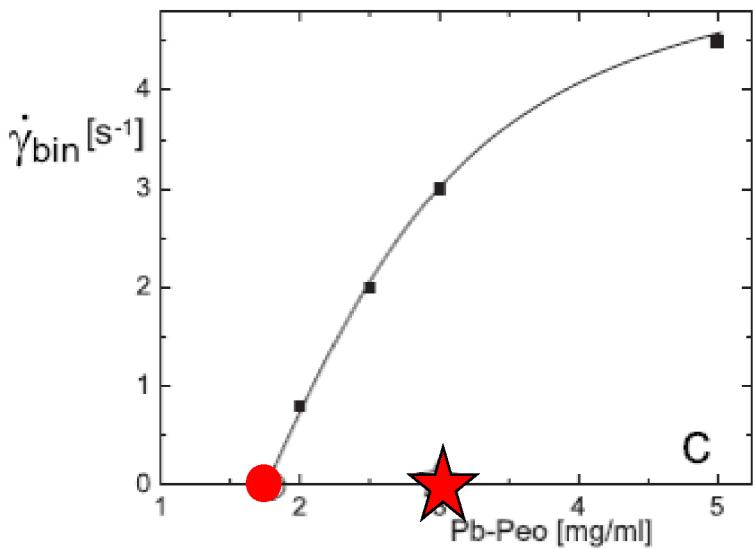
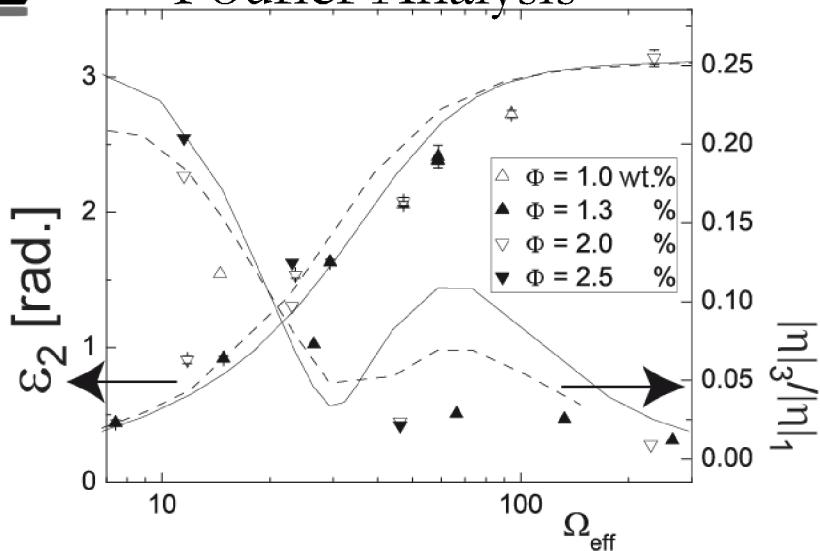


=?



KU LEUVEN

Fourier Analysis



$$\Sigma_D = 2 \dot{\gamma}_0 \hat{E} \sum_{n=0}^{\infty} |\eta|_n \sin(n\omega t + \delta_n)$$

$$P_2(t) = \sum_n |P_2|_n \cos(\omega t + \epsilon_n)$$

Scaling frequency:

$$\Omega_{\text{eff}} = \omega / D_R^{\text{eff}}$$

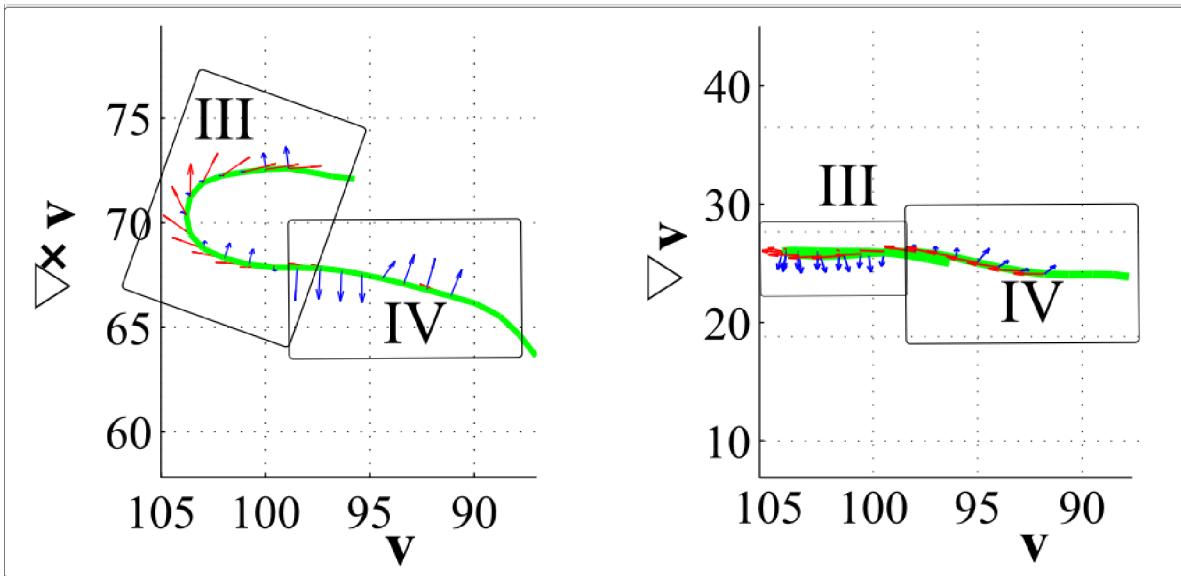
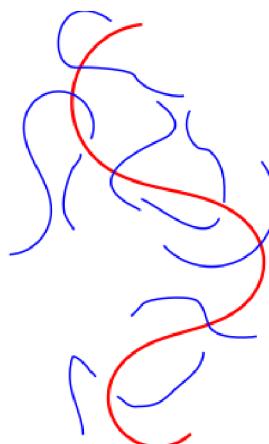
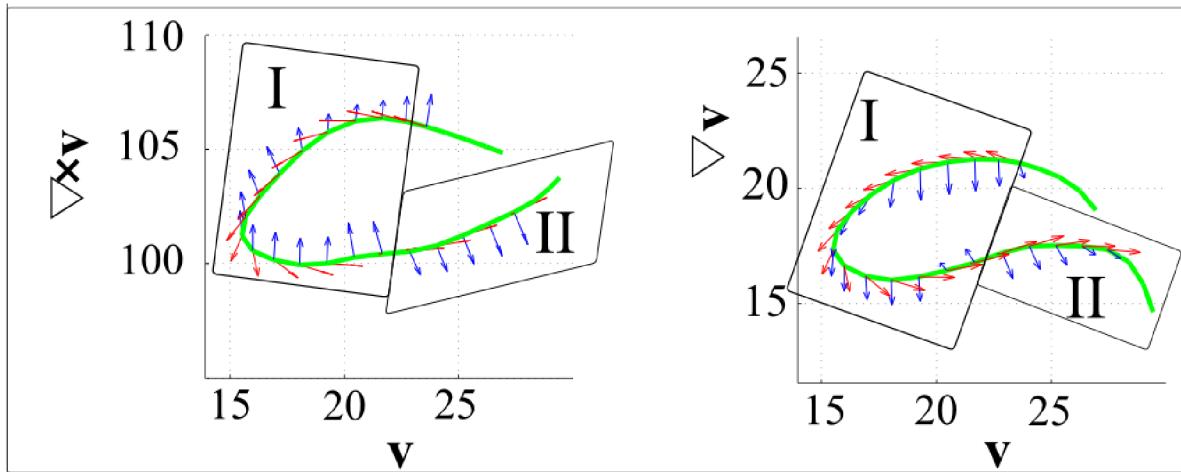
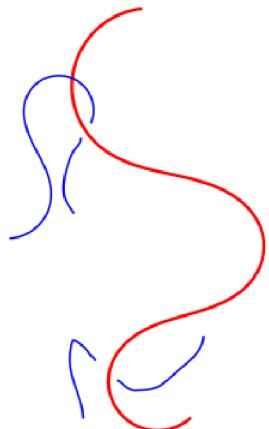
$$D_R^{\text{eff}} = D_R (1 - \varphi / \varphi_{IN})$$

We obtained the I—N spinodal point!

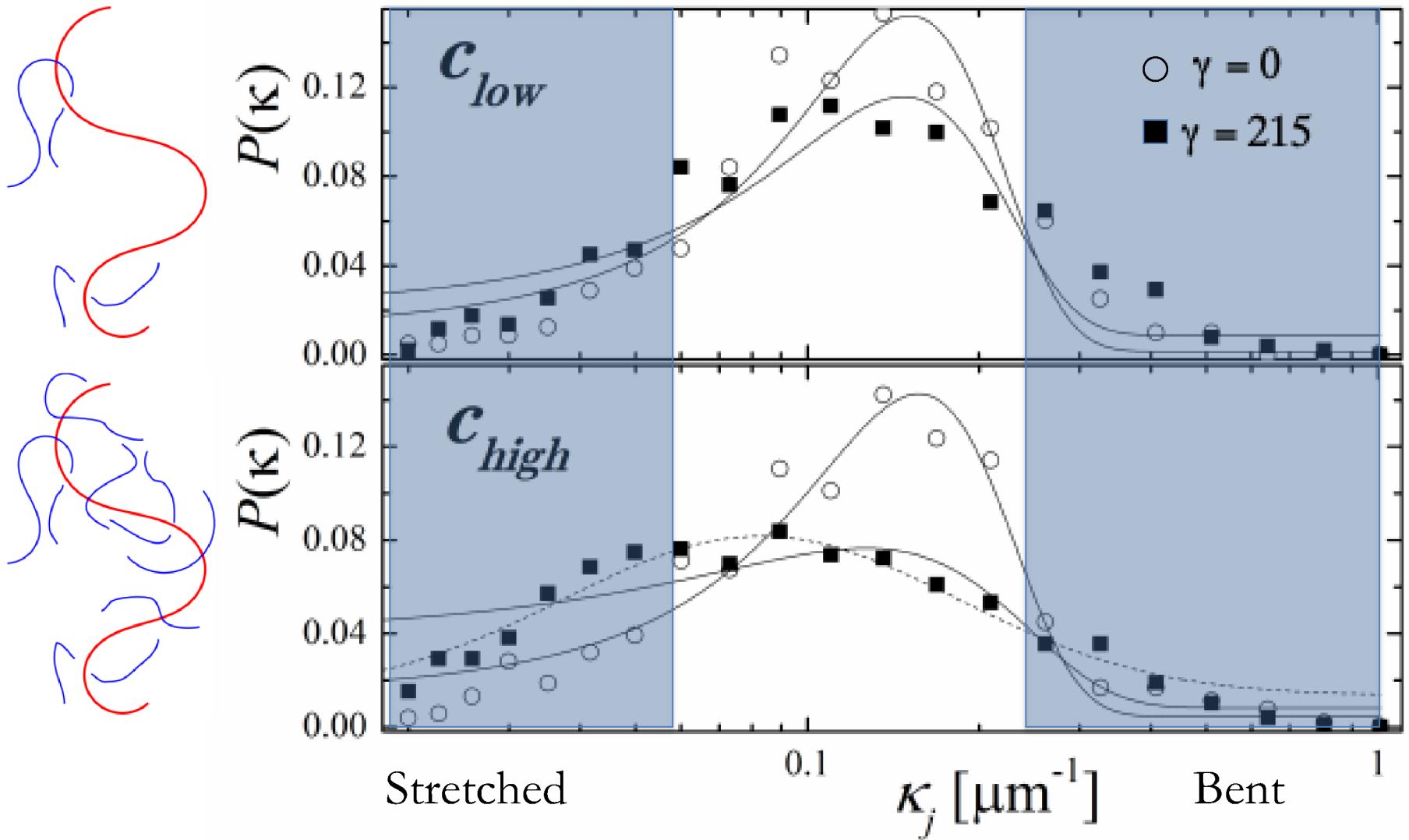
$$\frac{L}{d_{\text{eff}}} \varphi_{IN} = 3.0$$

& $D_R = 0.04 \text{ s}^{-1}$

Typical examples:



Distribution of curvatures:



Characterizing parameters

$$\bar{S}_T = \int_0^\pi \int_0^{2\pi} d\theta d\phi \sin(\phi) f(\theta_T, \phi_T) \hat{T} \hat{T}$$

$$\bar{Q} = \frac{1}{2}(3\bar{S} - \mathbf{I})$$

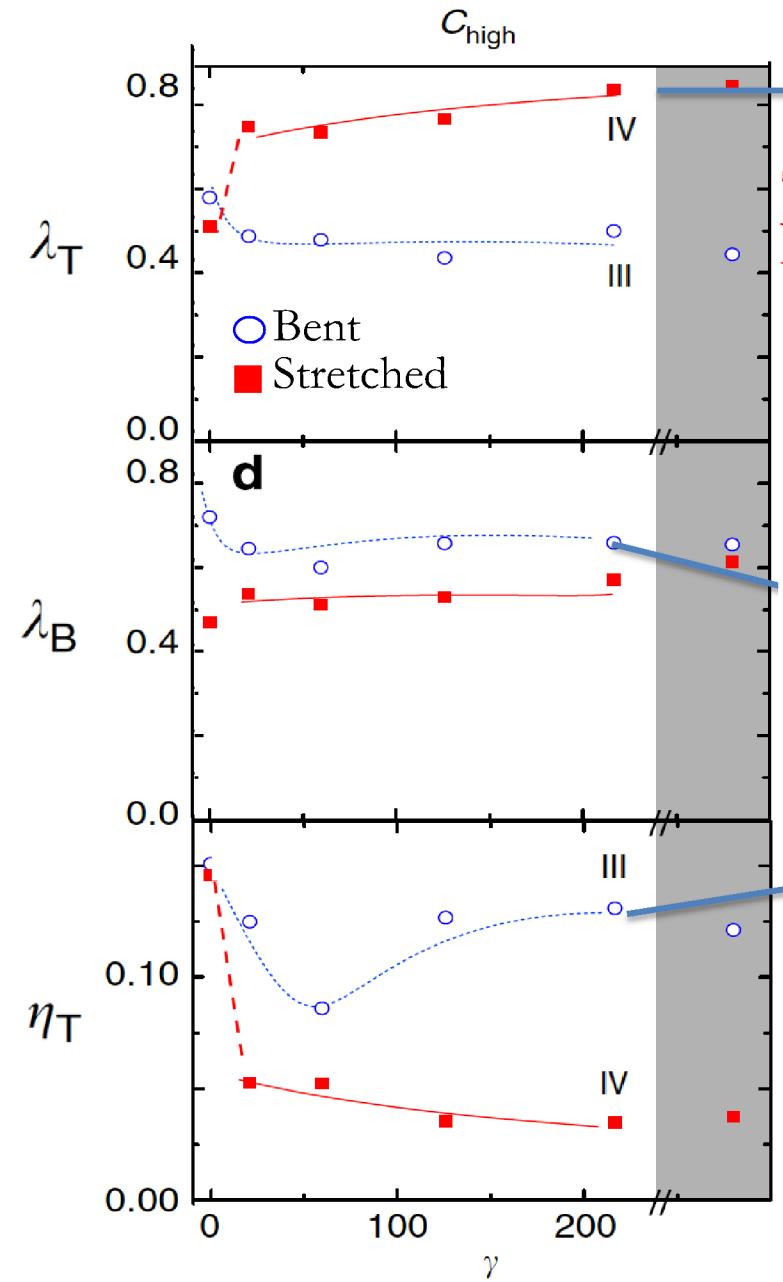
Biaxiality

$$\bar{Q}_{T,B} = \begin{pmatrix} -\frac{1}{2}\lambda_{T,B} - \boxed{\eta_{T,B}} & 0 & 0 \\ 0 & -\frac{1}{2}\lambda_{T,B} + \eta_{T,B} & 0 \\ 0 & 0 & \boxed{\lambda_{T,B}} \end{pmatrix}$$

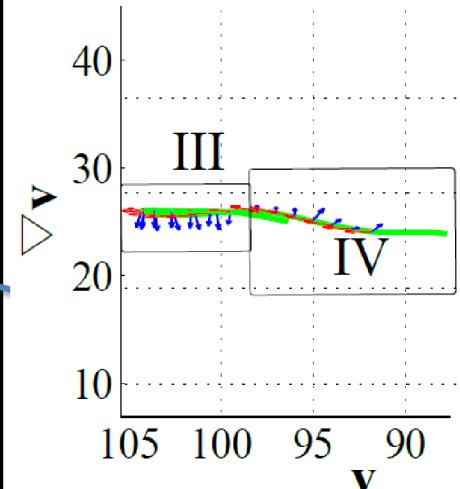
Orientational order parameter

Note: this is the input for calculating stress tensor

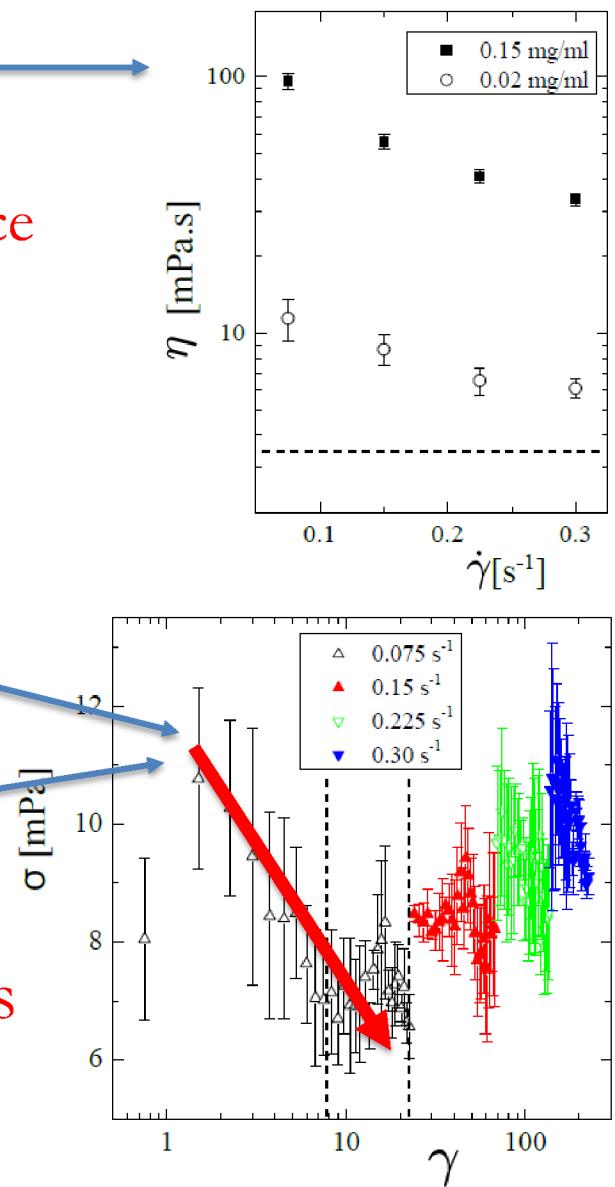
Connection between ordering and stress



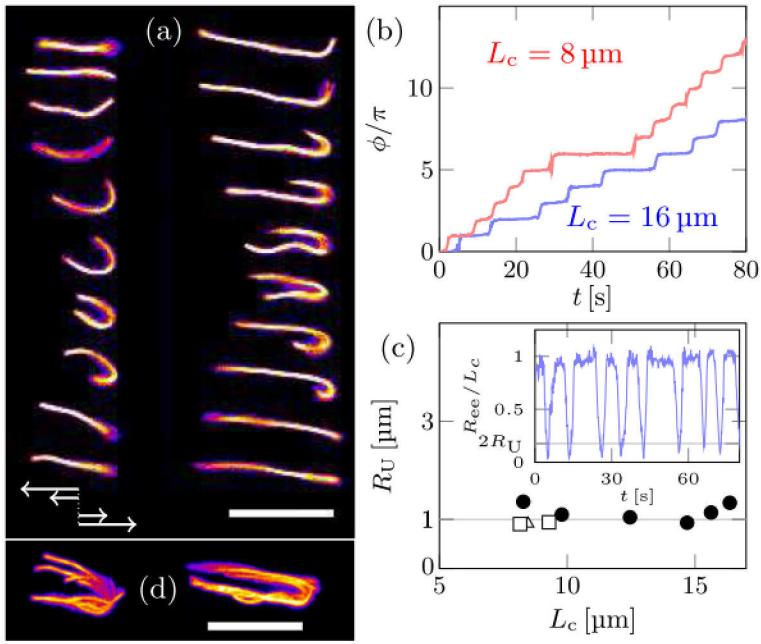
Shear thinning:
Nematic=more space



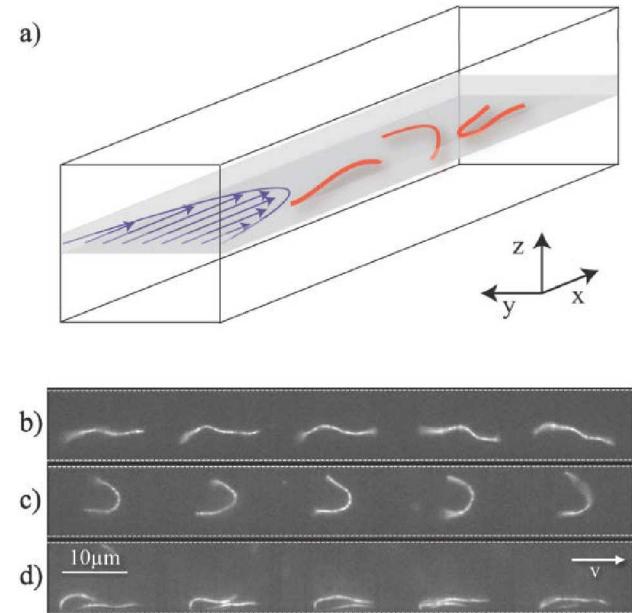
Strain softening :
Oriented hairpins



Shear experiments on F-Actin



Direct Observation of the Dynamics of Semiflexible Polymers in Shear Flow
Harasim et al, PRL, 110 (2013)

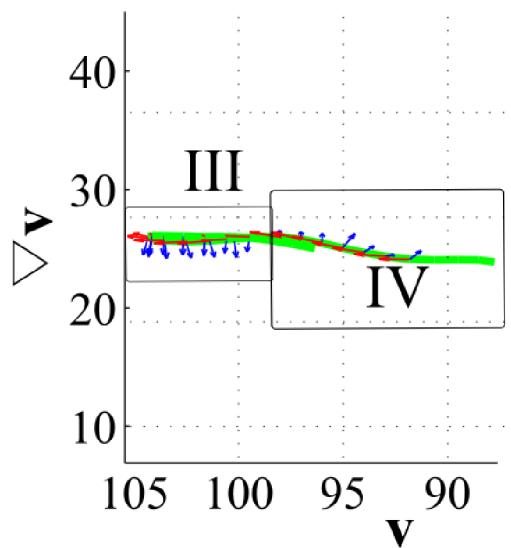
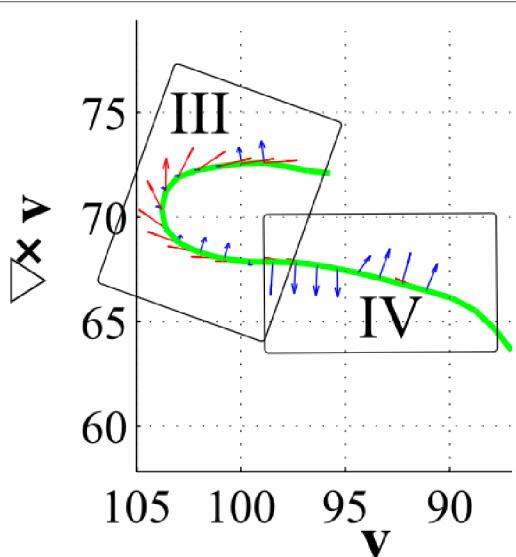
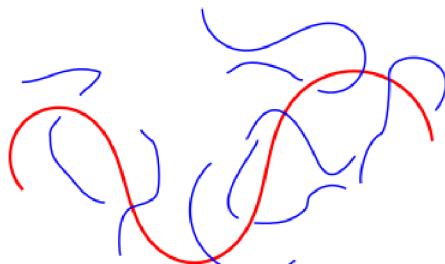


Mobility Gradient Induces Cross-Streamline Migration of Semiflexible Polymers
Steinhauser et al, ACS Macroletters, p. 542 (2012)

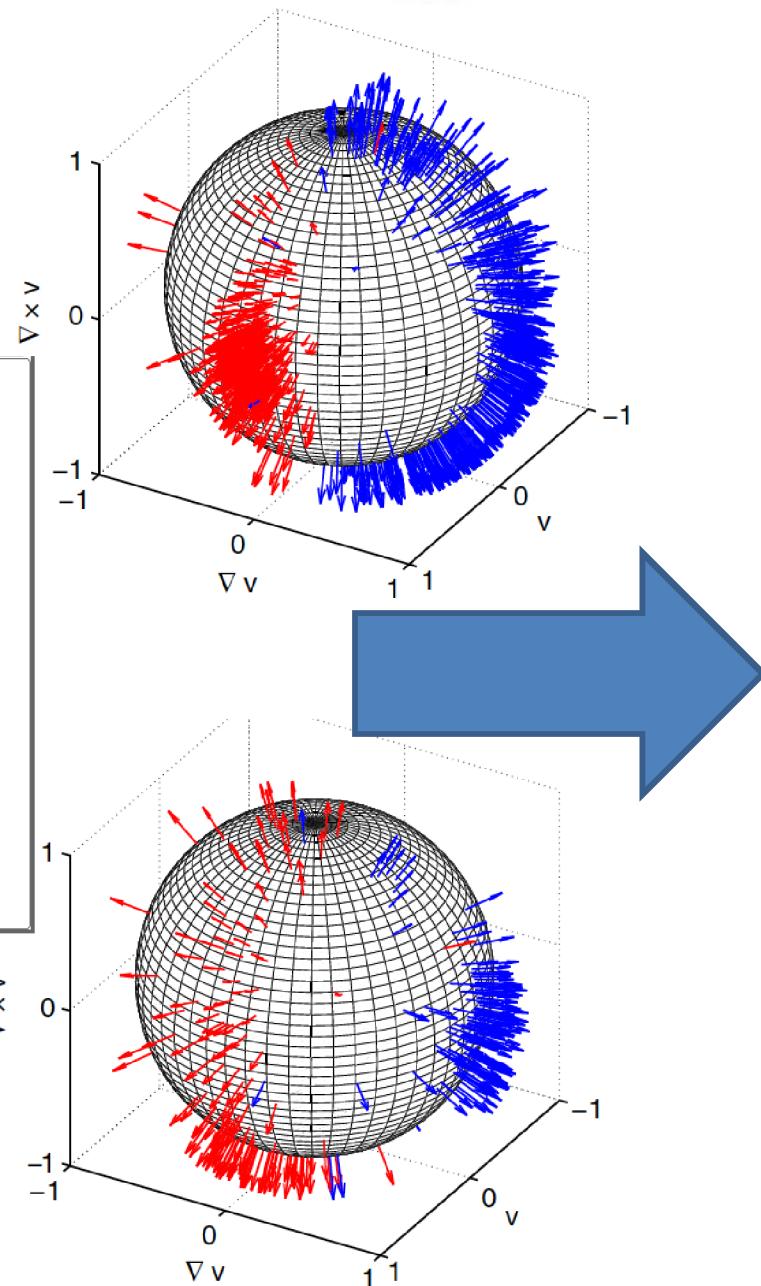
Ill defined geometries; Infinite dilute; 2-D imaging

Distribution of angles

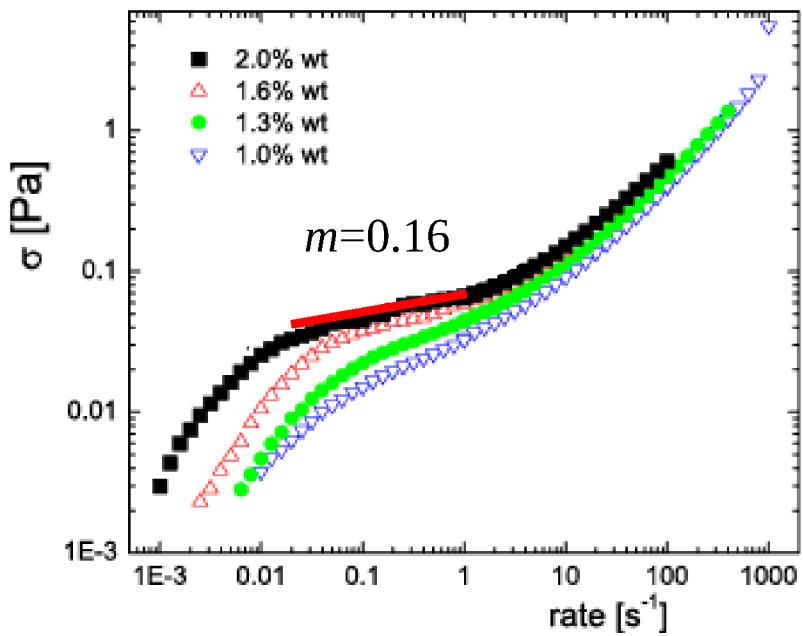
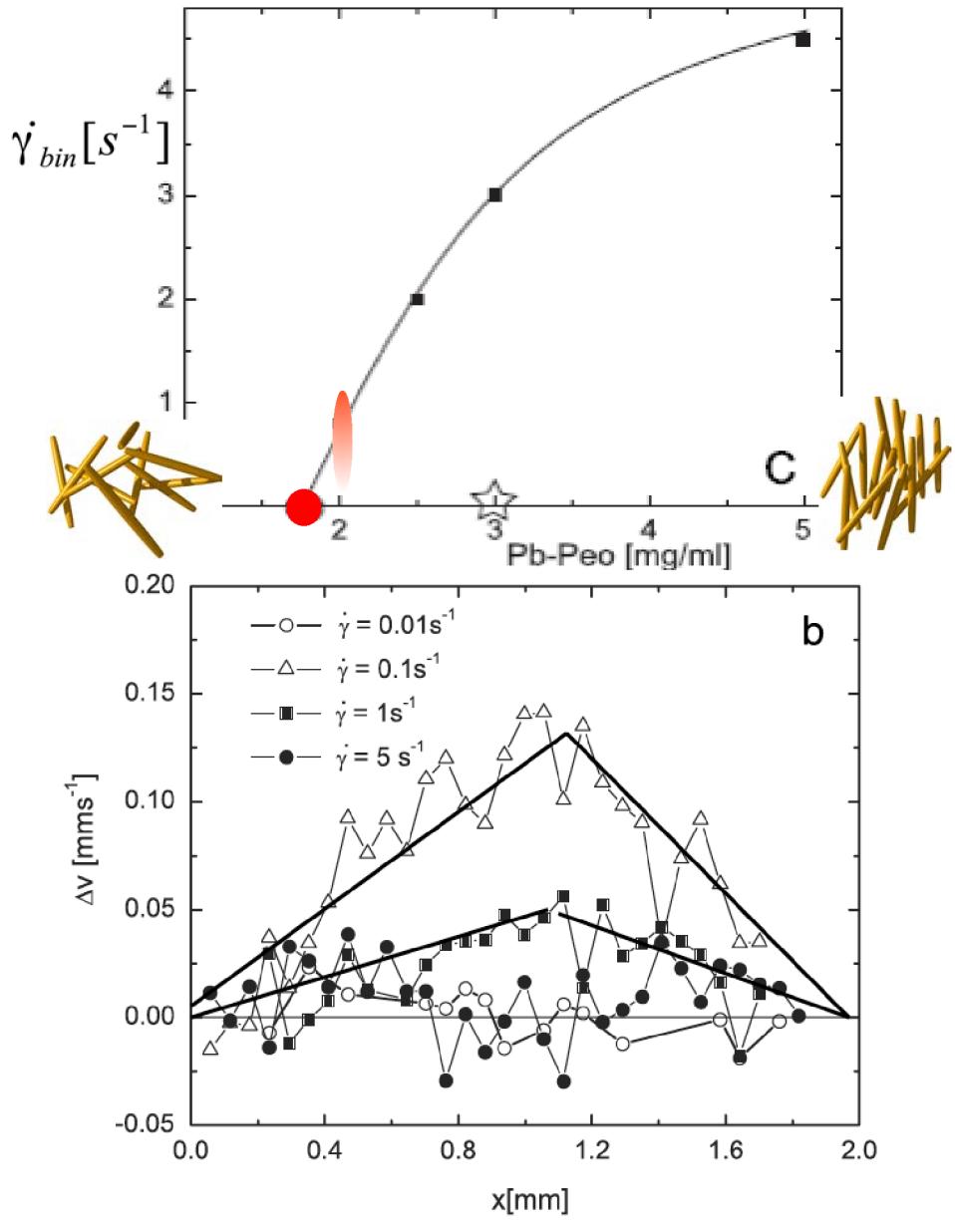
Stretched: IV



Bent: III

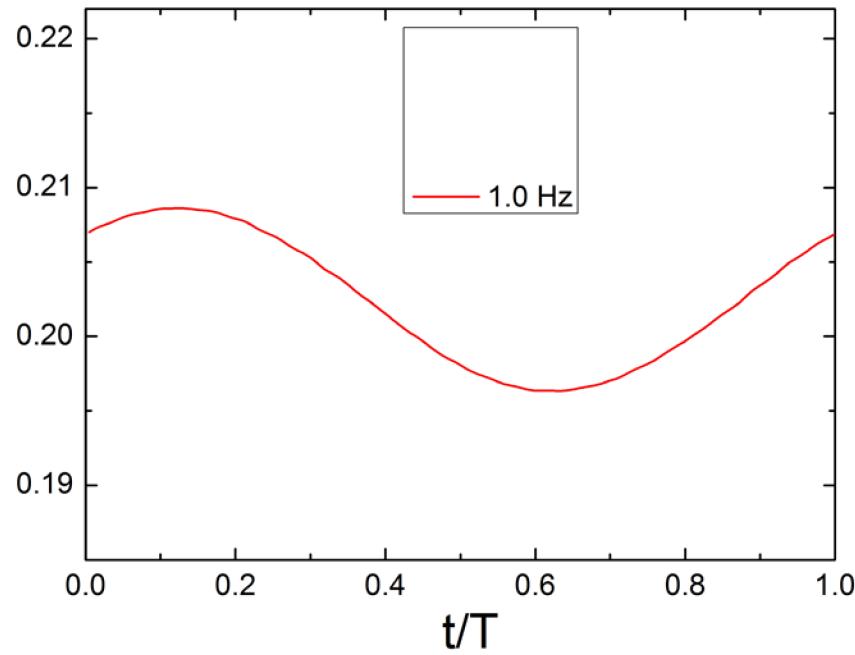
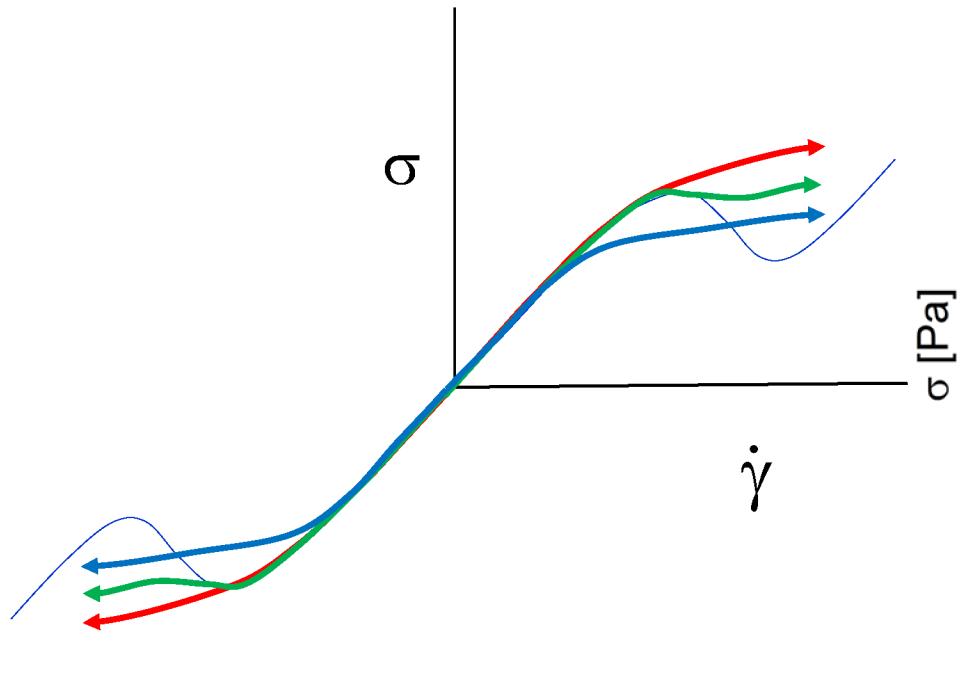


Non-equilibrium isotropic-nematic binodal



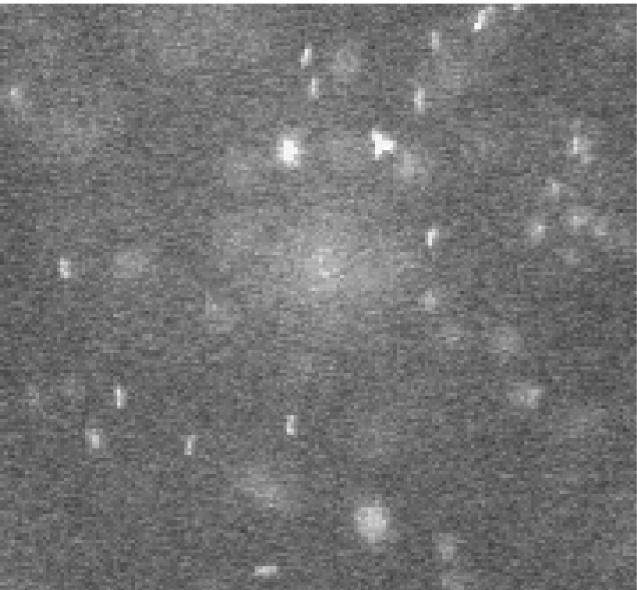
Probe dynamics

Probe dynamics with Large Amplitude Oscillatory Shear

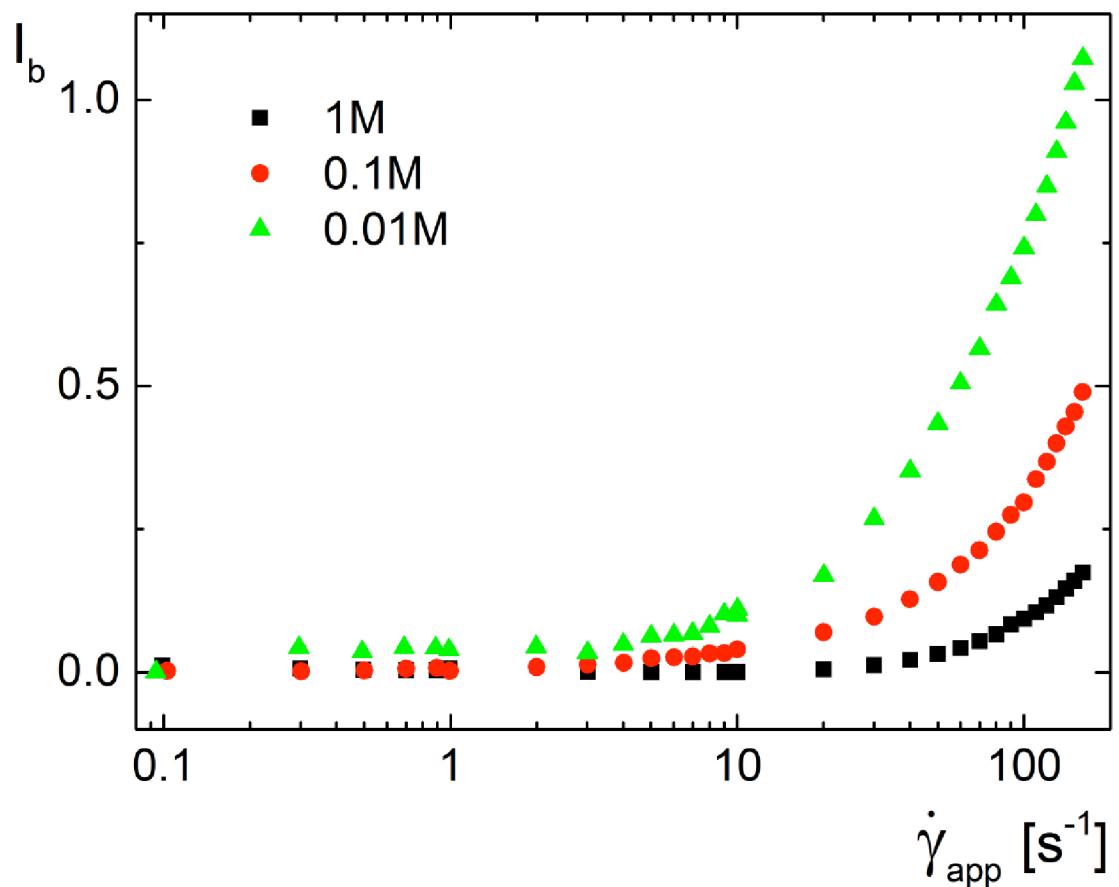
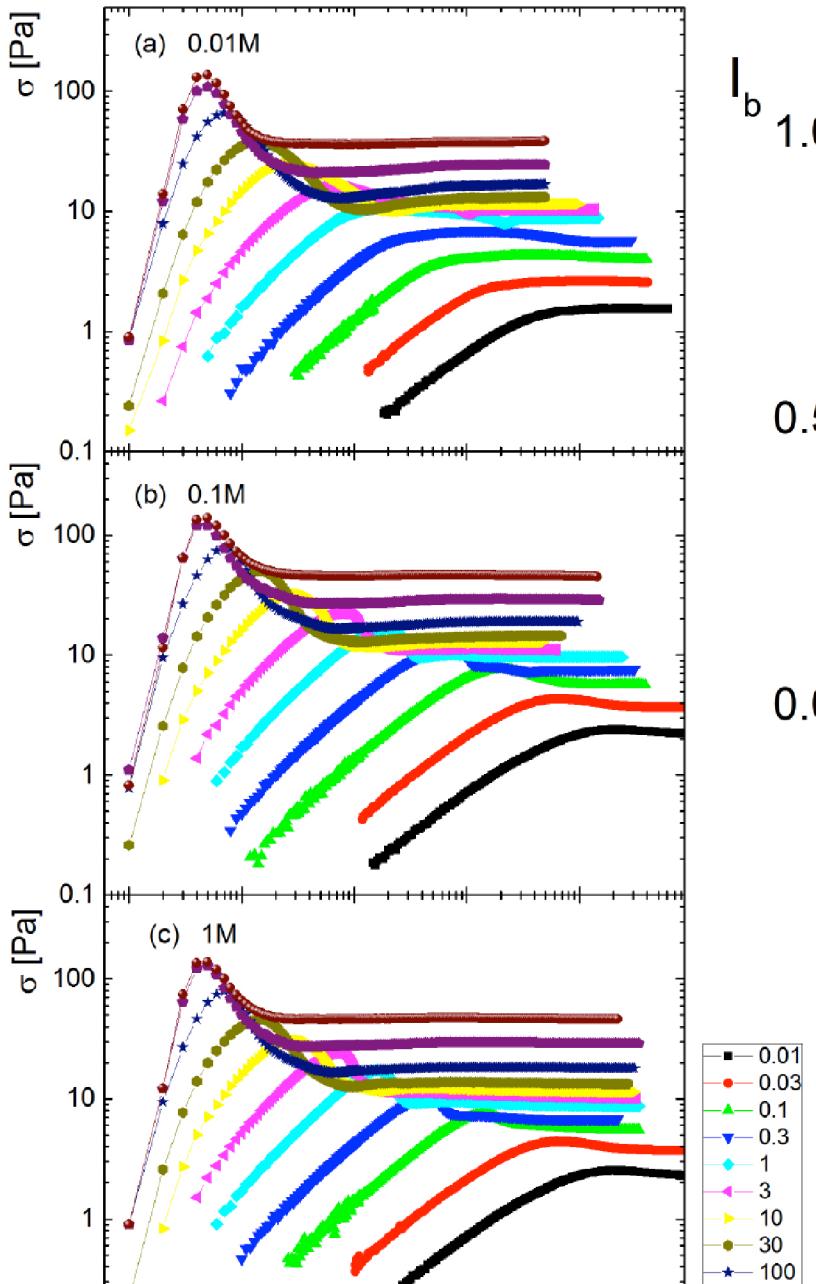


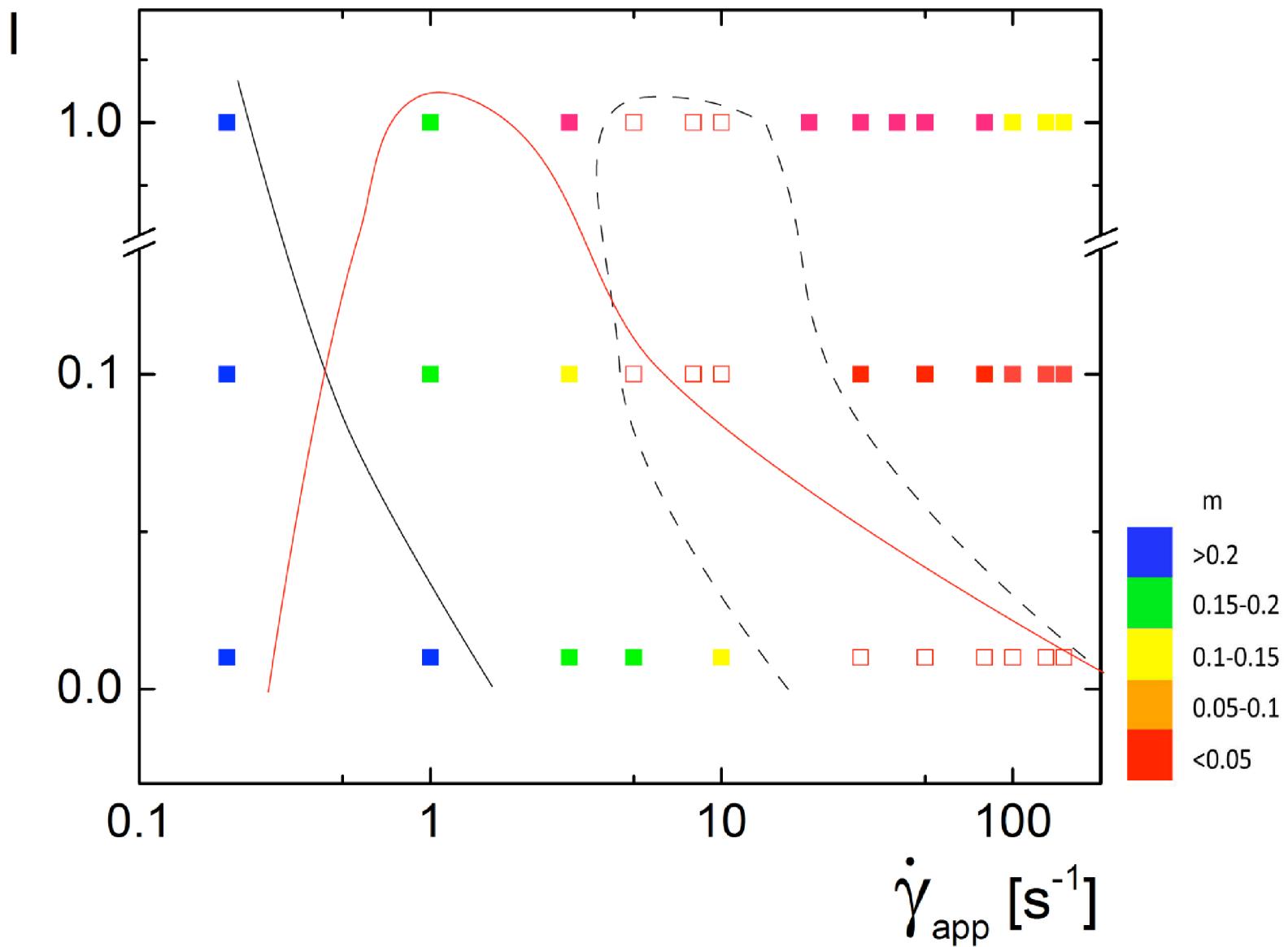
Probe structure with *in situ* scattering methods over broad range of length-scales and time-scales

Complex flow: Complex fluids



start up on DNA at different ionic strengths





We see a reentrant behavior in shear thinning:
Far away from I-N —> nothing
around I-N but flexible —> shear banding
towards ideal rod—> loose it
Ideal—> nothing

Strong shear thinning does not mean that you will get Sms

Systems:
high salat DNA/xanathan
low salt DNA/xanthan AND pb-peo
pf1 / F-actin
fd-y21m

done:

Open:

effect of salt works different directions, comparing DNA with pf1

How does system sustain orientation after disentanglement?

→ shear rate should now be scaled by local rotation motion and not reptation time.

For the how strong is strong question we have that indeed systems that have $m > 0.3$ don't band
Possible reason could be that γ_{high} and γ_{low} are too close to each other.

Suggestion:

stiff rods go into nematic before reaching really high concentration

but

0.7% xanthan is not that high

0.5 mg/ml DNA also not that high

2 mg/ml for pb-Peo

- We see a link between I-N and shear banding

Is it the charge?

Screening charge aids SB for DNA

screening charge reduces SB pf1 (if at all)

PbPeo is uncharged.

—> no

But: both very long contour length!

xanthin, DNA and pb-peo are all long. F-actin also.

Is it the length or is it polydispersity?

What tuning tells us:

collateral understanding: understand stiff polymers and rods

- we got hold on shear thinning using new theory and ideal rods
- We understand shear thinning stiff polymers. No theory!

Hint:

stress overshoot in LAOS when WLMs are overstretched

Shear banding in polymer solutions

Michael Cromer,^{1,2} Michael C. Villet,³ Glenn H. Fredrickson,^{1,2,4}
and L. Gary Leal^{1,4,5}

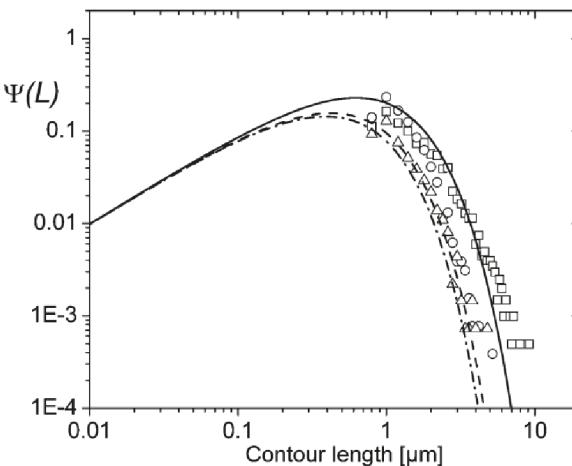


Figure 3. Probability density distribution of contour lengths for different molar fractions of DMF (solid line, square for $f = 0$; dashed line, circle for $f = 0.025$; dotted line, triangle for $f = 0.06$). The curves correspond to an exponential distribution with the parameters determined by DLS. The symbols are the data obtained from microscopy.

Merchant and Rill

DNA Phase Transitions

TABLE 1 Lengths, length distributions, and critical concentrations of DNA samples

| DNA (bp) | Length* (nm) | Range [#] | | SD ^{\$} (bp) | M_w/M_n [¶] | C_c^* (mg/ml) |
|-------------|-----------------|--------------------|-----------|--------------------------|------------------------|--------------------|
| | | (bp) | (nm) | | | |
| 147 | 50 | 135–162 | 46–55 | ±12 | 1.07 | 135 |
| 170 | 58 | 131–210 | 44–71 | ±32 | 1.07 | 122 |
| 336 | 114 | 311–355 | 105–120 | ±19 | 1.01 | 48 |
| 570 | 190 | 257–1140 | 87–386 | NA | 1.23 | 23 |
| 1450 | 490 | 766–2400 | 262–804 | ±690 | 1.14 | 13 |
| 8000 | 2700 | 4k->23k | 1352–7774 | NA | ND | 13 |