

Density variations in liquid tellurium: Roles of rings, chains, and cavities

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Liquid tellurium has been studied by density-functional/molecular-dynamics simulations at 560, 625, 722, and 970 K and by high-energy x-ray diffraction (HEXRD) at 763 K and 973 K. The HEXRD measurements agree very well with earlier neutron-scattering data of Menelle *et al.* The density maximum near the melting point (722 K) reflects the competition between twofold and threefold local coordination, which results in chain formation and changed ring statistics at lower T , and the variation with T of the volume of cavities (26–35 % of the total). A higher-order gradient expansion of the exchange-correlation functional is needed to describe structural details. Changes in the electronic properties (band gap and dc conductivity) upon cooling are consistent with a transition from a high-temperature metal to a semiconductor.

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I. INTRODUCTION

The remarkable properties of liquid elemental tellurium have challenged our understanding for more than 50 years. At 626 ± 3 K there are extrema in the specific heat, thermal-expansion coefficient, compressibility, and related quantities¹ that point to a structural phase transition, and there is a semiconductor-metal (SC-M) transition close to the melting point T_m (722 K). Most striking is the maximum in the density near T_m (Refs. 1 and 2) that is also found in water.³ There is continuing controversy about the origin of this very unusual feature in both materials.⁴ Although water and tellurium differ profoundly in their interactions (hydrogen vs covalent bonds), constituents (atoms vs molecules), and structures, the above thermodynamic quantities are astonishingly similar on temperature scales normalized to the respective melting points.³ Detailed studies of liquid Te could then provide insight into the critical behavior of water, the most important liquid on Earth, particularly because Te can be supercooled to temperatures on this scale well below that at which water crystallizes spontaneously. However, scattering experiments in Te have led to structure factors with considerable variation,^{5,6} and it has been notoriously difficult to calculate structure factors and pair-correlation functions that resemble those measured.^{7–11} This may be due in part to the relatively small samples considered in previous DF/MD simulations (64, 80, and 125 atoms per unit cell in Refs. 8, 9, and 11, respectively).

Crystalline Te comprises parallel helices (bond length 2.835 Å) separated by 3.491 Å,¹² and structural models of liquid Te have been proposed with domains of twofold and threefold coordination, both without¹³ and with^{14–16} chains. A reverse Monte Carlo analysis of liquid Te (Ref. 16) indicated the presence of voids and rings. Furthermore, density-functional (DF) calculations in Te are unusually sensitive to the choice of exchange-correlation functional,^{8,17} and no sat-

isfactory agreement between theory and experiment has been found to date. Here we present high-energy x-ray diffraction (HEXRD) data at 763 and 973 K and the results of DF simulations that agree well with them. The density anomaly and the SC-M transition in liquid Te can be understood in terms of large cavities (voids) and rapid changes in the medium-range structure. This description does not rely on longer-ranged density inhomogeneities, such as the metal/semiconductor domains proposed in Te (Ref. 18) or low-density/high-density regions in water.¹⁹

II. HIGH-ENERGY X-RAY DIFFRACTION

The HEXRD experiments were performed at the beamline BL04B2 at SPring-8 (Ref. 20) using a two-axis diffractometer dedicated to glass, liquid, and amorphous materials and a high- T furnace. The incident energy of x rays was 113.4 keV ($Q_{\max} = 25$ Å⁻¹). The sample was sealed in a silica glass tube (2 ± 0.05 mm inner diameter and 0.21 ± 0.05 mm wall thickness), and diffraction patterns of samples in this tube and an empty tube were measured in vacuum in a transmission geometry. The collected data were corrected using a standard program,²¹ and the absorption coefficients,²² incoherent-scattering functions,²³ and atomic scattering factors²⁴ were used for correction and normalization, leading to the Faber-Ziman total structure factor $S(Q)$.²⁵

The radial distribution function (RDF) or pair-correlation function $g(r)$ is the spherically averaged distribution of interatomic vectors,

$$g(r) = \frac{1}{\rho^2} \left\langle \sum_i \sum_{j \neq i} \delta(\vec{R}_i) \delta(\vec{R}_j - \vec{r}) \right\rangle, \quad (1)$$

where \vec{R}_i are the coordinates of atom i and ρ is the number density. In molecular-dynamics (MD) simulations $g(r)$ can be calculated directly from these coordinates, and experi-

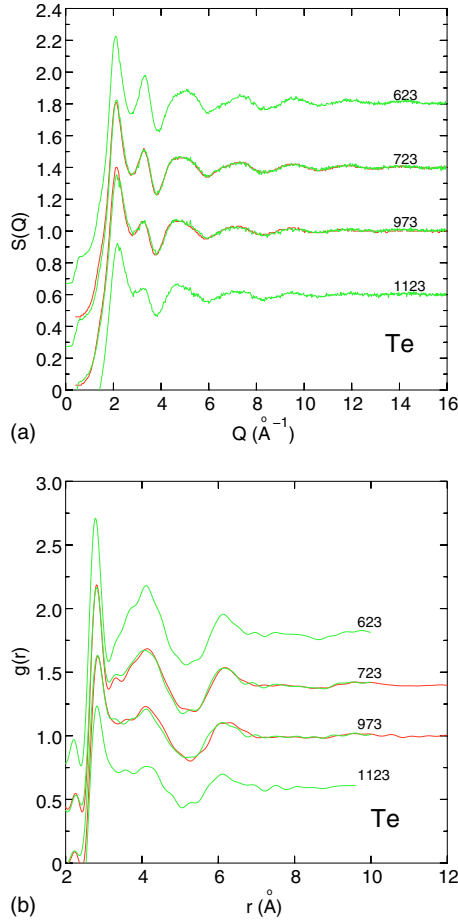


FIG. 1. (Color online) (a) Structure factor $S(Q)$ and (b) radial distribution function $g(r)$ for ℓ -Te. Red (dark gray): present HEXRD data (763 and 973 K); green (light gray): neutron-diffraction data (623, 723, 973, and 1123 K, Ref. 5).

mental values can be determined by Fourier transformation of $S(Q)$,

$$g(r) = 1 + \frac{1}{2\pi^2\rho r} \int_{Q_{\min}}^{Q_{\max}} dQ Q [S(Q) - 1] \sin(Qr). \quad (2)$$

In Fig. 1 we compare our HEXRD measurements of $S(Q)$ and $g(r)$ with neutron-diffraction (ND) results at several temperatures.⁵ A value of Q_{\max} of 16 \AA^{-1} was used in the Fourier transformation in both cases.

III. DENSITY FUNCTIONAL CALCULATIONS

The DF/MD calculations used the CPMD program (Born-Oppenheimer mode),²⁶ as described elsewhere.²⁷ We used time steps of 6.050 fs (initialization) and 3.025 fs (data collection), scalar-relativistic Troullier-Martins pseudopotentials,²⁸ and periodic boundary conditions with a single point ($\mathbf{k}=0$) in the Brillouin zone. The large time steps were made possible by using a stable and efficient predictor-corrector algorithm.²⁹ We apply a Nosé-Hoover-chain thermostat³⁰ (frequency 800 cm^{-1} , chain length 4), and the rms error in the temperature during the simulations ranged

TABLE I. Structure of crystalline Te (hexagonal cell) for different E_{xc} functionals. a , c : lattice parameters; r_0 , r_d : intra- and inter-chain bond lengths; α : intrachain bond angle; E_c : cohesive energy; and E_d : interchain component of E_c .

	a (Å)	c (Å)	r_0 (Å)	r_d (Å)	α (deg.)	E_c (eV)	E_d (eV)
LDA	4.29	5.87	2.90	3.30	100.4	3.28	0.48
PBE	4.51	5.89	2.90	3.48	100.8	2.42	0.19
PBESol	5.04	5.84	2.81	3.95	102.6	3.12	0.05
BLYP	5.11	5.91	2.84	4.01	102.8	2.00	0.04
TPSS	4.55	5.90	2.85	3.55	102.4	2.28	0.12
Exp. ^a	4.456	5.921	2.835	3.491	103.1	2.06 ^b	

^aReference 12.

^bReference 39.

from 26 to 43 K between 560 and 975 K. The Perdew-Burke-Ernzerhof (PBE) (Ref. 31) and TPSS (Ref. 32) approximations were employed for the exchange-correlation energy E_{xc} . The kinetic-energy cutoff of the plane-wave basis was 20 Ry (40 Ry for c -Te, Table I), and the minimization of the energy functional during the simulations used the direct inversion of the iterative subspace. A detailed analysis of the electronic structure for selected geometries (snapshots) has been carried out with the Lanczos diagonalization scheme and the free-energy functional of Alavi *et al.* ($T=1000 \text{ K}$).³³ We have used the experimental density for liquid Te (Ref. 1) (Table II) at each T .

Table I summarizes the properties of crystalline Te for different E_{xc} functionals. TPSS results are consistently good, while the local-density approximation (LDA) overestimates the cohesive energy and leads to a short interchain distance, and PBE overestimates the bond length r_0 . We also show the

TABLE II. Calculated (TPSS) properties of ℓ -Te. r_0 (r_{\min}): first maximum (minimum) in RDF; N : total coordination number (cutoff 3.2 \AA); C_2 : fraction of twofold (including onefold) coordinated Te; $\langle \ell(C_2) \rangle$: chain length; V_c : fraction of cavities; $\langle v \rangle$: average cavity size; $W(\epsilon_F)$: DOS at the Fermi level (relative to 970 K); $\sigma(0)$: dc conductivity ($\omega=0$); P : pressure; and D : diffusion constant.

Temperature	560 K	625 K	722 K	970 K
ρ (atoms/ \AA^3)	0.02618	0.02653	0.02725	0.02653
r_0 (Å)	2.82	2.84	2.87 (2.98 ^a)	2.91
r_{\min} (Å)	3.28	3.60	3.71	3.77
$N(\text{Te})$	2.39	2.39	2.45 (2.58 ^a)	2.40
C_2 (%)	66.7	63.8	59.0	60.9
$\langle \ell(C_2) \rangle$	2.80	2.48	2.15	2.07
V_c (%)	34.8	32.3	26.2	34.4
$\langle v \rangle$ (Å ³)	80.1	76.7	67.4	78.3
$W(\epsilon_F)$	0.57	0.75	0.81	1.00
$\sigma(0)$ ($\Omega^{-1} \text{ cm}^{-1}$)	800	1400	2400	2700
P (kbar)	1.4 ± 8.1	3.8 ± 9.4	-2.3 ± 9.7	-7.6 ± 9.9
D ($10^{-5} \text{ cm}^2/\text{s}$)	0.89	1.54	2.40 (2.69 ^a)	4.62

^aSeparate simulations of 30 ps with PBE functional.

structural parameters found using the PBEsol (Ref. 34) and BLYP approximations.^{35,36} PBEsol has the same form as PBE, but with parameters optimized for extended systems. While the PBEsol description of bond distances (lattice constants) is generally improved for chalcogenide alloys,^{37,38} it overestimates significantly the interchain distance and cohesive energy of *c*-Te. The BLYP form leads to poor results for the lattice constant *a* and the interchain distance, which arises from a shallow potential-energy minimum, but other results are satisfactory.

The simulation of *ℓ*-Te was started at 3000 K on a 343-atom s.c. geometry in a cubic cell (side 23.47 Å) appropriate to the density at 970 K. After cooling (110 ps, PBE functional) and data collection at 970 K (10 ps, TPSS), the temperature was lowered to *T_m* (722 K, 86 ps, PBE), where data were collected (18 ps, TPSS). The samples were initialized at each temperature (2 ps, TPSS) prior to data collection. The procedure was repeated for 625 K (structural phase transition) and 560 K (density minimum), where the times for cooling (PBE) and data collection (TPSS) were 60/60 ps and 11/10 ps, respectively. The total simulation time was 413 ps, and the results are summarized in Table II.

The pressure on the (fixed) cell was small throughout, and its sign was consistent with the density changes. The dependence of the pressure (stress) on the cutoff in the plane-wave basis has been tested by performing calculations with cutoffs of 30 Ry and 40 Ry for three selected snapshots at each temperature. The calculated pressures were within the error bars quoted in Table II in all cases. The diffusion constant *D* can be calculated from the velocity-velocity autocorrelation function *C_v*(*t*)

$$D = \frac{1}{3} \int_0^\infty dt C_v(t) \quad (3)$$

or directly from the coordinates \vec{R}

$$D = \lim_{t \rightarrow \infty} \frac{\langle |\vec{R}(t) - \vec{R}(0)|^2 \rangle}{6t}. \quad (4)$$

IV. RESULTS

A. Structure factor, pair distribution function, and coordination numbers

The structure factors *S*(*Q*) measured by HEXRD at 763 and 973 K agree remarkably well with the ND results of Menelle *et al.*⁵ [see Fig. 1(a)]. This is an important result since previous scattering measurements on liquid Te have not led to a consistent picture. Apart from minor discrepancies in the peak heights at 2.2 and 3.2 Å⁻¹ and the shape of the third maximum at 4.5 Å⁻¹, the TPSS calculations show the best agreement with experiment reported to date for DF methods, although the differences in *S*(*Q*) increase as *T* are lowered to 625 K [see Fig. 2(a)]. This probably reflects the lower atomic mobility that requires even longer simulations. The PBE result is less satisfactory: the second maximum is shifted and broadened, and the phase of the high-*Q* oscillations differs from experiment.

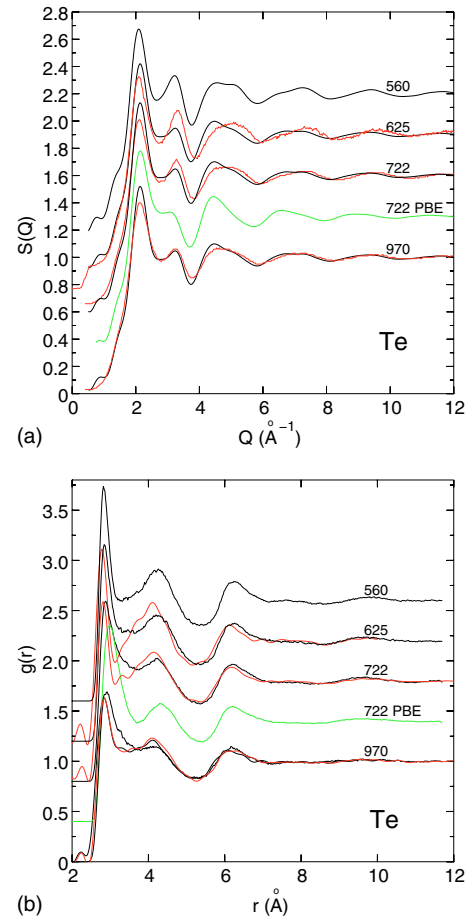


FIG. 2. (Color online) (a) Structure factor *S*(*Q*) of *ℓ*-Te at 560, 625, 722, and 970 K (TPSS calculation), with XRD data for 763 and 973 K, and neutron-diffraction data for 623 K [red (dark gray)]. The PBE result for 722 K [green (light gray)] is also shown. (b) Corresponding radial distribution functions *g*(*r*).

The differences in *S*(*Q*) are reflected in the radial distribution functions [RDF, Fig. 2(b)]. The experimental RDF show narrow first peaks at 722 (2.80 Å) and 970 K (2.83 Å), and a minimum near 3.2 Å. The most prominent features (narrow first peak, increased weight in the second peak) are present in the TPSS calculations, but the first peak is broader and the bonds ~2% longer, and there is more weight at interchain distances 3.2–3.7 Å. The distributions of near-neighbor separations (Fig. 3) show that the first two

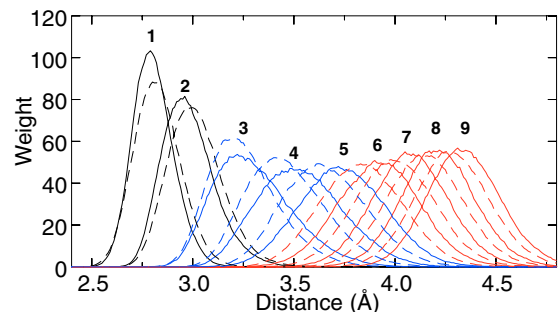


FIG. 3. (Color online) Nearest-neighbor distributions of Te atoms in *ℓ*-Te at 560 K (full curves) and 722 K (dashed curves).

TABLE III. Calculated coordination numbers as a function of cutoff R_c , with HEXRD values (exp) at 763 and 973 K.

R_c (Å)	560 K	625 K	722 K	763 K (exp)	970 K	973 K (exp)
3.1	2.04	2.00	1.99	2.13	1.94	1.92
3.2	2.39	2.39	2.45	2.49	2.40	2.34
3.3	2.74	2.78	2.91	2.87	2.86	2.73
3.4	3.13	3.19	3.37	3.32	3.28	3.15

(2.80 and 2.98 Å), which comprise the first peak of the RDF at 560 K, are significantly shorter than the interchain distances. The short bonds are longer and interchain separations shorter at 722 K so that the position of the first minimum in $g(r)$ shows a pronounced T dependence as the overlap increases.

Integration over the first peak of $g(r)$ to the experimental minimum (3.2 Å) yields TPSS coordination numbers $N(\text{Te})$ at 722 and 970 K (2.45 and 2.40) close to the HEXRD results (2.49 and 2.34). The PBE results show longer bonds and overcoordination, a tendency also found in Te clusters.⁴⁰ If we classify the atomic coordination as twofold (or less, C_2) or threefold (or higher, C_3), the former contribute to chains analogous to the helices in the crystal. They form the majority at each T (see the average coordination numbers in Table II), and their fraction is highest (66.7%) at 560 K. For completeness we provide in Table III the coordination numbers calculated for cutoffs between 3.1 and 3.4 Å. Also

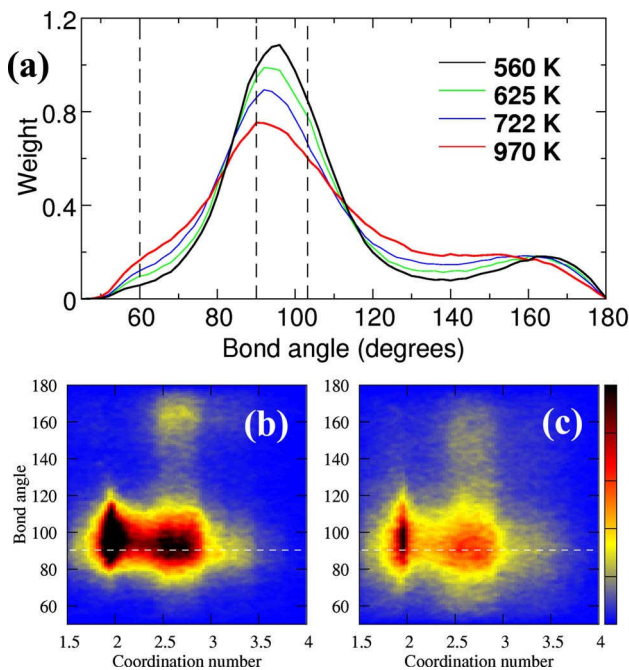
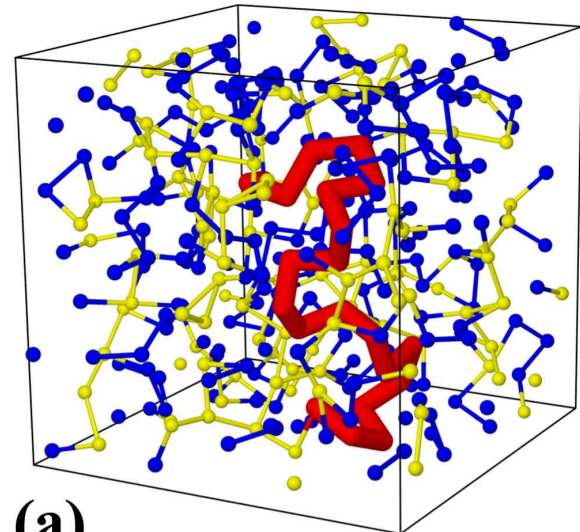
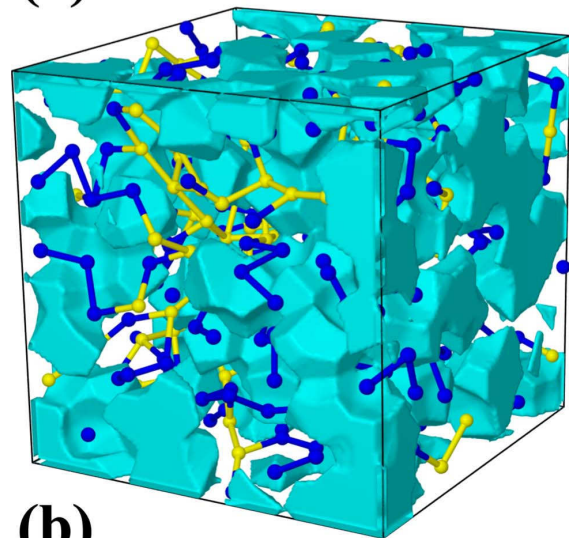


FIG. 4. (Color online) (a) Angular distributions at 560 (sharpest peak), 625, 722, and 970 K. Dashed lines at 60, 90, and 103.2° are for atomic close packing, octahedral (cubic) coordination, and helices in c -Te, respectively. The bond cutoff is 3.2 Å. (b) and (c) Angular distributions at 560 and 970 K as a function of coordination number (smooth cutoff around 3.2 Å). Dashed white line: 90°.



(a)



(b)

FIG. 5. (Color online) Chains and cavities in snapshot of ℓ -Te at 560 K: (a) a 21-membered chain of Te (red); (b) cavities (cyan, 30.1% of volume). Blue: twofold (or onefold) atoms; yellow: threefold (or higher) coordinated atoms.

shown are the coordination numbers obtained from reverse Monte Carlo analyses of the HEXRD data at 763 K and 973 K.

B. Bond angle distributions

The bond angle distributions [Fig. 4(a)] show octahedral features, with a strong peak at 90° and a weaker peak at 180°. Both broaden gradually as T increases and provide few signs of the density anomaly. The maximum shifts from 96 to 90° upon heating from 560 to 970 K, and triangular configurations (bond angles near 60°) are more evident. Figures 4(b) and 4(c) show that twofold atoms have strongly peaked distributions near 100° and few linear configurations, while Te atoms with higher coordination show octahedral bond angles. The distribution at 970 K is significantly broader, but the narrow range of twofold Te atoms remains evident. We

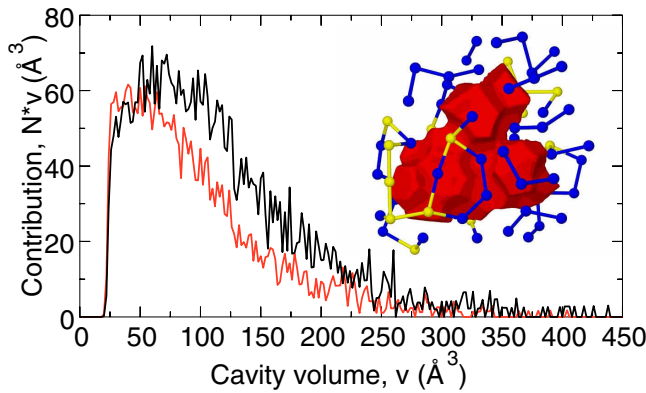


FIG. 6. (Color online) Cavity volumes at 560 K (black, upper curve) and 722 K (red) weighted by their contribution in total volume. Inset: large cavity and environment (333 Å³, 560 K).

find no evidence of short-long alternation in the intrachain bonds, and they do not prefer linear configurations.^{7,11}

C. Chains, rings, and cavities

Information about the density anomaly point requires simulation cells large enough to describe chains and cavities [Figs. 5(a) and 5(b)], i.e., beyond the coordination numbers, bond lengths, and bond angles accessible to small samples. Chains are characteristic of liquid Te and can be terminated either by a threefold Te (not included in the chain) or a onefold Te (included) and can have over 20 links at 560 K [Fig. 5(a)]. The bond angles in such chains are similar to those in *c*-Te [Figs. 4(b) and 4(c)], but the interchain distances vary widely and allow numerous empty regions: “cavities” (voids, vacancies).

Cavities have been analyzed by determining points that are farther from *any* atom than a given cutoff (here 2.8 Å) and building cells around them according to the Voronoi prescription.²⁷ A similar method has been applied recently to study cavities in amorphous and liquid Sb₂Te₃.⁴¹ Cavities in Te have complicated shapes, often far from spherical,¹⁶ and occupy between 26% and 35% of the total volume (Table II), depending on *T*. Their volume contributions as a function of size are shown in Fig. 6 for at 560 K (density minimum) and 722 K (density maximum). Cavities with up to 10 atomic volumes (~ 300 Å³) are not uncommon. Larger cavities (50–250 Å³) are more abundant at 560 K, where the total cavity volume is 8.6% larger than at 722 K. Since this increase is greater than the density change (4.1%), the volume ascribed to atoms *increases* at the maximum density (722 K).

The distributions of irreducible rings, i.e. closed paths along bonds, differ dramatically at 560 and 722 K [Fig. 7(a)]. Larger rings dominate at 560 K, small and intermediate rings (<20) at 722 K. Most twofold coordinated atoms occur in very short chains [Fig. 7(b)], but the chains are longer at 560 K. The average coordination numbers (560 K: 2.39; 722 K: 2.45) are very similar, however, so that the short-range structures must differ significantly. The increased tendency to chain formation at lower *T* is reflected in fewer small rings, and threefold coordinated atoms cluster and form branching sites between the entangled chains/rings [Fig. 7(c)]. These

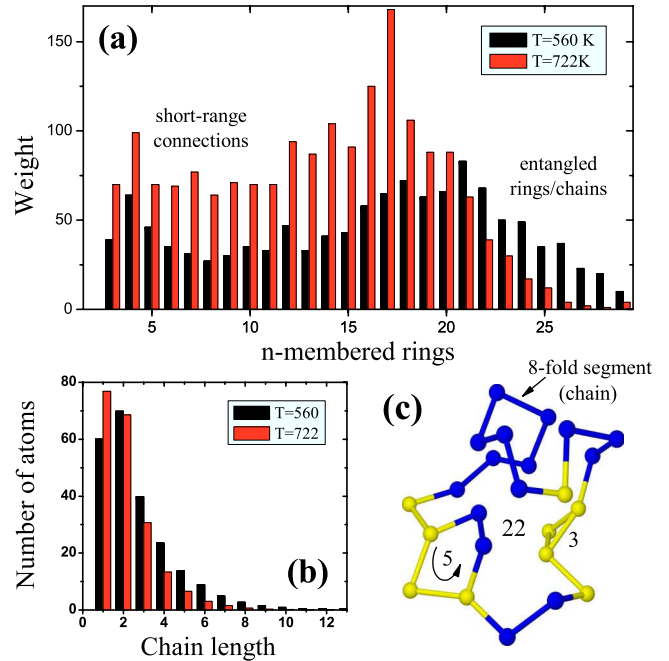


FIG. 7. (Color online) (a) Statistics of irreducible *n*-membered rings at 560 and 722 K. (b) Size distribution of chains. (c) A large, partially entangled, ring of 22 members with two smaller rings, and chains (blue atoms). Yellow: atoms with higher coordination.

changes are closely related to the cavity distribution, as many rings surround cavities. There are more small rings and fewer large cavities at 722 K.

Ring statistics and the cavity analysis provide information about the probability that two atoms in a network are connected by bonds. Large cavities—especially those with irregular shapes—reduce such connections, and more highly coordinated atoms increase them by acting as branching sites. Classical simulations using the TIP5P force field⁴² indicate that cavity volumes, ring sizes, and changes in the local coordination are also crucial to understand the anomalous density variations of *water* at low *T*. This work is being extended and will be described elsewhere. Such analyses go beyond the short-range atomic arrangement, and several order parameters have been used (e.g., orientational and translational order).⁴³

D. Diffusion

The diffusion constant *D* has been evaluated directly from the motion of the atoms (Eq. (4)) and is given for four temperatures in Table II. Measurements of the self-diffusion coefficient in Te have been made using neutron scattering,⁴⁴ where the broadening of the elastic peak caused by diffusion of the atoms in a liquid can be related to *D*,⁴⁵ and by observing the motion of radioactive tracer atoms (such as Te¹²⁷) in a capillary geometry.⁴⁶

The experimental data are characterized by substantial error bars and difficulties in extrapolating values measured at higher temperatures to the melting point. Nevertheless, the agreement between experiment and calculated values is reasonable: quasielastic neutron scattering gives a value of *D* of

$2.6 \pm 0.6 \times 10^{-5}$ cm²/s at 723 K (our value is 2.40), capillary data extrapolated to T_m give $D=1.8$ in the same units.

E. Electronic properties

The Kohn-Sham eigenvalues ψ_i and eigenfunctions ε_i differ from the many-particle wave functions of states of the system, and calculations of electronic excitations using them often lead to discrepancies with measured spectra. It is well known, for example, that many approximations for E_{xc} lead to underestimates of the band gap in semiconductors and insulators. Nevertheless, such calculations often lead to qualitative insight into the electronic properties and are used here in this spirit. In Te, a gap in the electron density of states (DOS) emerges gradually on cooling, and this is monitored in Table II as the ratio of the DOS at the Fermi energy $W(\varepsilon_F)$ to its value for the metallic system at 970 K. A similar T dependence of the DOS was found by Zhao *et al.*¹¹

The eigenvalues ε_i and eigenfunctions ψ_i can also be used to estimate the conductivity using the Kubo-Greenwood expression⁴⁷

$$\sigma(\omega) = \frac{2\pi e^2}{3m^2V\omega} \sum_{i,j} (f_i - f_j) |\langle \psi_i | \vec{p} | \psi_j \rangle|^2 \delta(\varepsilon_i - \varepsilon_j - \hbar\omega), \quad (5)$$

where the f_i denote occupation numbers, and V the cell volume. The finite size of the unit cell means that the eigenvalue spectrum is discrete, and the dc conductivity is found by polynomial extrapolation of $\omega \rightarrow 0$. The decrease in this estimate of the dc conductivity from 2700 (at 970 K) to 800 Ω^{-1} cm⁻¹ (at 560 K) (Table II) is consistent with the increasing band gap.

V. CONCLUDING REMARKS

DF/MD simulations (343 atoms, over 400 ps) on liquid Te at 560, 625, 722, and 970 K show that the TPSS functional³²

describes Te better than other approximations in widespread use. The TPSS approximation was designed to give good results for molecules and solids, and our positive experience with a molecular solid is consistent with the original expectations.^{32,48} New HEXRD measurements at 763 K and 973 K agree well with previous ND data.⁵ This is an important result, because earlier scattering data on liquid Te had not led to a consistent picture.

The rapid changes in the atomistic structure that occur in liquid Te can be understood without invoking the existence of domains. Twofold and threefold coordinated atoms are present, and the density anomaly arises from temperature-dependent changes in chain lengths, ring distributions, and cavity volumes. The supercooled liquid at 560 K resembles *c*-Te, but the absence of translational order leads to cavities in the interchain regions and the density minimum. Preliminary calculations indicate that cavities are also essential to understand the density variations in water. In Te, DF simulations on samples with several hundred atoms are needed to understand these features, since the short-range structure (coordination number, bond lengths and angles) does not suffice. The SC-M transition, evident in changes in both the band gap and the dc conductivity, is related to the broadening of bond angle range at higher T also found in amorphous Ge₂Sb₂Te₅.⁴⁹

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