

Fig. 1. Schematics of a ferromagnetic thin film with low magnetocrystalline anisotropy in z-direction and closure domains on the surfaces.

36 nm thick FePd layer is grown by shuttered growth at room temperature, followed by a second layer of 34 nm grown by codeposition at  $T_s = 500$  K. The layer thicknesses were measured and calibrated using X-ray reflectometry (XRR) measurements. X-ray diffractometry was used to evaluate the long-range ordering of the FePd films [9]. It is observed that (a) S1 has highest degree of structural ordering, (b) S2 has no long range ordering, and (c) S3 shows moderate long-range ordering.

### 3. Magnetization analysis

Fig. 2 shows the hysteresis loops of the three samples where magnetization  $M$  is plotted as a function of the applied field. It is observed that S1 has a strong PMA along  $\langle 001 \rangle$ -direction, while S2 has a strong in-plane anisotropy. However, in S3 the  $\langle 001 \rangle$  direction still denotes the easy axis but with a higher in-plane component than S1.

Due to the different strength of magnetocrystalline anisotropy in the three samples, we need to consider the effective uniaxial anisotropy constant  $K_{eff}$  as the sum of the magnetocrystalline and shape anisotropy constants  $K_{eff} = K_u + K_{sh}$ .  $K_{eff}$  can be obtained by integrating the area under the hysteresis loops over the difference from the out-of-plane magnetization [10]:

$$K_{eff} = K_u - \frac{1}{2} \mu_0 M_s^2 = \int_0^{M_s} (H_{\perp} - H_{\parallel}) dM,$$

where  $M_s$  is given in [A/m],  $\mu_0 H_s$  in [T] and  $K_{eff}$  in [J/m<sup>3</sup>]. From the hysteresis loops we deduce  $K_u$  to determine the quality factor  $Q$  of the thin film, which can be calculated by the ratio of  $K_u$  and the shape anisotropy constant  $K_{sh} = \frac{1}{2} \mu_0 M_s^2$  [11]. For  $Q > 1$  the thin film has a strong PMA, whereas  $Q < 1$  denotes an in-plane easy axis of magnetization. The  $Q$ -values and  $K_u$  for the three samples are listed in Table 1.

Fig. 3 shows the MFM images of the three samples. The strong PMA in S1 ( $Q = 2.17$ ) results in a maze domain pattern, while S2 has no preferred in-plane orientation [Fig. 3(a)]. S3 shows a moderate PMA with a higher in-plane component than S1.

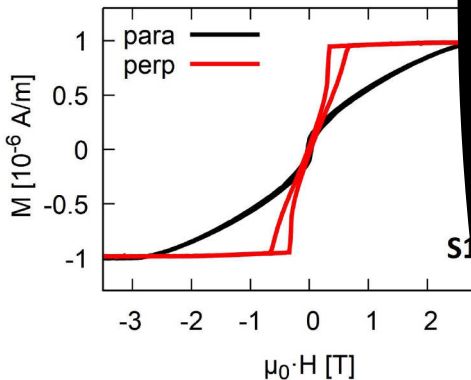


Fig. 2. Hysteresis loops of S1, S2 and S3 measured at 300 K (parallel orientation) and perpendicular to the surface plane ("perp"),

#### Parameters

$K_u$ [kJ/m <sup>3</sup> ]
$Q$

plane magnetization

[Fig. 3(b)].

aligned domain

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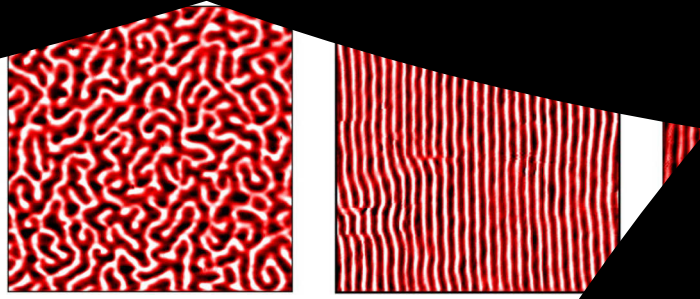


Fig. 3.  $3 \times 3 \mu\text{m}$  MFM measurements in the as grown state of the

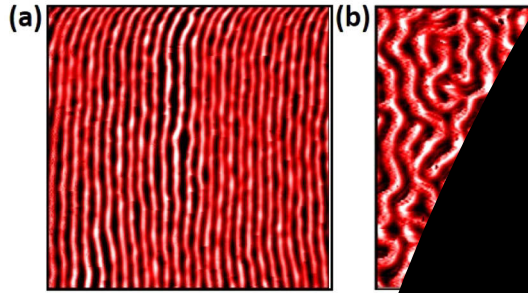


Fig. 4.  $3 \times 3 \mu\text{m}$  MFM measurements at 0 applied field of S3 after application of a 0 applied field (a) after in-plane oscillating demagnetization.

From the resulting detector image [Fig. 5(a)] the following features were observed – (i) the specular spot at  $Q_y = 0 \text{ nm}^{-1}$  and  $Q_z = 0$  with an incident angle of  $0.97^\circ$  and (ii) two peaks on the  $Q_z$  axis with  $Q_z = 0.169 \text{ nm}^{-1}$  (marked by a horizontal line in Fig. 5(a)). The  $Q_y$ - $Q_z$ -map and the intensity as function of  $Q_y$  at  $Q_z = 0.169 \text{ nm}^{-1}$  are shown in Fig. 5(b) and (c).

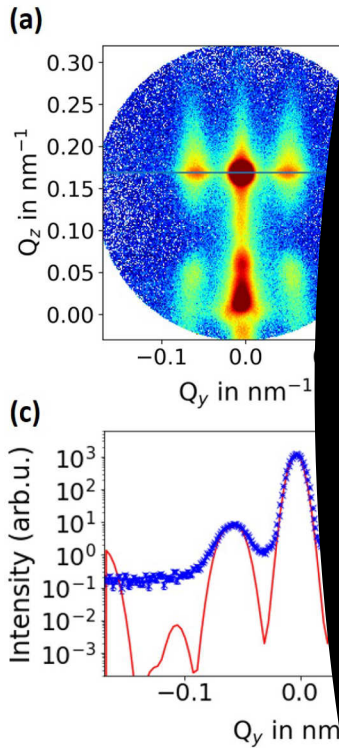


Fig. 5. (a)  $Q_y$ - $Q_z$  scan of FePd with parallelly aligned magnetic domains. (b) Intensity as function of  $Q_y$  at  $Q_z = 0.169 \text{ nm}^{-1}$  of the experiment (blue) and the simulation (red). (c) Intensity as function of  $Q_y$  at  $Q_z = 0.169 \text{ nm}^{-1}$  of the experiment (blue) and the simulation (red). Interpretation of the references to colour in this figure legend is provided in the online version of this article, which is available at <http://www.nature.com/scientificdata/>.

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## References

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- [2] G. Beutier, E. Dudzinski, using Rev
- [3] T
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